

Clathrate structure determination by combining crystal structure prediction with computational and experimental ^{129}Xe NMR spectroscopy[†]

Marcin Selent,[‡] Jonas Nyman,[‡] Juho Roukala, Marek Ilczyszyn, Raija Oilunkaniemi, Peter J. Bygrave, Risto Laitinen, Jukka Jokisaari, Graeme M. Day,^{*} and Perttu Lantto^{*[a]}

We present an approach for the structure determination of clathrates using NMR spectroscopy of enclathrated xenon to select from a set of predicted crystal structures. Crystal structure prediction methods have been used to generate an ensemble of putative structures of *o*- and *m*-fluorophenol, whose previously unknown clathrate structures have been studied by ^{129}Xe NMR spectroscopy. The high sensitivity of the ^{129}Xe chemical shift tensor to the chemical environment and shape of the crystalline cavity makes it ideal as a probe for porous materials. The experimental powder NMR spectra can be used to directly confirm or reject hypothetical crystal structures generated by computational prediction, whose chemical shift tensors have been simulated using density functional theory. For each fluorophenol isomer we find one predicted crystal structure whose measured and computed chemical shift tensors agree within experimental and computational error margins and these are thus proposed as the true fluorophenol xenon clathrate structures.

^[a]M. Selent, Dr. J. Roukala, Prof. J. Jokisaari, Dr. P. Lantto

NMR Research Unit, Faculty of Science, University of Oulu, FI-90014 Oulu (Finland); Tel: +358 294481312; E-mail: perttu.lantto@oulu.fi

J. Nyman, Dr. P. J. Bygrave, Prof. G. M. Day

School of Chemistry, University of Southampton, Southampton (UK); Tel: +44 2380599174; E-mail: g.m.day@soton.ac.uk

Dr. M. Ilczyszyn

Faculty of Chemistry, Wrocław University, 50-383 Wrocław, Joliot Curie 14, Wrocław (Poland)

Dr. R. Oilunkaniemi, Prof. R. Laitinen

Laboratory of Inorganic Chemistry, University of Oulu, FI-90014, Oulu (Finland)

[†] Supporting information (ESI) for this article is available on the WWW under

<http://dx.doi.org/10.1002/chem.2014xxxxx>:

supporting_info.pdf: Detailed description of crystal structure prediction (CSP) and its performance for known high density forms of crystals as well as the method to compute guest to host volume ratios. Revised Williams99 parameters for H...A interactions, as well as potential parameters for F and Xe are explained. Details of DFT structure optimization, Xe NMR calculations, and modeling of Xe dynamics. Xe NMR results at different levels and their sensitivity for structure. Details of powder X-ray diffraction (PXRD) measurements and results. XeClathrateCSPNMR_CIFs.zip: a zip-package containing clathrate structures in CIF format. More detailed description of the cif-files in the supporting_info.pdf.

[‡] The authors have made equal contributions to this article

1 Introduction

Over the past few years, the combined use of solid state nuclear magnetic resonance (NMR) and computational methods has developed into a practical method for determining crystal structures.¹ This is due to the sensitivity of NMR chemical shifts to the molecular environment in a crystal, and the reliability of methods for predicting the relative NMR shielding of atoms in different environments. The approach involves first predicting the full set of low energy crystal structures available to a molecule, followed by chemical shielding calculations, which are matched against the measured chemical shifts of a material. Applications to organic crystals have found that, given a set of predicted crystal structures, the isotropic ^1H chemical shifts are usually sufficient to identify which of the predicted structures corresponds to the material under investigation.²

In 1972–73 Kazankin *et al.*^{3,4} reported that xenon clathrates can be formed from several monosubstituted phenol compounds. With the exception of phenol,⁵ hydroquinone,^{5–8} *p*-cresol⁹ and *p*-fluorophenol,¹⁰ Xe clathrates have not been described in the English literature and have remained relatively unknown in the West. The crystal structures of *o*- and *m*-fluorophenol xenon clathrates have, to our knowledge, never previously been determined. The NMR properties of Xe included in porous structures gives a potentially sensitive probe of clathrate structure, which we explore here as a means of determining the structures of the *o*- and *m*-fluorophenol xenon clathrates.

