

An investigation of PDMS structures for optimized ferroelectret performance

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Abstract: This paper reports the ANSYS simulation and fabrication processes for optimising PDMS ferroelectret performance. The proposed model extends the previously published analytical models and combines this with simulation of individual void geometry. The ferroelectret material is fabricated from PDMS using 3D-printed plastic mould. Mathematic model and Ansys simulation results predict that the performance of PDMS ferroelectret is varied with the geometry of voids and surface charge density, and the theoretical maximum piezoelectric coefficient d_{33} achieved was about 220 pC/N. The experimental maximum d_{33} obtained was 172 pC/N.

1. Introduction

A ferroelectret is a type of polymer electret that displays piezoelectric like properties, which is a polymer film with cellular structure. The thickness of ferroelectret films typically ranges from a few dozen to a hundred microns. The geometry of internal voids typically a lens shape with a thickness and width from a few to dozen micrometers. The separated positive and negative charges which are trapped on the upper and lower gas-polymer interfaces in the voids cause the piezoelectric like property of ferroelectrets, because the separated charged form macroscopic dipoles in the voids. The macroscopic dipole moment can be determined by the quantity of the charges and the distance between the separated charges. When the ferroelectret is compressed by an external mechanical load, its volume decreases and compensating charges are generated on the external surface. The resulting macroscopic behaviour of ferroelectret is very similar with the well-known piezoelectric materials; but its microscopic mechanism of charge generation is entirely different from the latter. For ferroelectret, the interior voids not only reduce the overall mass and stiffness, but also generate great dipoles when the separated positive and negative charges are trapped on the upper and lower surface of the voids. Due to the combined effects of the internal dipole moment and the anisotropic cellular polymer structure, the majority of volume change takes place in the voids. Therefore when loaded the distance between positive and negative charges decreases, the dipole moment decreases. The amount of compensating charge decreases and a flow of charge is generated, which presents a strong piezoelectric like effect for ferroelectret [1]. Hence, ferroelectrets are utilized as functional materials in electromechanical sensors and actuators, also as electrically charged layers in electrostatic transducers [2, 3].

The piezoelectric like properties of ferroelectrets has already been theoretically analysed [4-7]. However, most ferroelectrets are formed by a modified film blow and extrusion process for producing the desired cellular structures. The results of this fabrication method are usually ill-controlled in both individual void geometry and overall cellular structure, and the existing modelling methods to approximate the piezoelectric like properties of these materials and structures are simplified models that do not consider individual void geometry and void distribution. Therefore, in this paper we report

a design, fabrication and testing methodology for optimising PDMS ferroelctret performance based on polymer MEMS fabrication technology.

2. Model

Piezoelectricity of ferroelectrets has been theoretically analyzed with a simplified model based on charged parallel polymeric and gaseous layers, which includes the finite-element calculations [4-7]. Based on the reference state of the model, the ANSYS simulation work of this research started from cellular PDMS structures (Figure 1) with a layer of rectangular voids. The ANSYS results predicted deformation of the voids when the structure with an external mechanical load. Assuming the charge densities of the voids surface was uniformed, the variation of the charge on the external surface can be calculated by the average of the deformations then the piezoelectric coefficient d_{33} can be estimated as:

$$d_{33} = 2\varepsilon_1\sigma(x_2\Delta x_1 + x_1\Delta x_2)(2x_1 + \varepsilon_1x_2)^{-2}F^{-1} \quad (1)$$

Where ε_1 is the dielectric constant of air and the relative dielectric constant of PDMS; σ is the charge density on void surface; x_1 and x_2 are the thickness of solid and void layers, respectively; Δx_1 and Δx_2 are the variations of the thickness; F is the external force;

For mathematical model, the piezoelectric coefficient d_{33} can be directly obtained as:

$$d_{33} = 2\varepsilon_1\sigma tr_1(1 - sr_1)^2 sr_1^{-1}[2 + \varepsilon_1 tr_1]^{-2} c_{33}^{-1} \quad (2)$$

Where sr_1 is the ratio of the effective area of the void layers to the effective area of the solid layers; tr_1 is the ratio of the thickness of void layers to the thickness of solid layers; σ is the charge density on void surface; ε_1 is the dielectric constant of air and the relative dielectric constant of PDMS; c_{33} is the elastic modulus of PDMS.

