Tailoring of YIG film properties via compositional tuning by multi-beam pulsed laser deposition

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Abstract: We report an investigation of the effects of variation of composition on the properties of YIG (yttrium iron garnet) films grown on YAG substrates by multi-beam pulsed laser deposition (PLD).

Magneto-optic garnets, such as YIG ($Y_3Fe_5O_{12}$), find application in areas including optical communication, where they can be used as optical rotators and isolators [1] via their Faraday effect, and also in the microwave range, where their ferromagnetic properties can be exploited e.g. in magnetostatic wave band pass filters [2].

YIG films have been grown by liquid phase epitaxy [3], sputtering [4] or pulsed laser deposition (PLD) [2, 5]. Single-beam PLD from a stoichiometric target usually leads to films that are iron (Fe) deficient, as previously reported in the literature [2] and confirmed in our experiments [5].

We have studied the composition, analysed by energy-dispersive x-ray spectroscopy, and the effects of stoichiometric variation on the resultant properties of PLD-grown YIG films, whose yttrium (Y) and Fe concentration is varied by co-ablation of two separate targets of polycrystalline YIG with either a Y_2O_3 or a Fe_2O_3 target. In this report, we analyse in detail our results with the latter target, for which we have most extensive data.

All depositions were performed in our multi-beam PLD system, described in [6]. YIG films were grown on $10 \times 10 \text{ mm}^2$ 1 mm-thick YAG (100)-oriented substrates, heated to T=1400 K, the highest temperature possible in our system, lower than the optimum value ($T_{\text{OPT}}=1600 \text{ K}$) for YIG/YAG growth [5]. The off-axis configuration of target holders relative to the substrate causes a lower deposition rate, compared to that reported in [5], using our single-beam PLD system; for this reason, we reduced the target-substrate distance to d=4 cm and, at the same time, increased the oxygen pressure to $P_{\text{O2}}=3.4 \text{ Pa}$, following the general rule of thumb: $d \cdot P_{\text{O2}}=\text{const.}$, in order to maintain the same plume dynamics. Multi-PLD experiments of YIG + Fe₂O₃ were performed with two different laser configurations, as described below.

Samples Y45 and Y48-51, whose data are reported in the upper half of Table 1, were deposited by ablating the YIG target with a frequency-quadrupled Nd:YAG laser (wavelength: = 266 nm; fluence: $F=1.3 \text{ J/cm}^2$; pulse repetition rate: f=10 Hz) and the Fe₂O₃ target with a KrF laser (wavelength: = 248 nm; fluence: $F=1.3 \text{ J/cm}^2$), whose pulse repetition rate was varied between 0 and 4 Hz to increase the Fe concentration in the YIG films

Samples Y52-58, whose data are reported in lower half of Table 1, were deposited by ablating the YIG target with the KrF laser (fluence: $F=2.3 \text{ J/cm}^2$; pulse repetition rate: f=20 Hz, except for Y57, grown at f=16 Hz) and the Fe₂O₃ target with the Nd:YAG laser ($F=1.3 \text{ J/cm}^2$), whose pulses were gated through a programmable shutter, whose opening and closing times were changed to achieve equivalent repetition rates of 0-4 Hz. The target ablation ratio r is defined as the ratio of the number of laser pulses per second on the Fe₂O₃ target over the number of laser pulses per second on the YIG target.

As shown in Figure 1, FMR (ferromagnetic resonance) linewidth tends to increase with increasing target ablation ratio and consequently with increasing Fe concentration, whatever laser system is used, although excimer ablation of YIG tends to produce better quality films: in fact, in spite of the higher Fe concentration, YIG films grown by ablating YIG with the KrF laser (Y48-51) have broader FMR linewidth than Y52-58, grown by ablating YIG with the excimer laser. Moreover, Y48-51 have darker colour, lower optical transmission, higher surface roughness and particulate density than Y52-57. Comparing the un-doped YIG films, Y45 and Y58, the latter (grown with the KrF laser) have narrower FMR linewidth than the former (grown with the Nd:YAG laser), but they are both uniform and smooth, unlike Y48-51; this means that it is the ablation of Fe₂O₃ that is critical for the production of smooth YIG films. In fact, deposition of pure Fe₂O₃ testers on *c*-cut, (0001)-oriented sapphire (-Al₂O₃) substrates showed that ablation of Fe₂O₃ with the Nd:YAG laser produced crystalline hematite (-Fe₂O₃) films with lower particulate density and surface roughness than PLD of Fe₂O₃ with the KrF laser. Optical spectroscopy showed that transmission decreases with increasing Fe concentration and a red-shift of the absorption edge occurs in Y52-58.

Tab	ole 1. Tar	get ablat	ion ratio <i>r</i>	, thickness t, co	omposition,	FMR linewi	dth (<i>H</i>)	and colou	ir of YIG samples	

SAMPLE	r	<i>t</i> [μm]	Y concentration [formula number]	Fe concentration [formula number]	Y/Fe	<i>H</i> [mT]	Sample colour				
YIG samples grown by ablating the YIG target with the Nd:YAG laser and the Fe ₂ O ₃ with the KrF laser											
Y45	0	2	3.37	4.63	1.37	4.67	Yellow				
Y48	0.1	1.4	3.10	4.89	1.58	17.25	Orange				
Y49	0.2	1.8	3.02	4.98	1.65	21.12	Light red				
Y51	0.3	1.7	2.93	5.07	1.73	24.05	Dark red				
Y50	0.4	1.7	2.92	5.08	1.74	29.81	Dark red				
YIG samp	YIG samples grown by ablating the YIG target with KrF laser and the Fe ₂ O ₃ target with the Nd:YAG laser										
Y58	0	2.5	3.50	4.50	1.28	3.87	Yellow				
Y55	0.05	2.2	3.40	4.60	1.35	6.03	Yellow				
Y52	0.1	3	3.22	4.78	1.48	6.50	Dark yellow				
Y57	0.125	3	3.03	4.97	1.64	9.21	Brownish / dark yellow				
Y54	0.15	4	2.88	5.12	1.78	8.70	Light brown				
Y53	0.2	3.5	2.89	5.11	1.77	16	Brown				

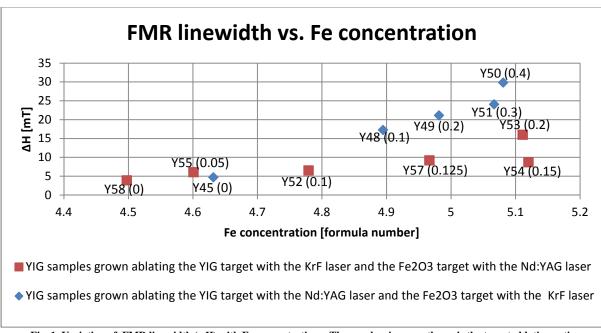


Fig. 1. Variation of FMR linewidth (H) with Fe concentration – The number in parentheses is the target ablation ratio.

In conclusion, Fe-deficient YIG films seem to have better optical and magnetic properties. We are also currently depositing and characterising YIG samples grown by multi-PLD of YIG and Y_2O_3 : we suspect that yttrium doping may produce a reduction in FMR linewidth and an increase in optical transmission.

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