Crystal, Magnetic and Dielectric Studies of the 2D Antiferromagnet: β -NaMnO₂

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ABSTRACT

In this paper we present our recent studies on the crystal, magnetic and dielectric properties of the β -NaMnO₂. Experimental results of neutron powder and electron diffraction combined with measurements of the dielectric permittivity suggest that the β -NaMnO₂ is an excellent candidate for studying the coupling between the magnetic and electric degrees of freedom. Neutron powder diffraction data reveal the existence of a commensurate and an incommensurate magnetic structure at 200 K and below 100 K, respectively. Dielectric anomalies which appear at the temperature regions where the two magnetic structures emerge, indicate the appearance of magnetodielectric coupling.

Keywords: magnetic frustration, neutron powder diffraction, dielectrics, magneto-electric coupling.

1. INTRODUCTION

The coupling between the magnetic and electric degrees of freedom has been proved to be a particularly appealing subject of research, since the breathtaking revival of the magneto-electric effect. This phenomenon was firstly realized on the Cr₂O₃ in the early 1960's [1]-[4] and has triggered a tremendous interest for two reasons. The first one arises from the need to address the theoretical frame and to describe the magnetic symmetry considerations which would explain the mechanisms that generate and allow the phenomenon. The second reason is the potential for technological applications such as multifunctional devices. The magnetoelectric effect allows the control of the magnetic properties *via* an external electric field and *vice versa*.

Most of the compounds that exhibit the magneto-electric coupling belong to the perovskite family, such as the TbMnO₃,^[5] the YMnO₃.^[6] In "type II multiferroics", widely known as magnetic ferroelectrics,^[7], ^[8] the ferroelectric order appears below the temperature of the magnetic transition. Nevertheless, the explanation of the underlying mechanism is not only based on the appearance of specific spin configurations such as cycloid or proper, but requires also a complete understanding of the crystal and magnetic symmetry. The recent discovery that the magnetic frustration is related to the appearance of the magnetoelectricity,^[9] lead to the study of the compounds with the chemical formula ABO₂ (B is a 3d magnetic ion such as Fe, Mn, and A: non magnetic ion usually Na, Li, Cu) as possible magnetoelectrics. Indeed, the compounds CuFeO₂,^[10] ACrO₂ (A: Cu, Cr) ^[11] and AgFeO₂ ^[12] have been reported to exhibit the magnetoelectric coupling. Most interestingly, it has been reported that the mechanism, which explains the phenomenon in these compounds, is not related to the Dzyaloshinskii Moriya interaction,^[13]-^[15] as it happens in several magnetoelectrics. Recent studies suggest that the underlying mechanism of the magnetoelectric coupling in the ABO₂ compounds is the spin orbit interaction of the d-p hybridization.^[16], ^[17]

In view of the possible magneto-electric coupling of the β -NaMnO₂ we have studied the crystal, magnetic and dielectric properties of the compound. β -NaMnO₂ (Mn⁺³,3d⁴, S=2), belongs to a vast family of oxides, namely the Na_xMnO₂ oxides, which has been first reported by Parant et al .^[18] The Na_xMnO₂ have been widely studied as initial precursors in chemical reactions which provide various modified layered structures such as the birnessite structure. ^[19]- ^[21] Moreover Na_xMnO₂ have attracted a lot of industrial interest due to their possible applications as cathode materials

in rechargeable batteries. [122]-[24] In the β -NaMnO₂, the Mn cations map out on a corrugated 2D triangular lattice, whereas the MnO₆ edge-sharing octahedra are separated by two layers of Na cations (figure 1). The triangular arrangement of the Mn, is favorable for the existence of competing magnetic interactions between the magnetic cations, as it has been proved for the case of the α -NaMnO₂. [25]-[28]