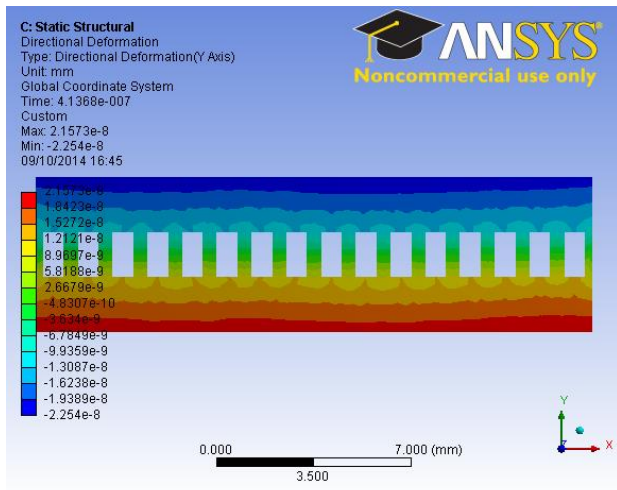


Figure 1. Cellular structures with a layer of rectangular voids

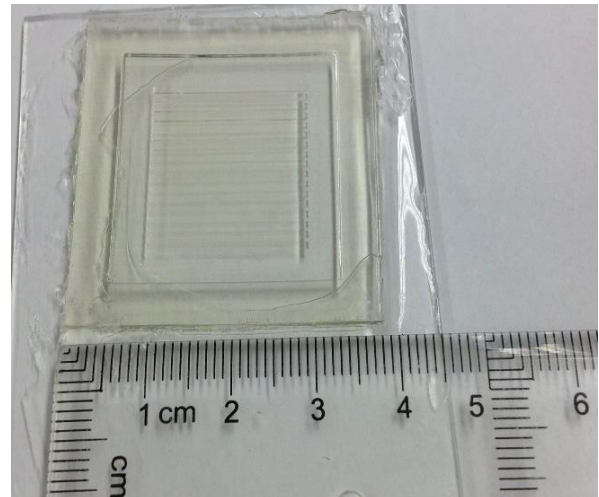


Figure 2. The image of a prepared 3D-printed mould

3. Fabrication and Measurement

3.1. 3D-printed mould preparation

A 3D CAD model of the mould structure was constructed with Solidworks 2012 (Dassault Systems, MA, USA), exported as an STL-format file, and printed on a Connex350TM 3D printing system (Stratasys, MN, USA). The used moulds can be made from VeroWhitePlusTM, VeroGeryTM or VeroClearTM, generated in 'glossy' printer mode on a bed of FullCure@705 support material. Because the 3DP material inhibits PDMS polymerization and the printed structure tends to warp some extent when it is removed from the printer, the resulting 3D-printed structure is not directly suitable as a mould for PDMS casting. The moulds should be baked in oven at 80 °C overnight. In this bake step, the moulds were placed and glued to a glass slide by a drop of degassed non-polymerized PDMS, with its base in contact with the PDMS film. After this procedure, the glass-backed moulds were exposed to a silane vapour for 1 hour to coat it with a thin layer of trichloro(1H,1H,2H,2H-

perfluorooctyl)silane (Sigma Aldrich, MO, USA). An example of a finished mould of VeroClear™ material, with outer dimensions of 38×38 mm, is shown in Figure 2. Depending on the maximum precision of the 3D printer and the restrictions of fabrication process, the thickness of the solid layer and void layer was selected as 2 mm and 1.6 mm, respectively; the width of the voids and the gap between the voids was selected as 0.75 mm and 0.5 mm, respectively.

