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3 **Effect of RVC porosity on the performance of PbO₂ composite**
4 **coatings with titanate nanotubes for the electrochemical oxidation**
5 **of azo dyes**

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1 **Abstract**

2 Reticulated vitreous carbon (RVC) of different porosities (20, 45, 60, 80, and 100 ppi –
3 pores per inch) has been used as a large surface area substrate for preparing 3D-like PbO₂
4 coatings (RVC/PbO₂) as well as composite coatings with hydrothermally synthesized titanate
5 nanotubes (RVC/PbO₂/TiNT) by galvanostatic electrodeposition from baths containing
6 lead(II) methanesulfonate and methanesulfonic acid. The effect of the RVC porosity on the
7 coating thickness, the electrocatalytic behaviour and the ability to remove the colour and total
8 organic carbon (TOC) from solutions containing the azo dye Methyl Orange has been
9 systematically assessed. As shown from scanning electron micrographs, the greatest thickness
10 (up to 120 µm) was obtained using > 60 ppi, but the β-PbO₂ nanocrystallites mainly grew on
11 the external surface, leaving mostly uncoated inner RVC stripes and ending in planar-like
12 PbO₂-based electrodes. In contrast, thinner but perfectly adherent and homogeneous coating
13 of the inner and outer surface was achieved with 20-60 ppi, showing electrodes with an
14 optimal three-dimensionality. This was especially confirmed by cyclic voltammograms for
15 the composite coatings, as deduced from their highest electroactivity that can be related to
16 enhanced adsorption onto the TiNT clusters and the larger ability to produce active
17 PbO₂(•OH). The comparative electro-oxidation of 0.25 × 10⁻³ mol dm⁻³ Methyl Orange acidic
18 solutions in 0.05 mol dm⁻³ Na₂SO₄ at 0.6 A demonstrated that RVC (45 ppi)/PbO₂/TiNT was
19 the optimum material. It allowed the quickest decolourisation, reaching 60% in 2.5 min and >
20 98% at 45 min, and > 55% TOC abatement at 240 min. The anode presented a perfect surface
21 coverage, with no evidence of RVC degradation. The effect of dye concentration and
22 supporting electrolyte nature was studied, revealing a very positive effect of NaCl.

23 *Keywords:* Electrochemical advanced oxidation processes; Lead dioxide anode;
24 Methanesulfonic acid; Reticulated vitreous carbon; Water decontamination

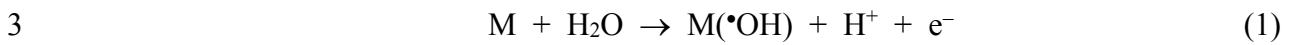
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1 **1. Introduction**

2 In the last decade, water demand has increased due to its intensive use in different
3 activities and, as a result, projection scenarios suggest that there will be an increasing number
4 of water-deficit regions worldwide [1]. Therefore, great effort is needed for developing
5 powerful water treatment technologies to allow reusing industrial water once the complete
6 removal of hazardous pollutants like heavy metals, dyes, pesticides and pharmaceuticals has
7 been ensured [2]. There exists a large variety of physicochemical and biological processes for
8 water decontamination, including electrochemical technologies that have been successfully
9 tested for treating pesticides [3], pharmaceuticals [4] and synthetic dyes [5]. Among those
10 organic chemicals, azo dyes are especially relevant because they are widely used in textile,
11 food and paper and leather industries and produce odor, color and serious environmental and
12 health issues [6]. These compounds contain one or more azo bonds ($-N=N-$) as chromophore
13 group linked to aromatic structures with functional groups such as $-OH$ and $-SO_3H$, among
14 others [5]. Many methods have been used to remove azo dyes from water, as for example
15 adsorption [7], coagulation [8] and advanced oxidation processes (AOPs) like Fenton's
16 reagent [9]. However, these processes are expensive, inefficient, present operational problems
17 like complex setups or handling and, frequently, yield either secondary pollution or hazardous
18 by-products that tend to accumulate in final solutions. Consequently, there has been
19 increasing interest in the development of simpler and more efficient separation and
20 degradation electrochemical technologies to address this problem.

21 Lately, it has been demonstrated that the electrochemical AOPs (EAOPs), whose
22 oxidative ability is governed by hydroxyl radicals ($\bullet OH$), possess peerless ability for
23 destroying azo dyes contained in water matrices [10]. In EAOPs, the destruction of the
24 organic molecules is mediated by $\bullet OH$ mainly formed from water electro-oxidation (EO) at

1 the anode (M) via reaction (1) [11-20] or by means of the electrogenerated Fenton's reagent in
2 the bulk [17,19-23].



4 Among them, EO is the most widespread method because of its simplicity, adaptability
5 and scale-up as compact modularized systems. At high applied current, the degradation of the
6 organic molecules by adsorbed M(\bullet OH) prevails over their direct oxidation at M [12,24,25].
7 Dimensionally-stable anodes and Pt favour the conversion of the initial pollutant to stable
8 intermediates, whereas high oxidation power anodes like boron-doped diamond (BDD) and
9 PbO₂ promote the total mineralization of solutions because the M(\bullet OH) are physisorbed [26].
10 Despite the lower degradation kinetics achieved with PbO₂ compared to BDD [27], it has
11 been employed by many authors because it is less expensive, presents a high stability in a
12 wide range of electrolytes and can be readily prepared by electrodeposition [28]. EO with
13 PbO₂ anodes has been proven an effective technology for the degradation of azo dyes like
14 Methyl Red [11], Methyl Orange [29], Reactive Blue 194 [30] and Reactive Red 195 [31].

15 Traditionally, the PbO₂ coatings have been synthesized by anodic deposition onto
16 planar substrates, being Ti the preferred material [29-33]. Regarding the electrolyte for
17 performing the electrodeposition, oxidising and toxic acidic media such as HClO₄, HNO₃ and
18 H₂SO₄ are usually employed but, in recent years, methanesulfonic acid (MSA) has arisen as a
19 more environmentally friendly medium. MSA yields a higher deposition rate of β -PbO₂
20 [34,35], which is the most suitable allotrope for organics oxidation owing to its open structure
21 that provides a larger active surface area [36,37]. Despite the good results obtained in EO
22 with planar anodes, the potential benefits of using materials with a larger surface-to-volume
23 ratio are undeniable since a clear mass transport enhancement is expected, which therefore
24 counteracts the limitations of the low space-time yield and normalized space velocity [38]. As
25 a result, the performance of the electrochemical reactor in terms of degradation kinetics is

1 significantly improved [39,40]. For example, BDD meshes with a continuous honeycomb
2 structure have been used for the electro-oxidation of Indigo dye [41], whereas perforated
3 metal plates, metal foams or DSA meshes are also available in the market. Conversely, the
4 use of cheaper, raw carbonaceous materials as the anode in EO becomes severely restricted
5 due to their low ability to generate $\bullet\text{OH}$ and, more importantly, their poor mechanical
6 stability, which limits their applicability to 3D electrochemical reactors equipped with packed
7 beds of discrete bipolar particles [42]. Worth highlighting, 3D-like anodes composed of thin
8 active layers of metal or metal oxide coatings onto continuous porous carbonaceous matrices
9 seems rather advantageous, since the amount of catalytic material to produce $\bullet\text{OH}$ can be kept
10 to a minimum by controlling the electrodeposition parameters. Liu et al. [43] prepared Bi-
11 doped SnO_2 -coated carbon nanotubes for EO of organics, whereas some of us recently
12 reported the manufacture of PbO_2 anodes on reticulated vitreous carbon (RVC) with a fixed
13 porosity [44]. As far as we know, that paper and a recent one by Chai et al. [45] are the sole
14 studies on electrodeposited 3D-like PbO_2 anodes for EO of organics.

15 As a step further from our previous paper, this work focuses on the galvanostatic
16 preparation of PbO_2 coatings on RVC (RVC/PbO_2) and PbO_2 composites with titanium
17 nanotubes ($\text{RVC}/\text{PbO}_2/\text{TiNT}$) using RVC of different porosities (20, 45, 60, 80, and 100 ppi –
18 pores per inch) in MSA medium, and eventually test them for the EO of Methyl Orange
19 solutions. The RVC porosity is a fundamental parameter to optimize the oxidation of water
20 pollutants but, until now, the investigation of the effect of porosity has been limited to raw
21 RVC cathodes employed for the electrochemical removal of metal ions like Cr(VI) , Cu(II)
22 and Zn(II) [46-48] and for the deposition of Cu and Ni for H_2 evolution [49]. Regarding the
23 use of TiNT, a first approach was made in our previous work [44], reaching a certain
24 enhancement in the destruction of organic molecules. The surface analysis of the prepared
25 coatings has been done by means of scanning electron microscopy (SEM) to elucidate their

1 morphological characteristics, whereas the electrochemical characterisation was carried out
2 by cyclic voltammetry. Their potential application in EO has been assessed from colour and
3 total organic carbon (TOC) removal analyses.