The first ^{129}Xe NMR experiments were performed in 1951,¹¹ but only a modest number of further studies appeared until the early 1980s when Ripmeester and Davidson investigated enclathrated xenon¹² and Ito and Fraissard proposed the use of ^{129}Xe NMR for probing the properties of zeolites.¹³ Today, the application range of ^{129}Xe NMR extends from materials property studies of micro- and mesoporous solids *via* polymers, liquid crystals, proteins and biosensors to magnetic resonance imaging (MRI) for medical purposes.^{14–17}

The reason for using xenon to probe the properties of various materials arises from the high sensitivity of the shielding of ^{129}Xe to its local environment, and that the variability in the shielding stems exclusively from environmental effects.^{18–20} Another reason is the good NMR receptivity of ^{129}Xe , which is about 33 times that of ^{13}C . A disadvantage in some cases may be the long ^{129}Xe spin-lattice relaxation time T_1 which is, for example, several minutes in xenon clathrate hydrate.¹² The application range of ^{129}Xe NMR and MRI further widened when the so-called optical pumping or hyperpolarization technique was invented,²¹ enabling the increase of the ^{129}Xe polarization by up to four or five orders of magnitude and making experiments possible with very small amounts of xenon gas.

The work of Ripmeester and Davidson¹² revealed the potential of ^{129}Xe NMR in studies of clathrates. Namely, the ^{129}Xe NMR spectrum of xenon clathrate hydrate consists of two broad resonance lines: one at 152 ± 2 ppm and the other at 242 ± 2 ppm downfield (smaller shielding) from the low pressure gas peak. The former resonance arises from xenon in large cages, and the latter from xenon in small cages. Thus, the ^{129}Xe chemical shift relates to the size of cages.²² Furthermore, the resonance at 152 ppm displays a cylindrically symmetric powder lineshape with a chemical shift anisotropy of 32 ± 3 ppm; the resonance lineshape relates to the shape of the cage accommodating xenon atoms. The line intensities in turn relate to the occupation number. An illustrative example is the distribution of xenon atoms in the alpha cages of the NaA zeolite.²³ The ^{129}Xe NMR spectrum of xenon in NaA consists of several distinct signals, the chemical shift being determined by the occupation number of a cage.

Modeling of ^{129}Xe NMR in clathrate cages dates back to the studies of chemical shifts²⁴ and NMR line shapes²⁵ inside clathrate hydrate structures I and II, which were modeled as molecular clusters extracted from the clathrate structures. Similar quantum chemical cluster models were also applied in studies of Xe inside fullerene cages,^{26,27} for which both relativistic^{28,29} as well as dynamical and environmental^{29,30} effects were studied. Current cluster modeling makes use of recent advances in the development of relativistic quantum chemistry methods, which have enabled very demanding studies of large heavy-element systems such as cryptophanes³¹ and self-organizing metallo-supramolecular cages.³²

Diffraction is the usual method for crystal structure solution. However, structure determination of clathrates by diffraction methods can be hindered by their instability. For the materials studied here, we are unable to obtain a powder X-ray diffraction (PXRD) pattern from the *m*-fluorophenol clathrate and could only obtain a low resolution diffraction pattern from the *o*-fluorophenol clathrate, from which structure determination would not be possible. Therefore, we cannot combine ^{129}Xe NMR with PXRD experiments as usually done for identifying new clathrate phases.^{33,34} However, the exceptional sensitivity of xenon’s chemical shift anisotropy to its environment should enable the distinction between different candidate clathrate structures. We investigate this hypothesis by comparing experimentally observed and quantum chemically modeled ^{129}Xe NMR isotropic and anisotropic shift parameters in clathrate structures obtained by computational crystal structure prediction (CSP).