3.2. PDMS casting and Bonding

Liquid PDMS and curing agent (Sylgard 184 from Dow Corning, MI, USA) were mixed at a 10:1 weight ratio and the mixture was degassed in a vacuum desiccator. The degassed PDMS was poured into the glass-backed moulds which were again degassed and then baked at 80°C for 1 hour. After detaching the polymerized PDMS from the moulds, an oxygen plasma treatment was applied on the patterned surface of PDMS layers (Femto Asher, Diener, Germany, 30 S at 35-40W) and then bonded to the other PDMS layers. To optimise the bonding effect, the samples were assembled with bonding times of 30 s, 60 s, 90 s and 120 s, separately. The bonded PDMS was then baked in oven at 80°C for 1 hour to improve the bonding effect, and also recover the original hydrophobic PDMS surface chemistry. The schematic of PDMS casting and bonding processes, and the cross-section view of the bonded sample are illustrated in Figure 3 and Figure 4, separately.

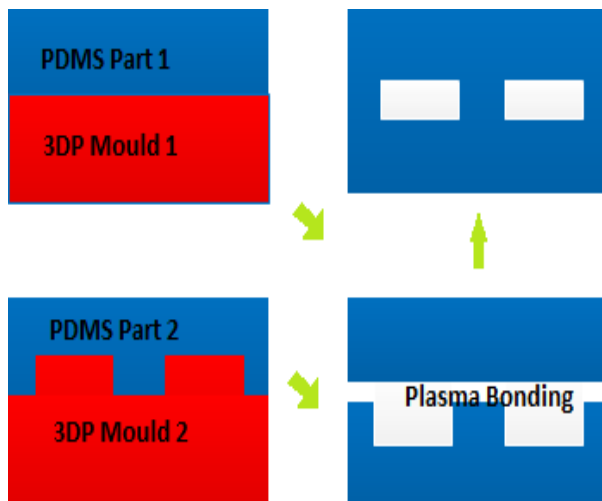


Figure 3. Schematic of fabrication processes

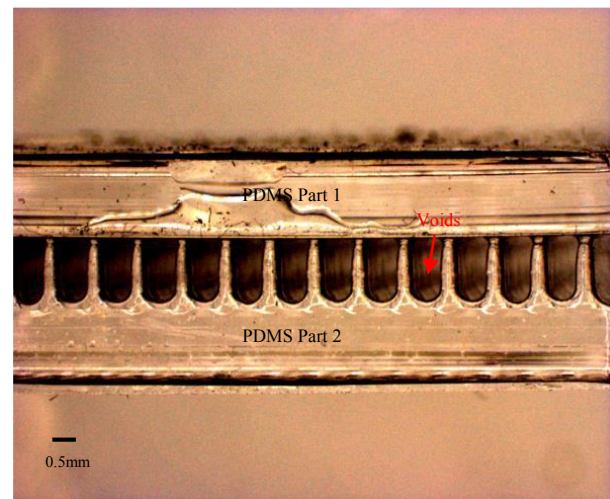


Figure 4. Image of cross-section view of single layer ferrelectret foam

3.3. Corona charging and Measurement

The samples were polarized with a point-to-plane corona discharge in air under ambient conditions. A corona-tip voltage of -25 kV and a charging time of 30 s were employed. Its piezoelectric charge constant d_{33} was measured using a PiezoMeter (PM300, Piezotest Ltd).

4. Results and Discussion

A mathematical result is provided in Figure 5, which numerically illustrates the impact of the different void geometry. While the ratio of the effective areas of the solid and void layers is increased, the piezoelectricity of PDMS ferrelectret was reduced. In other word, the piezoelectric properties of the PDMS ferrelectret improved with increasing void area. In contrast, when the ratio of the thickness of the solid and void layers is increased, the piezoelectric properties of the PDMS ferrelectret was improved and the optimised point has been demonstrated at around 0.8, with a peak piezoelectricity of 220 pC/N. However, considering the feasibility of the fabrication processes, the optimized structure which can be achieved with $sr_1=0.4$ and $tr_1=0.8$. Based on these ratios, the geometry of void structure is designed as 0.75 mm in width and 1.6 mm in height, and the distance between the voids and the thickness of the whole film is selected as 0.5 mm and 5.6 mm, respectively. For this structure, the theoretical d_{33} values is 62.5 pC/N at surface charge density = 100 $\mu\text{C}/\text{m}^2$.