4 **2. Experimental**

5 *2.1. Chemicals*

6 Reagent grade Pb(II) methanesulfonate ($\text{Pb}(\text{CH}_3\text{SO}_3)_2$, Aldrich, 50%) and MSA
7 ($\text{CH}_3\text{SO}_3\text{H}$, Aldrich, 70%), as well as Methyl Orange azo dye (4-[4-
8 (dimethylamino)phenylazo]benzenesulfonic acid sodium salt, Fisher Scientific, > 99%) were
9 used without further purification. The TiNT were prepared via hydrothermal treatment of
10 TiO_2 (anatase, Aeroxide® TiO_2 P25 from Evonik Degussa) in 10 M NaOH at 140 °C for 24
11 h, according to a modified procedure from the pioneering synthesis of nanotubular titanate
12 [50,51]. They presented a specific surface area (BET, N_2 desorption) and pore volume of ca.
13 $200 \text{ m}^2 \text{ g}^{-1}$ and $0.7 \text{ cm}^3 \text{ g}^{-1}$, respectively. Analytical grade anhydrous sodium sulfate and
14 sodium chloride were used as background electrolytes, whereas the initial pH of the solutions
15 was adjusted with sodium hydroxyde and H_2SO_4 . All reactants were purchased from Fisher
16 Scientific and Fluka and all solutions were prepared with ultra-pure water with resistivity > 18
17 $\text{M}\Omega \text{ cm}$ (at 22.5 °C), obtained from an Elga water purification system.

18 *2.2. Electrodeposition of 3D-like raw PbO_2 coatings and composite coatings*

19 The electrodeposition conditions for preparing the RVC/ PbO_2 coatings were selected on
20 the basis of our previous study aimed at optimising the obtention of perfect β - PbO_2 deposits
21 [44]. For this, the deposition baths contained 80 cm^3 of 1.0 mol dm^{-3} $\text{Pb}(\text{CH}_3\text{SO}_3)_2$ and 0.2
22 mol m^{-3} MSA. The deposition was carried out at a constant current of 2.5 A for 30 min in a
23 cylindrical, undivided glass cell thermostated at 60 °C employing a Grant LT D6G water

1 bath. The cell voltage was constant during the deposition experiments, i.e., 4 V. A similar
2 methodology was followed for the preparation of the RVC/PbO₂/TiNT composite coatings,
3 but it involved the co-deposition in the presence of 2.0 g dm⁻³ of the freshly synthesized
4 TiNT.

5 RVC samples (ERG Materials) of different porosities (20, 45, 60, 80, and 100 ppi –
6 pores per inch) were used as the anode. Their volume was 8.8 cm³, exhibiting a parallelepiped
7 form (i.e., 4.7 cm × 1.7 cm × 1.1 cm) with a surface area per unit volume that increased with
8 porosity from 13 to 65 cm⁻¹ [10]. A platinum wire was inserted into each RVC and glued with
9 silver epoxy adhesive to ensure an optimum electrical contact. A cylindrical stainless steel
10 mesh was used as the cathode. In all cases, the anode was placed in the centre of the
11 electrolytic cell, being completely surrounded by the cathode. Such a configuration favoured a
12 proper current and potential distribution in the RVC substrates, ending in more homogeneous
13 coatings. A fresh solution and anode were used for each experiment. The solution was
14 magnetically stirred at 800 rpm with a PTFE-coated magnetic follower to ensure reproducible
15 mass transport conditions as well as to allow the dispersion of the TiNT.

16 *2.3. Electrochemical oxidation of Methyl Orange azo dye solutions*

17 Raw RVC samples as well as the electrodeposited materials described in the previous
18 section were rinsed with ultra-pure water and dried. Then, they constituted the anodes for
19 treating azo dye solutions using an open, undivided glass cell with the same configuration
20 described for the electrodeposition trials. Solutions of 275 cm³ of 0.25-0.50 × 10⁻³ mol dm⁻³
21 Methyl Orange in either 0.05 mol dm⁻³ Na₂SO₄ or NaCl, as well as in mixed media, were
22 electrolyzed at 0.6 A, pH 3.0 and 22.5 °C. The solution pH was very stable throughout the
23 electrolyses. Vigorous stirring was provided by a magnetic follower.

1 2.4. Apparatus and analysis procedures

2 The deposition at constant current was performed with a computer-controlled
3 PGSTAT302N potentiostat/galvanostat from Autolab (EcoChemie, Netherlands) using the
4 Nova 1.7 software. The morphology and the thickness of all coatings were examined using a
5 thermal field emission scanning electron microscope model JSM 6500F with an accelerating
6 voltage of 15 kV. For the electrochemical characterisation, steady state cyclic voltammetry
7 was carried out with the same Autolab PGSTAT302N instrument, using a purpose-built,
8 three-electrode, undivided glass cell containing 50 cm³ of solution at pH 3.0 and 22.5 °C. A
9 large area Pt gauze and Ag|AgCl|KCl (sat) were used as the counter and reference electrodes,
10 respectively. The exposed area of the RVC, RVC/PbO₂ and RVC/PbO₂/TiNT samples tested
11 as working electrodes ranged between 113.1 and 565.5 cm² (corresponding values for 20 and
12 100 ppi, respectively). All potentials in this paper are referred to the silver/silver chloride
13 reference electrode potential.

14 The azo dye solutions were electrolysed using the same Autolab PGSTAT302N
15 instrument, and various samples were withdrawn at regular intervals for further analysis. The
16 decolourisation of the dye solutions was evaluated from their absorbance (A) decay at the
17 maximum visible wavelength (λ_{\max}) of 503 nm, measured at 35 °C on a Perkin Elmer Lambda
18 UV/Vis spectrophotometer controlled by UV-WinLab Explorer software. The percentage of
19 colour removal, also called decolorisation efficiency, could then be calculated as follows [17]:

$$20 \quad \text{Color removal (\%)} = \frac{A_0 - A_t}{A_0} \times 100 \quad (2)$$

21 where A_0 and A_t are the initial absorbance that corresponding at time t of the treatment.

22 The mineralisation of the Methyl Orange solutions was assessed from the decay of their
23 dissolved organic carbon, which corresponds to the TOC value because no heterogeneous
24 organic matter was present in the solutions. This analytical parameter was determined on a

1 Shimadzu VCSN analyzer. Samples were first microfiltered onto a hydrophilic membrane
2 (Millex-GV Millipore, pore size 0.22 μm), and then reproducible TOC values with $\pm 1\%$
3 accuracy were found using the non-purgeable organic carbon method. The cell voltage was
4 continuously monitored on the Autolab to ensure that the coatings and electrical contact did
5 not become damage during the electrolyses.

6 **3. Results and Discussion**

7 *3.1. Surface characterisation of the PbO_2 -based coatings*

8 A series of three-dimensional raw PbO_2 coatings (RVC/PbO_2) as well as
9 $\text{RVC}/\text{PbO}_2/\text{TiNT}$ composite coatings was prepared by galvanostatic electrodeposition at 2.5
10 A and 60 $^\circ\text{C}$ for 30 min. RVC with porosity ranging between 20 and 100 ppi was used as the
11 substrate, and SEM was employed for investigating the influence of porosity via a detailed
12 morphological analysis. Some representative SEM images are shown in [Fig. 1a-d](#). Note that
13 the raw RVC molds exhibited a rigid honeycomb structure composed of carbonaceous strands
14 assembled as so-called trigonal struts. The specific features of RVC samples with different
15 pore sizes can be observed elsewhere [\[38\]](#). Overall, the physical appearance of freshly-made
16 materials was outstanding, with no traces of uncoated RVC along the outer surface.
17 Furthermore, the coatings were highly uniform and adherent, since they could be handled
18 without apparently forming any debris or leading to a flake-off of the PbO_2 and PbO_2/TiNT
19 layers. Good uniformity and adherence mainly arise from the low resistivity of the RVC
20 substrates [\[52\]](#), as well as from the adequate selection of the deposition conditions to
21 manufacture high quality PbO_2 -based networks, being produced at a lower cost than those
22 prepared onto Ti or Nb grids.