CSP has until recently been focused primarily on predicting the single thermodynamically stable structure, or possibly a few low-energy polymorphs. The lowest energy crystal structures available to a given molecule are, in all but very rare exceptions,³⁵ close-packed, leaving no room for the inclusion of guest molecules. There have been few reports of the prediction of porous molecular crystals^{36–38} or solvates.^{39–41} The empty host molecule frameworks of observed inclusion structures have been shown to often exist as local minima on the lattice energy surface, albeit sometimes at relatively high energies compared to close-packed alternative structures.^{41–43} This suggests an efficient approach to the discovery of inclusion structures: searching for stable, empty frameworks using CSP methods and subsequently inserting the guest. This method should be particularly suited to weakly interacting guests such as Xe, where inclusion is expected to leave the host framework relatively unperturbed. The approach should be more efficient than searching the dramatically larger multi-component phase space defined by the host and guest together.

In this study, we have performed CSP calculations for *o*- and *m*-fluorophenol with the specific aim to predict realistic xenon clathrate complexes for which calculated ^{129}Xe NMR parameters can be compared to measured spectra. Advanced first principles density functional theory (DFT) electronic structure calculations of the ^{129}Xe NMR shielding tensors, with a proper inclusion of electron correlation as well as relativistic, periodic, and dynamical effects, have been carried out for a set of predicted structures. A comparison of the simulated isotropic and anisotropic Xe chemical shift parameters with experimentally observed solid state ^{129}Xe NMR data allows us to identify the crystal structure of these elusive clathrates.

2 Experimental section

2.1 Experimental NMR

2.1.1 Preparation of samples

Clathrate samples were prepared from commercially available *o*- and *m*-fluorophenol (Aldrich, 98% assay) and used directly without purification. The substances were transferred into pyrex glass tubes (4 mm outer diameter, 0.8 mm walls) and connected to a volume calibrated vacuum line. Two cycles of freeze-thaw were applied to reduce oxygen content. Isotopically enriched ^{129}Xe gas (Chemgas, 89% enrichment) was then frozen into the evacuated tubes. The amount of transferred gas was controlled by the pressure drop in the vacuum line with a digital pressure gauge (1 hPa precision). The amount of gas inserted into the tubes was chosen to be sufficient for saturation of a 3:1 host-guest clathrate stoichiometry and with excess to pressurize the sample tube so as to maintain clathrate structural stability at the desired experimental temperature of 250 K.^{3,4} The sample glass tubes were flame sealed, cooled in liquid nitrogen and equilibrated for a period of two weeks at *ca* 243 K prior to NMR measurements.

2.1.2 NMR experiments

The ^{129}Xe NMR spectrum of xenon in *m*-fluorophenol was measured at 251 K on a Bruker DSX300WB spectrometer (^{129}Xe Larmor frequency 83.03 MHz) using a 7 mm variable angle spinning (VAS) probe head (DOTY Scientific, Inc., USA), without spinning. The static powder spectrum (the axis of the solenoid coil was set perpendicular to the external magnetic field) was observed applying cross-polarization (CP) and proton decoupling. The following acquisition parameters were used: ^{129}Xe pulse width 5 μs , ^1H decoupling pulse width 16 μs , mixing time 5.4 ms, repetition time 35 s, strength of the ^1H decoupling field 34 kHz, and number of collected free induction decay (FID) signals 220. The aim of applying proton decoupling was to diminish the effect of the ^{129}Xe - ^1H dipolar coupling on the line width. The ^{129}Xe chemical shift was measured relative to the isotopic ^{129}Xe chemical shift in hydroquinone, in which the shift relative to zero-pressure xenon is known to be 222.1 ppm. Temperature calibration was based on the measurement of ^1H chemical shift difference in a separate methanol sample.⁴⁴ Prior to Fourier transformation the FID signal was multiplied by an exponentially decaying apodization function leading to $\simeq 100$ Hz line broadening.