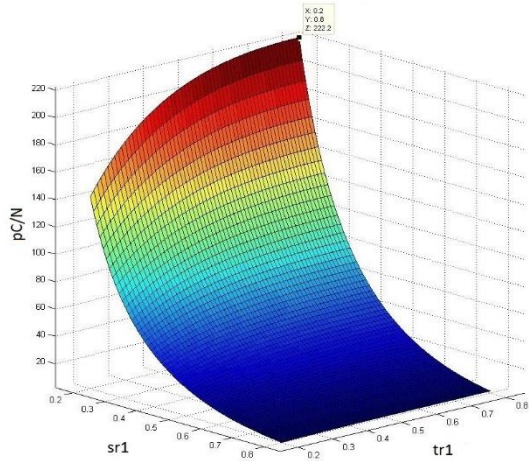


Figure 5. Mathematical rectangle model results varying with the size of voids

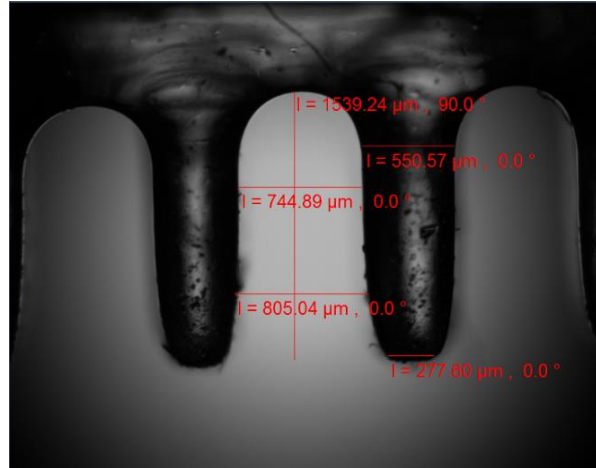


Figure 6. The image of cross section view of Realized voids

The realized geometry of the void inside the PDMS ferroelectret is illustrated in Figure 6. The realized geometry of void is different to the model due to the precision of fabrication process. The d_{33} value of the PDMS ferroelectret was determined to be around 70 pC/N. Figure 7 shows a comparison of ANSYS simulation results and Mathematical model results to illustrate the influence of the surface charge density based on the designed geometry. There was an obvious deviation between the ANSYS simulation results and mathematical model results because the mathematical model was based on the principle of a rigid body. In other word, the bending effects that happened on the upper and lower gas-polymer interfaces of the voids did not considered.

The performance of the PDMS ferroelectret can be influenced by plasma treatment. The impact of plasma treatment time is illustrated at Figure 8. The performance of the PDMS ferroelectret was substantially grown during the plasma treatment duration increasing. The experimental maximum d_{33} obtained is about 170 pC/N at 120 s plasma treatment condition. The possible reason of this phenomenon is the surface charge density of PDMS can be significantly improved by increase in plasma treatment time.