23 The manufactured 3D-like electrodes exhibited a very well-attached metal oxide to the
24 RVC strips. Unlike the scarcely adherent coatings, no cracked-muddy surfaces with chunks

1 were obtained under the selected deposition conditions. Their reticulated structure is clearly
2 evidenced in Fig. 1a-b. For example, the features of the RVC (20 ppi)/PbO₂ electrode are
3 shown in Fig. 1a, confirming the formation of a regular, continuous PbO₂ coating covering
4 the RVC skeleton. The three-dimensionality is a key characteristic for minimising the mass
5 transport limitations that often appear when using planar high oxidation power anodes like
6 PbO₂ plates or BDD thin films. Furthermore, being open-pore materials, the present
7 electrodes have a lower density than metallic grids, which is essential for obtaining
8 lightweight PbO₂ anodes compared to traditional massive deposits. In Fig. 1b, a general view
9 of the whole structure of the RVC (60 ppi)/PbO₂ electrode is depicted, showing again the
10 homogeneous coverage of the underlying RVC substrate. A higher magnification of the
11 previous image can be seen in Fig. 1c, which allows identifying the morphology of the grown
12 PbO₂ crystallites in this kind of electrodes. The coatings were composed of a compact but
13 ragged structure containing large irregular or pyramidal-like features 10–15 μm in height with
14 obvious, sharp boundaries. This agrees with the recently reported microstructure of PbO₂
15 deposited onto a planar carbon substrate under similar electrolytic conditions [36,37]. It can
16 also be observed that the deposit was nanostructured, being composed of crystallites always in
17 the range of a few nanometers (10-30 nm according to previous works on planar or 3D
18 substrates [36,44]). A perfect surface coverage by PbO₂ nanoelectrodeposits in an MSA bath
19 was previously ascertained by SEM analysis for planar substrates [35-37]. Worth noting, the
20 phase composition of the RVC/PbO₂ electrodes was not studied by XRD, but previous
21 investigations in MSA reported the formation of pure β-phase when carrying out the
22 deposition conditions at high applied current [34-37]. In fact, the morphology shown in Fig.
23 1c agrees very well with that expected for β-PbO₂ coatings [36].

24 The second series of electrodes was prepared following a co-deposition procedure in the
25 presence of previously synthesized TiNT via the hydrothermal route, under the same

1 electrolytic conditions described for the RVC/PbO₂ samples. As an example of the resulting
2 composite coatings, Fig. 1d shows the SEM image of an RVC (60 ppi)/PbO₂/TiNT electrode.
3 As can be seen, high quality nanocomposites were obtained via the electrochemical synthesis
4 route, with titanate nanotubes being incorporated in the PbO₂ matrix. More precisely, the
5 elongated TiNT appear well attached to the PbO₂ nanocrystallites, thus giving rise to a
6 consistent heterogeneous surface with multiple potential properties derived from the presence
7 of both kinds of materials. All the RVC substrates presented such a mixed coverage with
8 TiNT bundles distributed along the surface, onto and among the crystallites. Since the TiNT
9 were tightly attached, they were not readily detachable. Consequently, the stability of the
10 adsorbed titanate clusters allowed their subsequent utilization, as commented below.

11 Worth mentioning, only a few studies have reported the electrochemical co-deposition
12 of either PbO₂ and TiNT [44] or PbO₂ and TiO₂ particles [53] so far. Since TiNT are
13 characterised by their high specific surface area, i.e., 200-300 m² g⁻¹, their presence may play
14 a decisive role, as recently tested in the case of the EO of organic pollutants and the
15 preparation of polypyrrol coatings with enhanced mechanical properties [44,45]. Regarding
16 the mechanism to explain the incorporation of the inert TiNT during the PbO₂
17 electrocrystallisation process, a two-step adsorption was previously suggested [44]. This leads
18 to the concomitant, strong immobilisation of TiNT that ensures that the final
19 RVC/PbO₂/TiNT composite structure is long-lasting.

20 From the above findings, it is clear that, apparently, 3D-like PbO₂ electrodes and
21 nanocomposites with a range of porosities and a good quality in terms of coating adhesion
22 and surface coverage can be obtained following the proposed methodologies. Their dual use
23 for pollutant adsorption and oxidation by •OH is then expected to promote the
24 decontamination of water, as will be discussed below. However, first of all it is crucial to
25 evaluate both, the thickness of the deposits and the quality of the coatings inside the pores.

1 Fig. 2a-b show cross-sectional SEM images of RVC (60 ppi)/PbO₂ and RVC (100 ppi)/PbO₂,
2 respectively. As can be seen, deposits of up to 60-70 μm thickness were formed on RVC of 60
3 ppi, whereas even thicker coatings were obtained on the external surface of RVC of 100 ppi.
4 Therefore, the use of MSA truly yields thicker coatings compared to traditional acidic media
5 [35]. Such thickness values are highly desirable for the use in water treatment because they
6 might ensure a greater protection of the underlying RVC structure. Interestingly, the deposits
7 proved to have a good adherence despite their considerable thickness, as mentioned above.
8 Measurements were made for all the RVC/PbO₂ coatings prepared, and Fig. 2c shows the
9 dependence of thickness of the PbO₂ deposits (grown outward) with RVC porosity. An
10 exponential increase of the thickness was found at a greater porosity, although the difference
11 was less remarkable when comparing the electrodes of 20, 45 and 60 ppi and much more
12 evident when increasing the pore density over 60 ppi. Regarding the RVC/PbO₂/TiNT
13 composites, a similar effect of porosity on the coating thickness was found (not shown).
14 Slightly thicker coatings were achieved with those materials, which can be related to the
15 positive influence of the nanosized titanate structures on the deposition kinetics of PbO₂.
16 Since they provide extra nucleation sites from produced Ti-O radical species, the rate of PbO₂
17 electrodeposition and the final amount of grown PbO₂ are enhanced [54].

18 In contrast to the apparent benefits derived from the increased RVC porosity, Fig. 2a-b
19 also demonstrate that it can be detrimental since RVC can remain uncoated in the inner space
20 of electrodes of porosity > 60 ppi, as revealed from cross-sectional analysis. The RVC (60
21 ppi)/PbO₂ was perfectly coated in the whole volume, whereas an inhomogeneous coating
22 appeared in the RVC (100 ppi)/PbO₂ sample. This incomplete deposition onto the whole RVC
23 will reduce the active area of the PbO₂ anode to be used in EO, as explained later on.
24 Certainly, the faradaic efficiency was typically 90% in all cases, as determined by calculation
25 of the deposition charge compared to gravimetric analysis, but the coverage of the RVC

1 stripes was much more homogeneous for smaller pore sizes (≤ 60 ppi). In other words, the
2 electrodeposits generated using RVC of 20-60 ppi presented a perfect distribution on the
3 surface and into the pores, whereas higher pore densities (> 60 ppi) led to clogged pores that
4 prevented the growth of the PbO_2 inward. As a result, a greater surface thickness was
5 obtained at 80 and 100 ppi (Fig. 2c). This finding is very relevant as for the use of these
6 electrodes for degrading organic pollutants because, actually, the RVC/ PbO_2 and
7 RVC/ PbO_2 /TiNT electrodes that reached up to 120 μm thicknesses can be essentially
8 considered as planar ones, whereas only the electrodes prepared with RVC of 20-60 ppi can
9 be defined as real homogeneous 3D-like PbO_2 -based materials.

10 3.2. Electrochemical characterisation of the PbO_2 -based coatings

11 All the prepared PbO_2 -based electrodes but more particularly those considered to
12 possess 3D-like structures were expected to be very suitable for the EO of pollutants because
13 of their high specific surface area, which favours the contact between adsorbed organic
14 molecules and the $\text{M}(\bullet\text{OH})$ formed via reaction (1). To investigate this aspect, the
15 electrochemical characterisation of raw RVC as well as RVC/ PbO_2 and RVC/ PbO_2 /TiNT
16 electrodes was carried out by means of cyclic voltammetry. Fig. 3 shows the effect of RVC
17 pores per inch on the maximum current (I_{max}) measured from the 5th voltammogram recorded
18 from + 1.4 V to + 2.0 V vs. $\text{Ag}|\text{AgCl}|\text{KCl}$ (sat), in a 0.25×10^{-3} mol dm^{-3} Methyl Orange
19 solution with 0.05 mol dm^{-3} Na_2SO_4 , at pH 3.0 and 22.5 $^\circ\text{C}$. A steep current increase with no
20 oxidation peaks appeared during the forward potential scan, which can be explained by the
21 contribution of several anodic reactions, namely the oxidation of water to O_2 and that of the
22 azo dye to yield oxygenated by-products, as well as their immediate transformation, as was
23 recently verified for its EO at PbO_2 surfaces [37]. Note that, according to reaction (1), water
24 oxidation reaction using high oxidation power anodes involves a first step yielding the

1 oxidant $M(\bullet\text{OH})$ [12], which can in turn be responsible for the quicker degradation of the
2 initial pollutant and its intermediates.