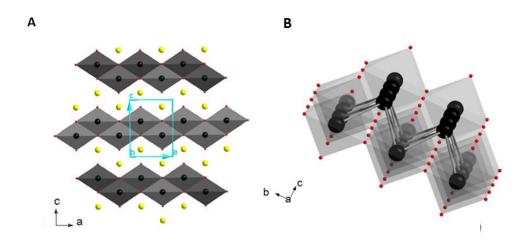


Figure 1.The crystal structure of the β -NaMnO₂. The black, red and yellow spheres represent the manganese, oxygen and sodium atoms, respectively. Figure 1a shows the layered structure, composed by sheets of MnO₆ edge sharing octahedra separated by two layers of sodium cations. The aqua lines represent the cell edges. The 2D triangular arrangement of the Mn cations is shown in figure 1b.

2. EXPERIMENTAL PART

2.1 Sythesis of polycrystalline powder of β-NaMnO₂

Polycrystalline powder specimens of β -NaMnO₂ have been synthesized by using high temperature solid state reaction at 950°C. The synthesis is based on the β -NaMnO₂ synthetic protocol reported by Parant. ^[18]

2.2 Characterization of β-NaMnO₂

The purity and crystallinity of the β-NaMnO₂ phase has been checked by X-Ray diffraction on a Rigaku D/MAX-2000H rotating Cu anode diffractometer (λ =1.5406 Å). The sample is placed on the holder and sealed with a mylar foil in the glove box, so that it is not exposed in the atmosphere during the X-Rays experiment. The synthesized compound is a phase-pure β-NaMnO₂ which crystallizes in the orthorhombic Pmnm symmetry (a=2.852 Å, b=6.33 Å, c=4.77Å, 90°, 90°, 90°). Thermogravimetric measurements were carried out on a SDT-Q600 while the sample was heated with a constant heating rate of 10°C /min under Argon atmosphere. Neutron powder diffraction experiments took place on the BT1 diffractometer, with the wavelength of 2.0787 Å at the NCNR-NIST, USA. High resolution powder diffraction measurements were conducted on the 11 BM beamline, of the Argonne National Laboratory, in USA using the wavelength of 0.41395 Å. The dielectric constant and dielectric loss, have been measured on 5mm pellets of β-NaMnO₂ by an Agilent 4284 LCR bridge (at 1 V or 2 V oscillation voltage amplitude) at the temperature range between 77 K-300 K. An average pressure of 6.5x108 Pa was applied with a 5mm homemade steel press in the glove box in order to form the β-NaMnO₂ pellets. Silver paste contacts have been made on the 5 mm β-NaMnO₂ pellets which would be used in the magneto-dielectric measurements. The samples were placed in Janis Cryostat of continuous nitrogen flow and cooled starting from room temperature to 77 K under zero electric and magnetic field. The capacitance and dielectric loss values were recorded upon heating at various frequencies. The calculation of the real part of the dielectric permittivity ε' has been done using the equation $\varepsilon'=\varepsilon_0Cg$, where ε_0 is the dielectric permittivity of the vacuum (8.85x10⁻¹² F/m, C is the capacitance (F) and g is the geometric factor (m) of the sample calculated by dividing the conductive area of the sample (A) with its thickness (d).

3. RESULTS AND DISCUSSION

The β -NaMnO₂, crystallizes in the orthorhombic structure at room temperature. It has been indexed upon the orthorhombic structure [18] (Space-group Pmnm a=2.852(3) Å, b=6.31(2) Å c=4.77(1) Å, ICSD 16271) as shown in figure 2. Red ticks correspond to the indexing based on the orthorhombic cell, whereas the broad intense peak between 20deg-30deg and the one at 38deg pointed with the grey asterisk are attributed to the mylar film used in order to protect the sample from moisture. The peak at 16.68 deg pointed with the blue tick corresponds to the 001 peak of the a-NaMnO₂ phase. An unusual characteristic observed at the X rays Powder Diffraction (XRPD) pattern is the broadening of the peaks at 38-40deg and more clearly at 55 deg.