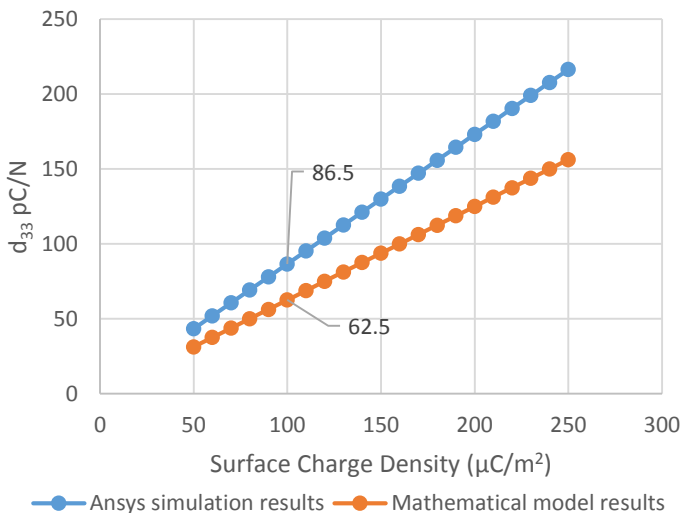


Figure 7. Mathematical model results and Ansys simulation d_{33} results for rectangle voids

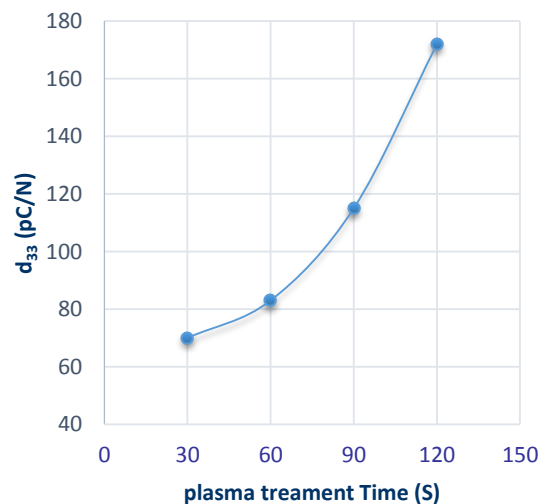


Figure 8. The piezoelectric performance varying with duration of plasma treatment

5. Conclusion and future work

In this work, the simulation and testing of PDMS ferroelectret with a variety of void geometries and surface charge density has been investigated. The piezoelectric properties of PDMS ferroelectret are improved with an enhancement of the void area and the optimized point can be achieved at the ratio $tr_1=0.8$ for a single void layer of PDMS ferroelectret material. In addition, the performance of PDMS ferroelectret can be significantly affected by the plasma treatment. In the future, PDMS ferroelectret will be optimised for different shapes, multilayer void structure and scaling down based on a MEMS fabricated mould.

6. Acknowledgement

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7. References

- [1] M. Paajanen, "The cellular polypropylene electret material-Electromechanical properties", Dissertation, VTT Publications 436, Technical Research Centre of Finland, Tampere University of Technology, 2001.
- [2] J. Backman. "Audio applications of electrothermomechanical film (ETMF)", *J. Aud. Eng. Soc.*, vol. 38 (5), pp. **364-371**, 1990.
- [3] M. Paajanen, J. Lekkala, and K. Kirjavainen, "Electromechanical film (EMFi) - a new multipurpose electret material", *Sens. Actu. A: Phys.*, vol. 84 (1-2), pp. **95-102**, 2000.
- [4] J. Hillenbrand and G. M. Sessler, "Piezoelectricity in cellular electret films", *IEEE Trans. Dielect. Electr. Insul.*, vol. 7 (4), pp. **537-542**, 2000.
- [5] M. Paajanen, J. Lekkala, and H. Välimäki, "Electromechanical modeling and properties of the electret film EMFi", *IEEE Trans. Dielect. Electr. Insul.*, vol. 8 (4), pp. **629-636**, 2001.
- [6] M. Paajanen, H. Välimäki, and J. Lekkala, "Modelling the sensor and actuator operations of the ElectroMechanical Film EMFi", Proceedings, 10th International Symposium on Electrets, 22-24 September 1999, Delphi, Greece (IEEE Service Centre, Piscataway, NJ, USA 1999), pp. **735-738**.
- [7] G. M. Sessler and J. Hillenbrand, "Electromechanical response equations of cellular electret films", Proceedings, 10th International Symposium on Electrets, 22-24 September 1999, Delphi, Greece (IEEE Service Centre, Piscataway, NJ, USA 1999), pp. **261-264**.