3 According to Fig. 3, the electroactivity of the different electrodes within the range 20-80
4 ppi was as follows: $\text{RVC/PbO}_2/\text{TiNT} \gg \text{RVC/PbO}_2 > \text{RVC}$. For all the samples, the value of
5 I_{max} increased when replacing 20 by 45 ppi, thus reaching a maximum and then, progressively
6 lower current values were obtained when using higher porosity substrates. This confirms that
7 coating the RVC stripes with PbO_2 and PbO_2/TiNT is not only necessary for minimising the
8 corrosion, but it also yields more electroactive materials, which at least partly contributes to
9 the quicker oxidation of the organic molecules, both directly and via $M(\bullet\text{OH})$. For example,
10 using a substrate of 45 ppi, current rised from 0.077 to 0.092 A upon PbO_2 electrodeposition,
11 which is consistent with the much higher amount of active $M(\bullet\text{OH})$ generated at a high
12 oxidation power anode like PbO_2 or, in other words, with the better performance of
13 $\text{PbO}_2(\bullet\text{OH})$ compared with $\text{RVC}(\bullet\text{OH})$. Indeed, PbO_2 anodes have a large oxygen evolution
14 overpotential and hence, the oxidation of the azo dye and its intermediates is promoted over
15 water oxidation at highly anodic potentials, thus enhancing the current efficiency referred to
16 solution decontamination. The existence of a maximum I_{max} at 45 ppi can be explained as
17 follows: 20 and 60 ppi RVC can be considered as 3D-like electrodes as well, as discussed
18 from Fig. 1-2; however, 45 ppi constitutes the optimum porosity since it presents a larger
19 surface area per unit volume compared to 20 ppi [10], and compared with 60 ppi, it allows
20 both the easier penetration of dissolved Pb^{2+} and three-dimensional growth of PbO_2 during the
21 electrodeposition process. In any case, I_{max} values were similar for 20-60 ppi, and all of them
22 can be considered as 3D-like $\beta\text{-PbO}_2$ -based materials with optimal three-dimensionality and
23 nanocrystallinity. As explained from Fig. 2, substrates of 80 and 100 ppi presented a much
24 smaller coated area, giving rise to planar-like electrodes. This allows justifying the lower I_{max}
25 values of 0.058 and 0.025 A, respectively. The most unexpected behaviour was found at 100

1 ppi, since the greatest I_{\max} was obtained using the raw RVC electrode. This result clearly
2 confirms the negative effect of the diminished coated area when increasing the pore density,
3 yielding modified materials with blocked pores whose inner volume remains almost
4 completely inaccessible for the electrolyte. To confirm this hypothesis, it is interesting to note
5 that the current value obtained for the RVC (100 ppi)/PbO₂ samples is very similar to that of a
6 planar PbO₂ electrode prepared onto a carbon plate [36].

7 A similar influence of porosity on I_{\max} can be observed in Fig. 3 for the
8 RVC/PbO₂/TiNT electrodes, but significantly greater values were obtained, reaching a
9 greatest $I_{\max} > 0.13$ A using 45 ppi (ca. 70% higher current than that obtained with raw RVC),
10 with similar values for 20 and 60 ppi. It is then evident the positive contribution of the TiNT
11 in the EO process. Such enhancement can be preeminently related to the synergistic action
12 between the greater adsorption of the dye molecules and some reaction intermediates in the
13 presence of the agglomerates of titanate nanotubes along with their EO by the PbO₂(•OH).
14 The action of these oxidising radicals is thus favoured when employing the composite anodes,
15 since the organic molecules become more largely adsorbed upon contact with the large
16 surface area β -PbO₂ nanocrystallites and the TiNT. It is worth to mention that the positive I_{\max}
17 at all porosities demonstrates that the greater number of adsorption sites did not cause anode
18 fouling by the organics or potential polymerisation derivatives, which is an evidence of the
19 simultaneous oxidative action of the PbO₂(•OH) at highly positive potentials.

20 3.3. *Electro-oxidation of Methyl Orange solutions*

21 Once confirmed that the coating of RVC yielded various electrodes that could
22 potentially exhibit a much better performance regarding the EO of organic molecules, all the
23 PbO₂-coated carbonaceous electrodes were used as the anodes to perform a protracted EO of
24 0.25×10^{-3} mol dm⁻³ Methyl Orange solutions with 0.05 mol dm⁻³ Na₂SO₄, at 0.6 A, pH 3.0

1 and 22.5 °C using a stainless steel mesh as the cathode. It is worth noting that Tang and Kong
2 obtained > 90% colour removal at times longer than 100 min when they degraded 0.16×10^{-3}
3 mol dm⁻³ Methyl Orange by EO at 25 mA cm⁻² contained in 200 cm³ of electrolyte using
4 various planar doped-β-PbO₂-coated Ti anodes [29].

5 [Fig. 4a-b](#) shows the influence of the RVC porosity on the percentage of color removal
6 vs. time during the electrolyses using RVC/PbO₂ and RVC/PbO₂/TiNT anodes, respectively.
7 The trials were also carried out with raw RVC anodes but, as expected, they became severely
8 degraded. Conversely, all the coated anodes led to > 98% absorbance decay in 45 min, with
9 no presence of carbon in solution thanks to the good isolation of the carbonaceous substrates.
10 In [Fig. 4a](#), it can be seen that RVC (45 ppi)/PbO₂ yielded the quickest decolourisation during
11 the first 10 min of electrolysis. For example, after 2.5 min, 50% colour removal was achieved
12 instead of ca. 40% reached with the other anodes. The superiority exhibited at this porosity
13 can be related to the greater electroactive area discussed from [Fig. 3](#). From 15 min, the results
14 became quite similar for all the electrodes, which can be explained by the appearance and
15 adsorption of degradation by-products that were simultaneously oxidized by PbO₂(•OH),
16 thus decelerating the removal of the initial dye. Note that even the RVC (100 ppi)/PbO₂
17 anode, whose electroactivity was lower due to the smaller coated area, was able to reach the
18 complete colour removal at 60 min. This suggests that large amounts of active PbO₂(•OH) are
19 generated even with the sole participation of the external surface.

20 As can be observed in [Fig. 4b](#), composite anodes yielded a faster decolourisation of
21 Methyl Orange solutions, especially those of 20-60 ppi. Again, > 98% absorbance decay was
22 attained at 45 min in all cases but, at 2.5 min, the RVC (45 ppi)/PbO₂/TiNT anode yielded the
23 quickest decolourisation (60%) among all the electrodes. The superiority of this material was
24 evident up to 10 min, as discussed above for the RVC/PbO₂. This finding confirms that the
25 TiNT really act as an important anchorage site for the organics, thus enhancing the efficiency

1 of $\text{PbO}_2(\bullet\text{OH})$ to cause their progressive degradation. The reaction kinetics using 80 and 100
2 ppi was not significantly enhanced, as expected from their much lower electroactivity (Fig. 3).
3 In fact, the presence of TiNT appears to be detrimental in these two cases if compared with
4 Fig. 4a, which can be explained by the excessive blockage of PbO_2 sites, thus resulting in a
5 lower accumulation of active $\text{PbO}_2(\bullet\text{OH})$. In conclusion, the RVC (45 ppi)/ PbO_2/TiNT anodes
6 yielded the largest decolorisation efficiency and thus, they were used for investigating the
7 effect of Methyl Orange concentration and electrolyte composition on colour and TOC
8 removal, as compared with RVC (20 ppi)/ PbO_2/TiNT .

9 Fig. 5a depicts the effect of Methyl Orange content on the percentage of colour removal
10 vs. time during the electrolyses of dye solutions with $0.05 \text{ mol dm}^{-3} \text{ Na}_2\text{SO}_4$, at 0.6 A, pH 3.0
11 and $22.5 \text{ }^\circ\text{C}$ using an RVC (20 ppi)/ PbO_2/TiNT or RVC (45 ppi)/ PbO_2/TiNT anode. The
12 much slower absorbance decay when treating $0.50 \times 10^{-3} \text{ mol dm}^{-3}$ Methyl Orange is evident
13 with both anodes. Certainly, $> 98\%$ colour removal was attained at 45 min in all cases but, at
14 10 min, 84% and 57% was reached from solutions containing 0.25 and $0.50 \times 10^{-3} \text{ mol dm}^{-3}$
15 dye, respectively. The acceleration during the first minutes at the lowest content can then be
16 related to the smaller extent of simultaneous oxidation reactions of intermediates, yielding
17 larger amounts of $\text{PbO}_2(\bullet\text{OH})$ readily available to degrade the dye. The electrolyses were
18 extended for several hours in order to obtain information on the behaviour of TOC (not
19 shown). The RVC (45 ppi)/ PbO_2/TiNT anode was a good material not only to provide the
20 complete decolourisation of solutions in less than 60 min, but also to gradually mineralise
21 them. For example, at 240 min, 55% and 60% TOC abatement was reached for solutions
22 containing 0.25 and $0.50 \times 10^{-3} \text{ mol dm}^{-3}$ Methyl Orange (initial TOC of 42 and $84 \times 10^{-3} \text{ g C}$
23 dm^{-3}), respectively. This means that, although a larger number of organic molecules has to be
24 degraded starting at a higher dye content, the $\text{PbO}_2(\bullet\text{OH})$ act more efficiently on them because
25 parasitic reactions involving radicals (i.e., self-destruction, reaction with inorganic ions)

1 become minimised. This also justifies the slower decolourisation kinetics shown in Fig. 5a.
2 This can also be demonstrated from the higher mineralization current efficiency (MCE) and
3 lower energy consumption per unit TOC mass (EC_{TOC}) for the electrolyses with larger dye
4 content, as calculated from the equations provided elsewhere [20,55]. MCE was 2.6% and
5 5.7%, whereas EC_{TOC} was 1.5 and 0.69 kWh (g TOC)⁻¹ for electrolyses with 0.25 and $0.50 \times$
6 10^{-3} mol dm⁻³ Methyl Orange, respectively.