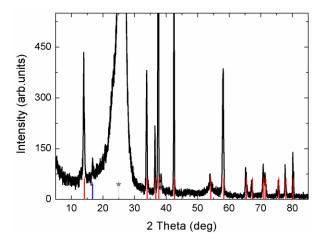


Figure 2. Room temperature XRD of the β -NaMnO₂ polycrystalline powder indexed upon the orthorhombic cell. Red ticks correspond to the indexing of the orthorhombic cell of β -NaMnO₂. The blue tick at 16.68 deg corresponds to the 001 peak of the monoclinic cell of α -NaMnO₂. The peaks shown with the grey asterisks at 25 deg and 38 deg are attributed to the my lar film.

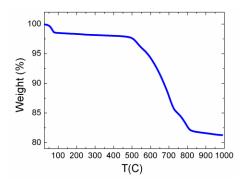


Figure 3. A graph showing weight loss (%) versus temperature for β-NaM nO₂.

In order to evaluate the stability of the β -NaMnO₂ phase in respect with the temperature, thermogravimetric measurements were carried out in the temperature range between 25°C and 1000°C (figure 3). The initial weight decrease of less than 1.5% around 100°C is attributed to the evaporation of water, which could be absorbed by the pores of the sample's surface while transferring the sample out of the glove box to the alumina pan of the SDT-Q600. The chemical composition of the sample remains stable at the temperatures 100°C -500°C, whereas the 16% weight loss noticed between 500°C - 800°C is mostly attributed to the loss of sodium which occurs at temperatures higher than 400 °C. Small amounts of unreacted Na₂CO₃, which has been used as a starting material, may also contribute to the weight loss in this temperature region.

The real and imaginary part of the dielectric permittivity β -NaMnO₂ reveals two dielectric anomalies, at 90 K and 200 K, as shown in figure 4. Two β -NaMnO₂ pellets both synthesized with the same synthetic protocol have been measured upon heating with 500 kHz and 100 kHz applied in the measurement shown in figure 4a and 4b, respectively. The dielectric permittivity measurement shown in figure 4a was carried out using an electric field of 700 V/m. This value of electric field is comparable with the 685 V/m applied in the measurement shown with the red line in figure 4b. Dielectric anomalies are observed in both frequencies tested, namely in 100 kHz and 500 kHz. Interestingly, the dielectric anomaly at 90 K is greatly enhanced when the applied electric field is doubled from 685 V/m to 1370 V/m as it is observed in the measurement represented with the blue line in figure 4b. On the contrary, no difference is observed in the hump of the 200 K when the electric field is increased.

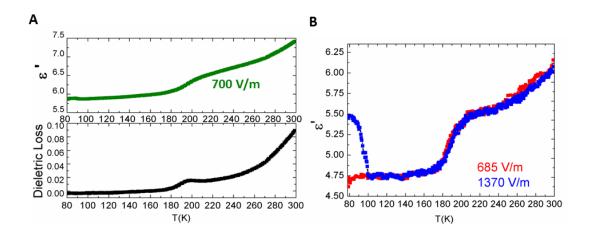


Figure 4. Real (ε') and imaginary part (ε" or dielectric loss) under zero magnetic field on two pellets of β -NaMnO₂ measured at the frequencies of 100 kHz and 500 kHz at 4a and 4b figures, respectively. The dielectric anomaly at 90 K is majorly affected when the applied electric field is increased.