The ^{129}Xe NMR spectrum of xenon in *o*-fluorophenol was in turn measured at 253 K on a Bruker DPX400 spectrometer (^{129}Xe Larmor frequency 110.70 MHz) using a 5 mm broad band observe (BBO) probe head. ^{129}Xe chemical shift is given with respect to the signal of zero-pressure xenon gas, which was determined using two samples with known xenon pressure and extrapolation using the second virial coefficient. Temperature calibration was in this case performed using ^1H chemical shifts in methanol, but with methanol placed in the annulus of a double tube system (outer tube 5 mm, inner tube 4 mm). ^{129}Xe pulse width of 29.5 μs was applied. Prior to Fourier transformation, the FID signal was multiplied by an exponentially decaying apodization function leading to 50 Hz line broadening.

The elements of the chemical shift tensors were determined in both cases using DMFIT.⁴⁵ The uncertainties were estimated to be ± 0.2 ppm, ± 0.4 ppm and < 0.04 , respectively, in the three adjusted NMR parameters: isotropic chemical shift (CS) $\delta \equiv \delta_{\text{iso}} = \sigma_{\text{iso}}^{\text{ref}} - \sigma_{\text{iso}}$ referenced to the zero-pressure limit Xe gas, chemical shift anisotropy $\Delta\delta = \delta_{\text{zz}} - (\delta_{\text{xx}} + \delta_{\text{yy}})/2$ (CSA) with positive/negative values for prolate/oblate

spheroids along the z -direction and asymmetry parameter $\eta = (\delta_{yy} - \delta_{xx})/(\delta_{zz} - \delta_{iso})$ with the value 0 for an axially symmetric tensor (with respect to z) and 1 for the fully asymmetric case, when $\delta_{yy} = \delta_{iso}$. The isotropic Xe nuclear shielding constant is the trace of the shielding tensor, $\sigma_{iso} = (\sigma_{xx} + \sigma_{yy} + \sigma_{zz})/3$. The components of the principal axis system (PAS) of the Xe shift tensor follow Haeberlen's convention:⁴⁶ $|\delta_{zz} - \delta_{iso}| \geq |\delta_{xx} - \delta_{iso}| \geq |\delta_{yy} - \delta_{iso}|$. A relatively large contribution to the uncertainties arises from the wavy background in the experimental spectra, which was therefore eliminated.

2.1.3 Powder X-ray diffraction (PXRD) experiments

Samples of both *o*- and *m*-fluorophenol were prepared in glass capillaries. Xenon gas was frozen into the evacuated capillaries, which were flame-sealed after submerging in liquid N₂. Capillaries containing the samples were kept immersed in liquid N₂ prior to mounting in a pre-cooled single crystal diffractometer. Diffraction data were collected at 100 K using graphite monochromated Mo K α radiation.

2.2 Crystal structure prediction (CSP)

CSP calculations involved five steps: i) the lowest energy conformations of the isolated molecules were identified; ii) hypothetical crystal packings were generated and then lattice energy minimized using a simple force field and rigid-molecule constraints; iii) a subset of the lowest energy predicted crystal structures were re-optimized using an anisotropic atom-atom force field; iv) intramolecular flexibility was introduced, allowing the hydroxyl group to reorient in response to packing forces in each low energy crystal structure; v) porous structures were identified, xenon atoms inserted into the pores and the structures once again re-optimized.

Molecular geometries, energies and charge densities, calculated by DFT using GAUSSIAN 09⁴⁷ with the B3LYP^{48,49} functional and 6-311G(d,p) basis set, were used throughout the CSP calculations. Two stable, planar conformers of each molecule were identified (Fig. 1) and DFT predicts that their energies are sufficiently close that either could form low energy crystal structures.⁵⁰ Therefore, both conformers of both molecules were included in the CSP study.