7 Fig. 5b reveals the very positive influence of the presence of NaCl in the supporting
8 electrolyte on color removal for the treatment of 0.25×10^{-3} mol dm⁻³ Methyl Orange
9 solutions using the RVC (45 ppi)/PbO₂/TiNT anode at 0.6 A. The absorbance decay was
10 extremely fast when 0.05 mol dm⁻³ NaCl or mixtures with 0.05 mol dm⁻³ Na₂SO₄ were
11 employed, achieving > 95% decolourisation at 2.5 min. This fact can be related to the
12 generation of active chlorine, a very powerful bleaching agent, upon oxidation of Cl⁻ anion at
13 PbO₂ [5,12], and opens the door to the application of this kind of composite anodes to the
14 treatment of industrial wastewater where Cl⁻ and SO₄²⁻ anions tend to be present, as in the
15 case of dye wastewater [5].

16 4. Conclusions

17 This work demonstrates the key influence exerted by the substrate porosity when
18 preparing three-dimensional pure PbO₂ coatings and composites with TiNT onto RVC.
19 Although all the electrodes led to the total decolourisation of dye solutions, only those
20 manufactured using RVC of 20-60 ppi ensured a complete isolation of the substrate that
21 prevented the material destruction upon prolonged electrolyses. Among them, the RVC (45
22 ppi)/PbO₂/TiNT anodes have shown the best performance in terms of solution
23 decontamination, which arises from their optimum coating thickness and PbO₂/TiNT
24 distribution along the whole volume. The elongated TiNT enhances the adsorption of organic

1 molecules, which are then more efficiently degraded by $\text{PbO}_2(\cdot\text{OH})$, as demonstrated by the
2 fast decolourisation and the significant mineralisation of dye solutions at ca. 45 and 240 min,
3 respectively. Therefore, these 3D-like $\beta\text{-PbO}_2$ anodes constitute suitable materials to
4 minimise the mass transport limitations that typically appear when employing commercial
5 plates, and could potentially serve as cheaper large surface electrodes for industrial
6 wastewater treatment.

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13

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9

1 **Figure captions**

2 **Figure 1.** SEM images of coatings prepared on RVC from solutions containing 1.0 mol dm^{-3}
3 $\text{Pb}(\text{CH}_3\text{SO}_3)_2$ and 0.2 mol dm^{-3} MSA at 2.5 A and $60 \text{ }^\circ\text{C}$ for 30 min in the absence
4 (RVC/ PbO_2) or presence (RVC/ PbO_2 /TiNT) of previously synthesized TiNT: (a) RVC (20
5 ppi)/ PbO_2 ; (b) RVC (60 ppi)/ PbO_2 ; (c) Magnified image from coating shown in (b); (d) RVC
6 (60 ppi)/ PbO_2 /TiNT coating.

7 **Figure 2.** SEM images of cross sections of (a) RVC (60 ppi)/ PbO_2 and (b) RVC (100
8 ppi)/ PbO_2 , and (c) dependence of thickness of PbO_2 deposit with RVC porosity for all the
9 RVC/ PbO_2 coatings prepared.

10 **Figure 3.** Effect of RVC porosity on the maximum current measured from cyclic
11 voltammograms (5th cycle) recorded at the RVC, RVC/ PbO_2 and RVC/ PbO_2 /TiNT electrodes
12 from + 1.4 V to + 2.0 V, in a $0.25 \times 10^{-3} \text{ mol dm}^{-3}$ Methyl Orange solution with 0.05 mol dm^{-3}
13 Na_2SO_4 , at pH 3.0 and $22.5 \text{ }^\circ\text{C}$. Potential sweep rate: 10 mV s^{-1} . Counter electrode: Pt gauze.
14 Reference electrode: $\text{Ag}|\text{AgCl}|\text{KCl}(\text{sat.})$.

15 **Figure 4.** Influence of the RVC porosity on the percentage of colour removal *vs.* time during
16 the electrolyses of $0.25 \times 10^{-3} \text{ mol dm}^{-3}$ Methyl Orange solutions with 0.05 mol dm^{-3} Na_2SO_4 ,
17 at 0.6 A, pH 3.0 and $22.5 \text{ }^\circ\text{C}$ using a stainless steel mesh as the cathode and different RVC-
18 coated anodes: (a) RVC/ PbO_2 ; (b) RVC/ PbO_2 /TiNT.

19 **Figure 5.** (a) Effect of Methyl Orange concentration on the percentage of colour removal *vs.*
20 time during the electrolyses of dye solutions with 0.05 mol dm^{-3} Na_2SO_4 , at 0.6 A, pH 3.0 and
21 $22.5 \text{ }^\circ\text{C}$ using a stainless steel mesh as the cathode and an RVC (20 ppi)/ PbO_2 /TiNT or RVC
22 (45 ppi)/ PbO_2 /TiNT anode. (b) Influence of the supporting electrolyte (each salt at

1 concentration of 0.05 mol dm^{-3}) on colour removal for the treatment of $0.25 \times 10^{-3} \text{ mol dm}^{-3}$
2 Methyl Orange solutions using the RVC (45 ppi)/ PbO_2 /TiNT anode.

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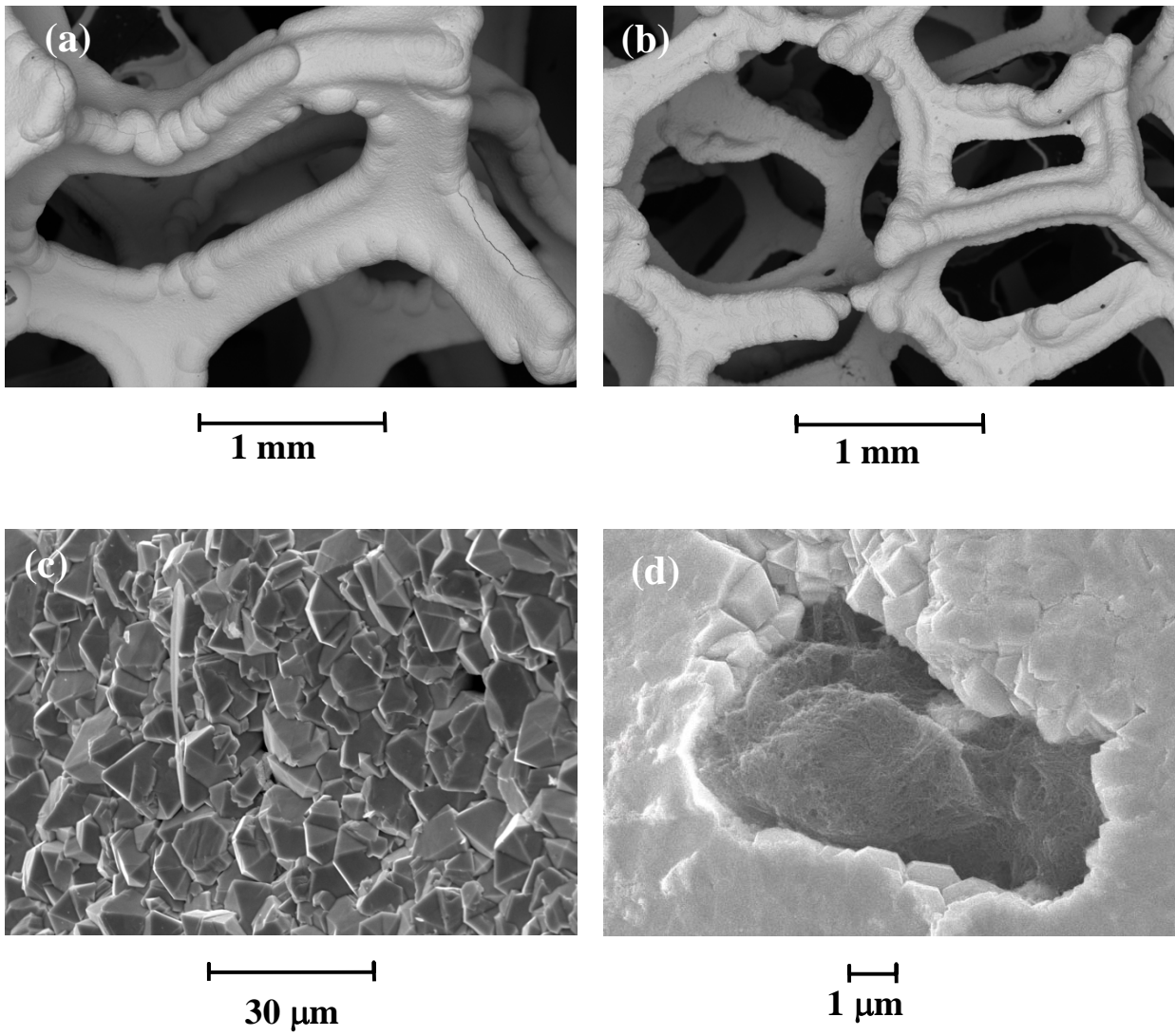
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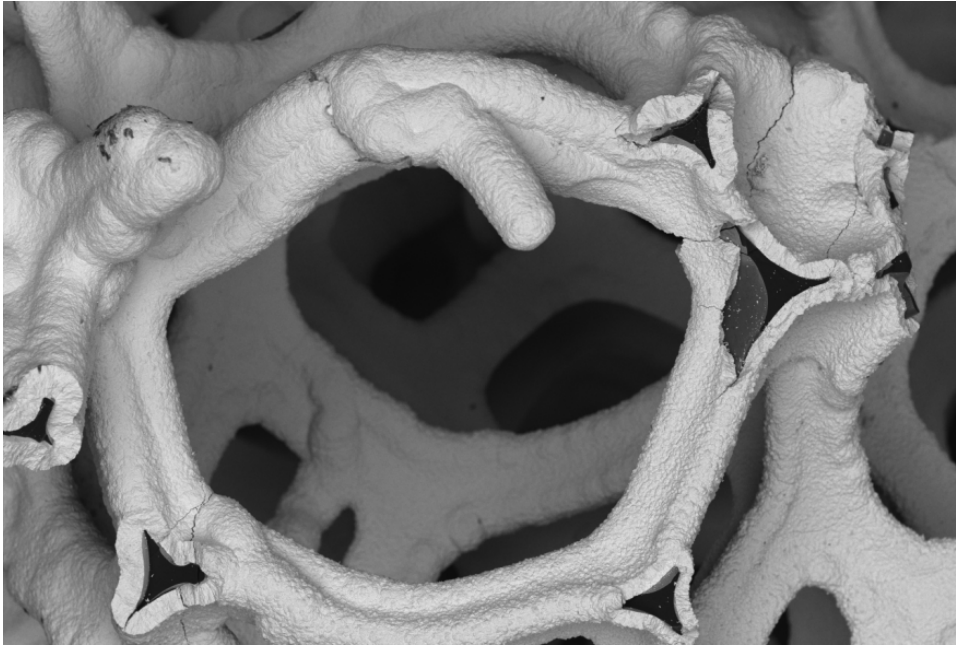
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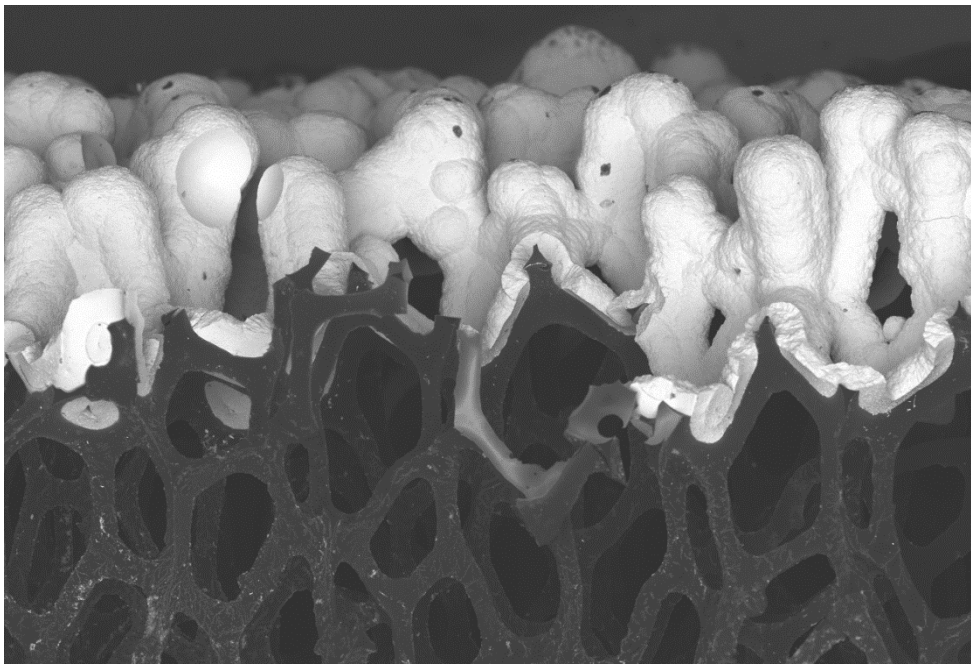
Figure 1