Neutron powder diffraction (NPD) data obtained at high resolution diffractometer BT1, shed light on the origin of the two dielectric anomalies that have been observed. Figure 5 shows temperature evolution of the square root (SORT) of the normalized intensity versus d-spacing (1.5 Å-5.0 Å) of the NPD patterns that have been obtained at the temperatures 300K, 200K, 150K, 100K and 50 K. The blue and green ticks, shown in the NPD pattern of figure 5, correspond to the indexing of the crystal and magnetic structure, respectively. Anisotropic broadening, asymmetric and satellite peaks are the dominant characteristics of the NPD pattern indicating a rather complex crystal structure. Moreover, there are two striking characteristics in the NPD patterns at the magnetic Bragg peaks observed in the dspacing of 3.28 Å and 4.6 Å: a) The first one is the emergence of the peaks below 200 K which corresponds to a commensurate magnetic structure. At the same d-spacing, below 100 K, at the pattern of 50 K, satellite magnetic peaks appear, pointing towards the emergence of an incommensurate magnetic structure. The existence of dielectric anomalies at the same temperature regions where magnetic ordering appears is undoubtedly a signature of magnetodielectric coupling. The anomalies in the real part of the dielectric permittivity ε' correspond to the emergence of the polarization (since $P = \varepsilon_0(\varepsilon' - 1)E$, where P is the polarization, ε_0 is the dielectric permittivity of the vacuum, and E is the applied electric field). Previous studies in magnetoelectric compounds with the stoichiometry ABO2 such as CuFeO₂ and ACrO₂ [11]-[12] with triangular lattice topology of the magnetic cations have proved that the mechanism which generates the magnetodielectric coupling is connected with the magnetic frustration which leads to a complex spin configuration.

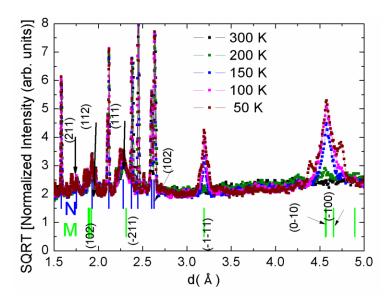


Figure 5. Neutron Powder Diffraction (NPD) patterns of the β -NaM nO₂ obtained at different temperatures. Magnetic Bragg peaks emerge at 200 K and below 100 K.

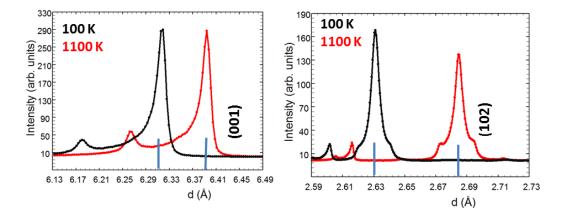


Figure 6. The 001 reflection on the left and 102 reflection on the right as shown on the patterns obtained at the 11BM syncthron of β -NaMnO₂ reveal sattelite and asymetric peak at low (100 K- black line) and high (1100 K-red line) temperatures.

To further explore the crystal structure, high resolution diffraction experiments were carried out in the 11BM synchrotron on the temperature range between 100 K-1100 K (figure 6). Surprisingly, the broadening of the peaks, as well as the satellites and asymmetric peaks remain at temperatures even as high as 1100 K. The analysis of the 11 BM data, on the basis of a modulated structure (superstructure) gave no acceptable results, since the satellites of the (102) peak could not be indexed. These two unexplained peaks are so close to the main reflection that a corresponding vector would be too short in the reciprocal space to account for some structural modulation in the crystal. The synchrotron results and the relevant analysis point towards the existence of irregularities in the building diffraction domains which should be checked with electron diffraction.

4. CONCLUSIONS

Magnetodielectric coupling of polycrystalline β -NaMnO₂ is reported. The coupling is evident by the appearance of two anomalies in the dielectric permittivity around 90 K and at 200 K, temperature regions in which according to the neutron diffraction patterns incommensurate and commensurate magnetic ordering develops. Increasing the applied electric field greatly affects the low temperature dielectric anomaly. A complete understanding of the mechanism that generates the magnetoelectric coupling requires the knowledge of the crystal and magnetic structure. The crystal structure shows irregularities which are observed in the neutron and synchrotron diffraction patterns. A complete explanation for the observed crystal structure complexity requires the observation of the β -NaMnO₂ by electron diffraction and high resolution transmission electron microscopy, which are ongoing experiments.

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