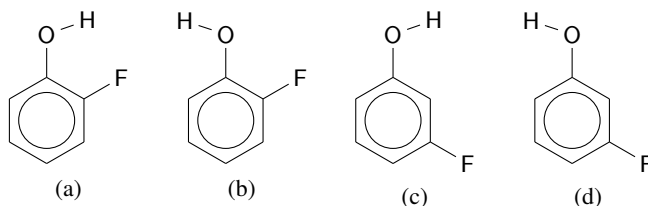


Figure 1: Conformers of the two fluorophenol isomers. a) *cis*- and b) *trans*-*o*-fluorophenol c) *cis*- and d) *trans*-*m*-fluorophenol.

Hypothetical crystal structures were generated with the rigid DFT-optimized molecular geometries using Monte Carlo simulated annealing⁵¹ with MATERIALS STUDIO. Searches were performed in the most commonly observed space groups of known molecular organic crystal structures; 25 space groups were searched with one molecule in the asymmetric unit ($Z' = 1$) and 5 space groups with $Z' = 2$, including all combinations of the two conformers in the asymmetric unit.

Structures with a lattice energy within a 15 kJ/mol window from the lowest energy structure were further refined with an anisotropic, atomic multipole-based intermolecular atom–atom potential model,⁵² combined with a DFT treatment of intramolecular energies and geometries. Duplicate crystal structures were removed using the COMPACT⁵³ program and the resulting unique structures were re-optimized using the CRYSTALOPTIMIZER⁵⁴ program to treat flexibility of the hydroxyl group within each crystal structure. Full details of the conformational analysis and CSP methods are provided in the ESI†.

2.3 Selection of likely clathrate host structures

Lattice energy differences between polymorphs are usually very small and rarely exceed 8 kJ/mol.⁵⁵ However, since voids in crystal structures are thermodynamically unfavourable,^{42,56} structures within a larger energy range than in the usual application of CSP to polymorph prediction were considered as putative inclusion frameworks. We considered crystal structures within an energy cutoff of 13 kJ/mol above the global minimum for each molecule.

To guide our selection of potential clathrate host frameworks, we examined the guest-to-host volume ratio R_g in 31 representative clathrate structures taken from the CSD (see ESI† for details). The volume ratio R_g is calculated as

$$R_g = 100 \cdot \frac{V_g}{V_H} \quad [\%], \quad (1)$$

where V_g is the van der Waals volume of the guest molecule and V_H is an individual void’s contact volume.⁵⁷

One xenon atom was inserted at each cavity’s centroid coordinates and the resulting xenon clathrate structures were geometry-optimized as described above, assuming rigid molecules, imposing no space group symmetry and using an *ad hoc* exp-6 potential for xenon (see ESI†).

2.4 ¹²⁹Xe NMR shielding calculations

The three ¹²⁹Xe NMR parameters of the predicted structures were modeled and compared with the experimental data in several stages. In NMR modeling the chemical shift reference is a free Xe atom. In the first step, all predicted clathrate structures were subjected to nonrelativistic (NR) NMR modeling of cluster models. The results from the NR cluster models were used to identify likely candidates for the experimentally observed clathrate structures. These candidates were then further studied using DFT calculations on their fully periodic models. Finally, a few crystal structures that agreed best with experimental NMR results were chosen for more detailed modeling. Full details are given in the ESI†.

2.4.1 Screening of structures by cluster modeling

The NMR parameters of probable clathrate structures were first modeled using clusters consisting of a single xenon-occupied cavity, including the Xe atom and all nearby fluorophenol molecules. Nonrelativistic Xe shielding tensor calculations were performed on the cluster models using TURBOMOLE.⁵⁸ The BHandHLYP^{59,60} hybrid functional including 50% of exact Hartree-Fock (HF) exchange (EEX) was chosen based on benchmark calculations on the xenon-benzene system, where high quality

nonrelativistic *ab initio* Xe chemical shifts are reproduced reasonably well.²⁹ Likewise, BHandHLYP has been shown to slightly underestimate Xe chemical shift and anisotropy, leaving room for improvements in the modeling by approaching the experimental values from below for Xe-containing molecules,^{61–63} Xe atoms moving freely inside buckminsterfullerenes^{29,30} and in self-organizing metallo-supramolecular cages.³² Those studies show that the typical overestimation of Xe CS and CSA by pure generalized gradient approximation (GGA) DFT functionals, such as PBE⁶⁴ and BLYP,^{59,65} is partly compensated when the amount of exact HF exchange is increased in B3LYP (EEX=20%) and BHandHLYP (EEX=50%) hybrid functionals. Calculations have been performed with a mixed basis set denoted as MHA/SVP (see ESI† for details).