500 μm

Figure 2a

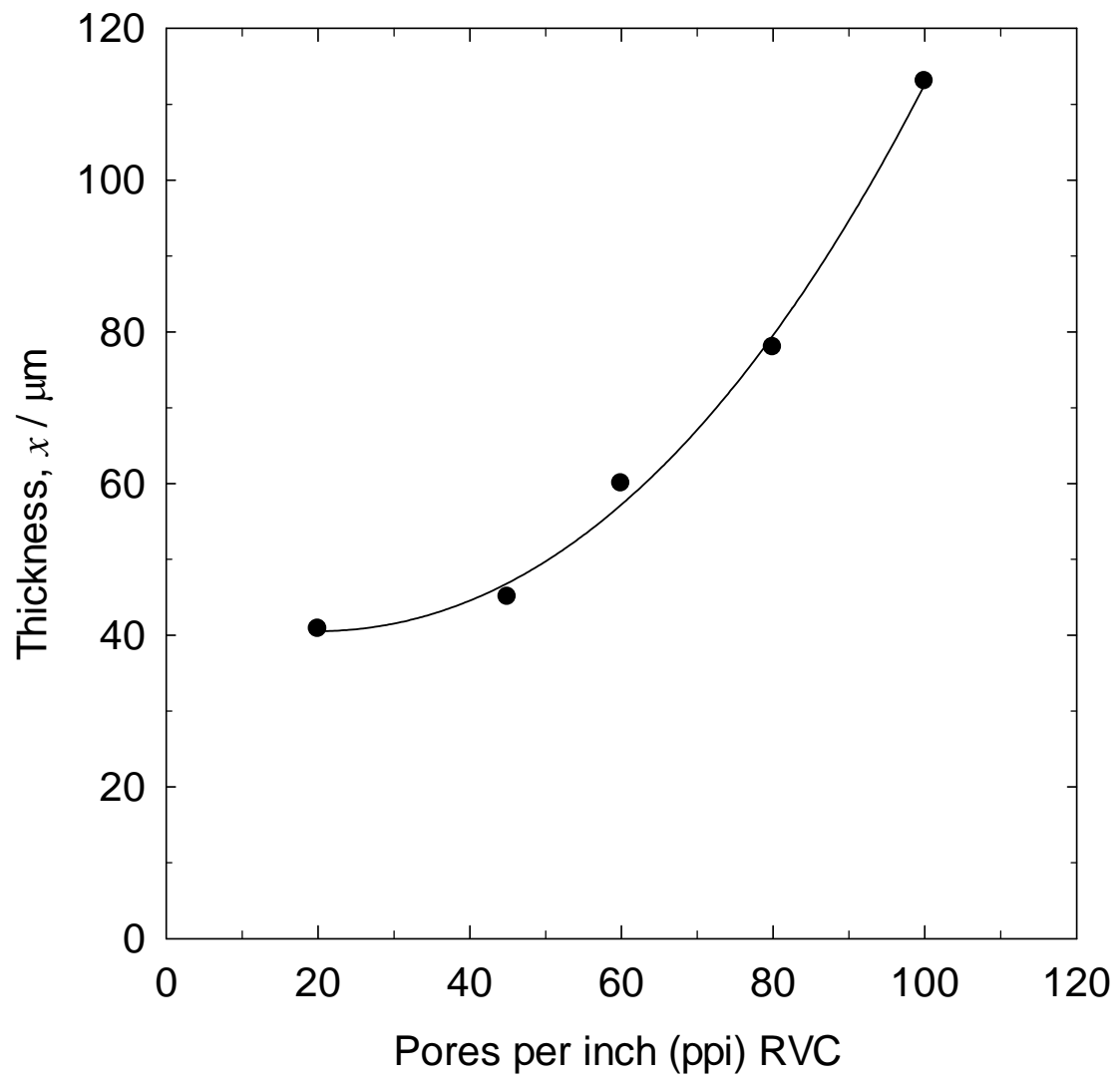
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500 μm

Figure 2b

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Figure 2c

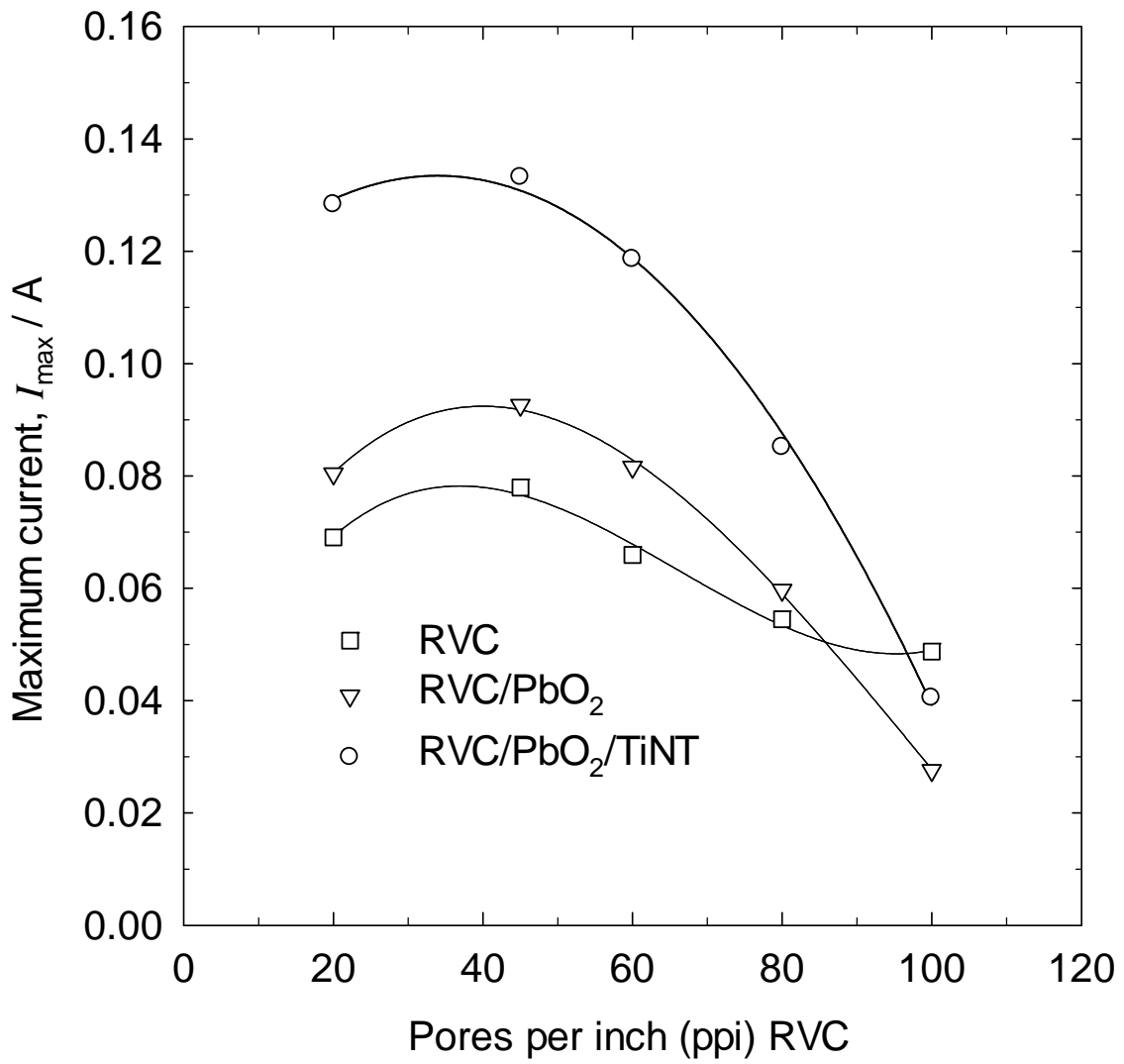


Figure 3

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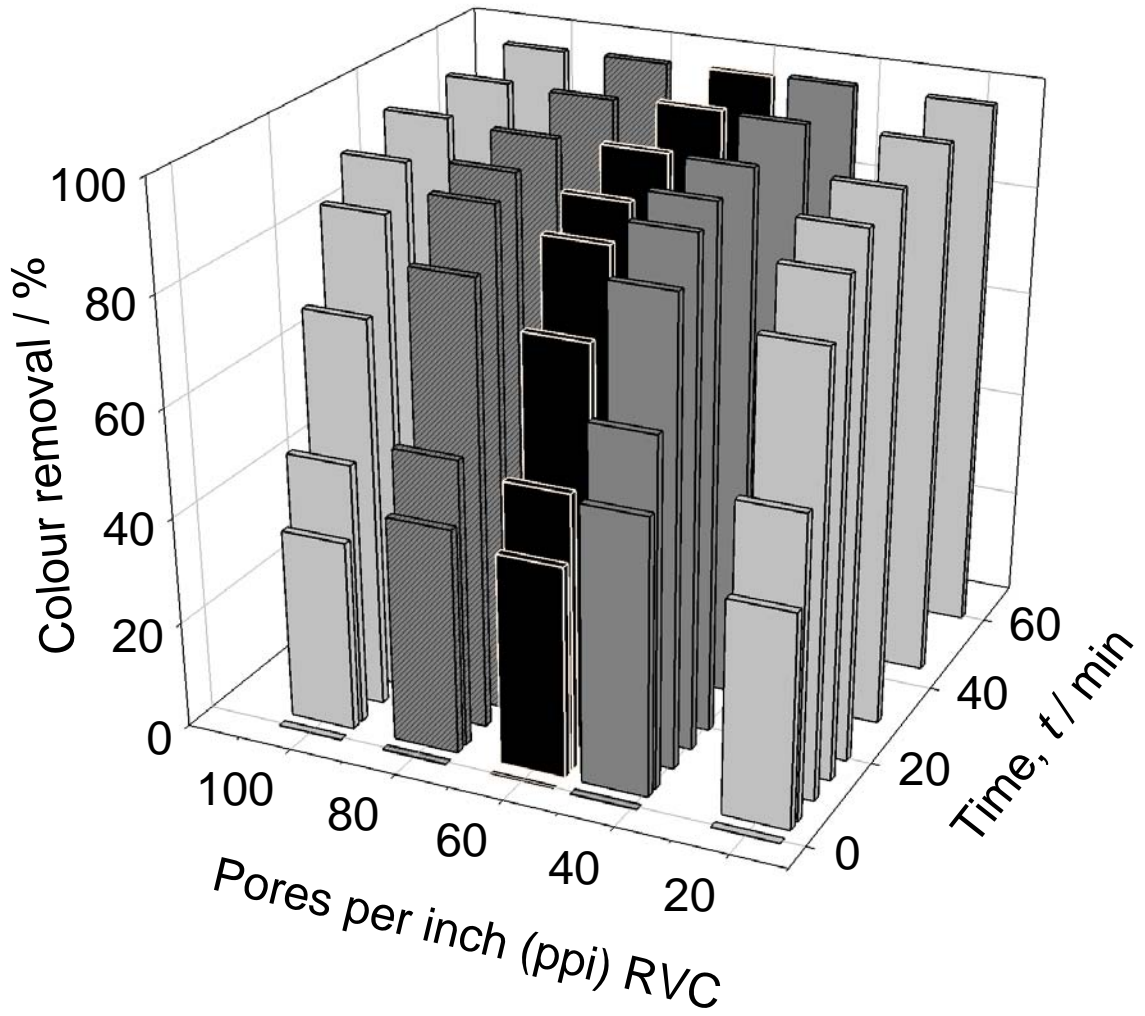


Figure 4a

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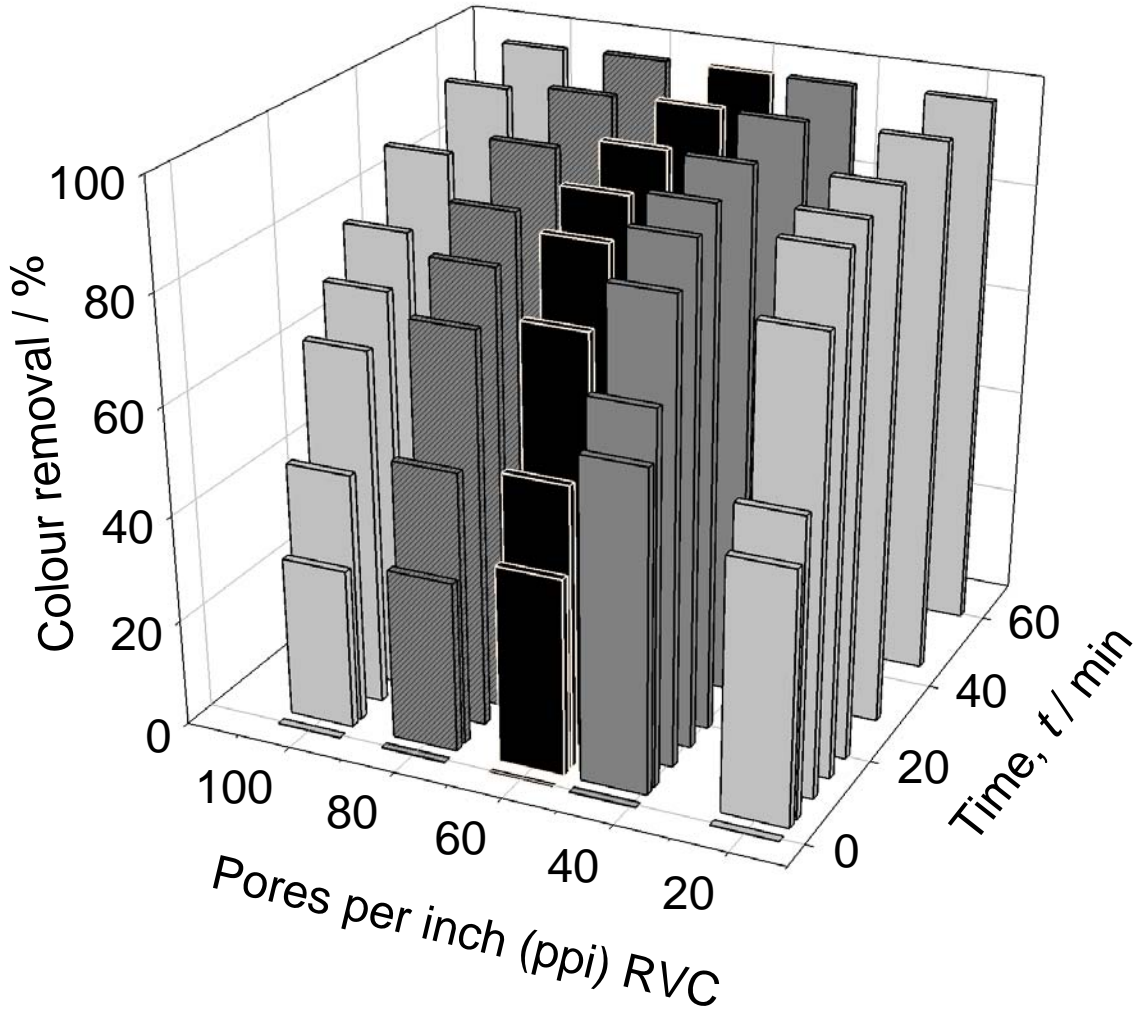


Figure 4b

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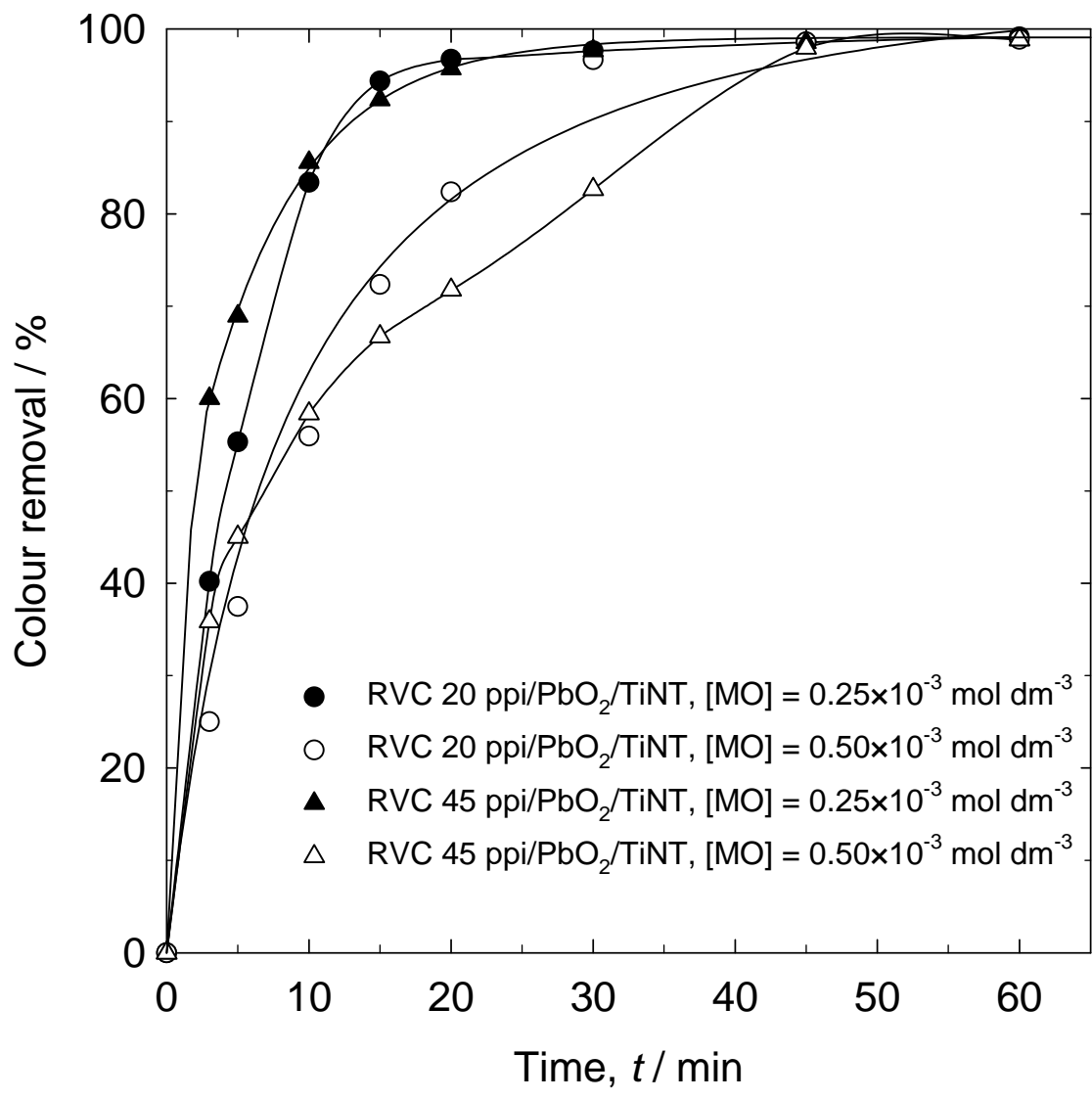
7

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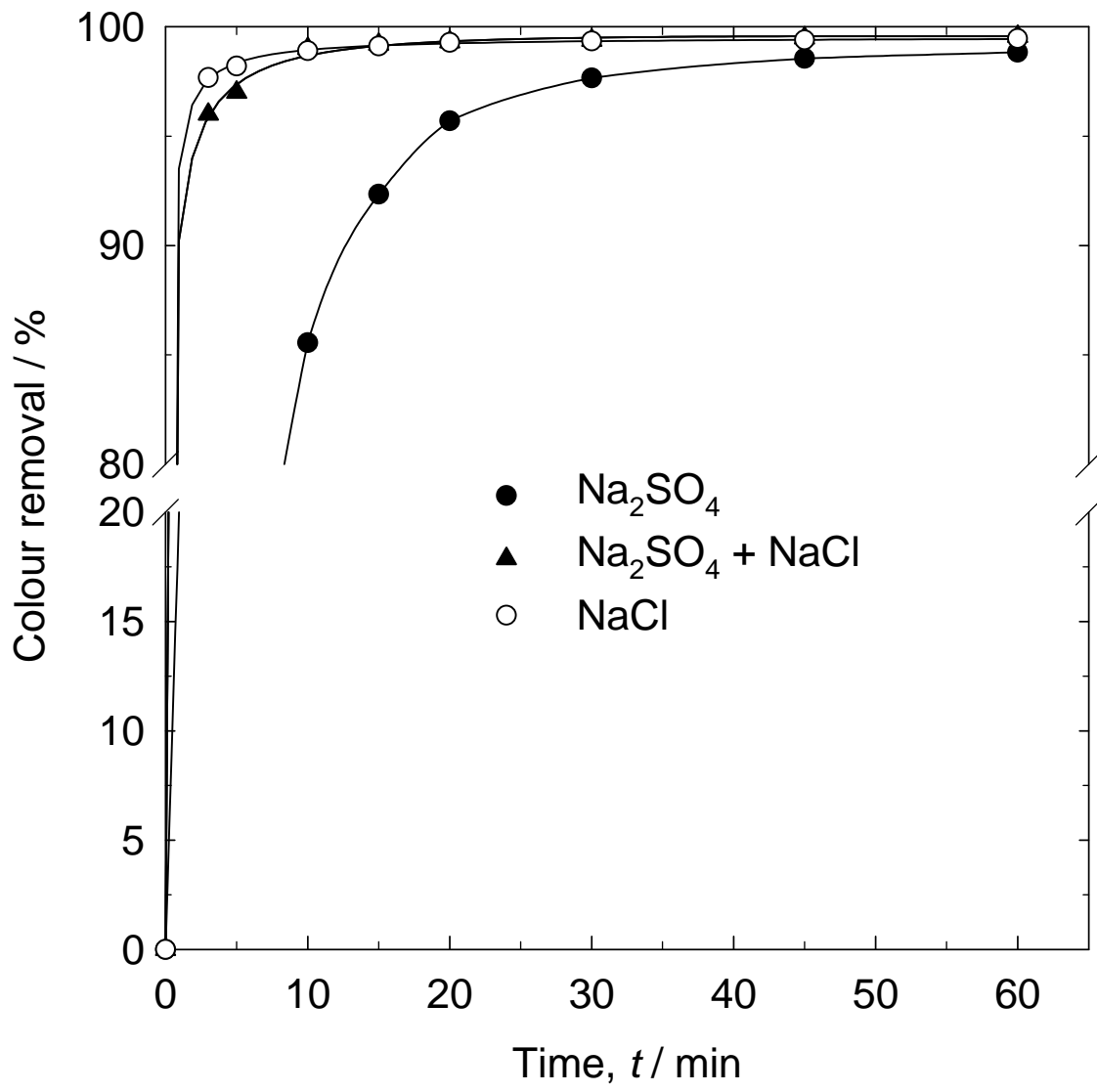
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Figure 5a

1
2
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Figure 5b