2.4.2 Periodic modeling of likely candidates

Five *o*-fluorophenol and seven *m*-fluorophenols crystal structures with screening level NMR parameters close to the experimental values were chosen for further scrutiny (*vide infra*). These structures were optimized with respect to both atomic positions and lattice parameters with planewave, periodic DFT with the PBE functional and Tkatchenko–Scheffler (TS) dispersion correction,⁶⁶ denoted as PBE-TS structures from now on.

Following optimization, NMR shielding tensors were computed with the PBE functional using the gauge-including projector augmented wave (GIPAW) method^{67,68} (see Figs. 6 and 7). These periodic DFT results include the scalar relativistic (SR) effects on Xe shielding at the 1-component zeroth-order regular approximation (ZORA)^{69,70} level of theory. Details of the periodic calculations with CASTEP^{71,72} are in the ESI†.

2.4.3 Detailed modeling of the most probable structures

After periodic NMR calculations of the DFT optimized structures, the most probable clathrate structures were selected for more detailed DFT NMR modeling using the Amsterdam Density Functional^{73,74} (ADF). In addition to the NR and SR-ZORA quantum chemistry, ADF provides 2-component spin–orbit SO-ZORA method including also SR effects.^{69,75}

As a much wider range of DFT functionals that is currently available in periodic CASTEP can be used in ADF calculations performed with the jcpl/TZP mixed basis set⁷⁶ (see ESI† for details) on *clusters* comprising a single cavity, the influence of correlation treatment and the amount of EEX on NMR parameters was tested using PBE, BLYP, B3LYP, and BHandHLYP functionals at SR-ZORA level. By using the results in scaling of the *periodic* PBE results, we obtain estimates of SR periodic BLYP, B3LYP and BhandHLYP results. We expect that, as the electron correlation description is improved by an increasing portion of EEX, the calculated NMR parameters for the true crystal structures should approach the experimental results. Hence, the BHandHLYP-scaled periodic PBE results provide the best estimation of ¹²⁹Xe NMR parameters and lack only contributions due to molecular dynamics and relativistic SO effects. The latter we treat as an additive correction obtained from the difference of the static SR- and SO-ZORA cluster calculations with the BHandHLYP functional.

The effect of Xe dynamics at $T = 300$ K for ¹²⁹Xe NMR shielding parameters was modeled for the few most probable clathrates by canonical NVT Metropolis Monte Carlo (MC-NVT) of Xe motion on a potential energy surface inside a fixed cluster cavity with fixed PBE-TS optimized geometry. The temperature effects on Xe chemical

shift and anisotropy were calculated as the difference between the thermally averaged shielding tensor and the reference tensor with Xe at the center of the cage.

3 Results and discussion

3.1 Experimental NMR results

Both isomers formed clathrates with xenon under the experimental conditions and allowed recording of properly shaped powder patterns without active mixing and crushing of crystals during their formation. A powder-like appearance and lack of macroscopic crystallites has been confirmed by visual inspection and fit to spectra. There is no preferential growth imposed by tube walls as this would not allow for proper random averaging of orientations of crystallites in relation to the magnetic field of the spectrometer.

Axially symmetric powder patterns have been observed for both clathrates (Figure 2), which confirms the existence of highly symmetric voids occupied by xenon atoms. High values of ^{129}Xe chemical shift for both clathrates, $\delta = 228.5$ ppm for *m*-fluorophenol and $\delta = 256.0$ ppm for *o*-fluorophenol, suggest tight cages with only one occupational site. In such cases, where ^{129}Xe NMR powder patterns are expected to reflect only interactions of xenon atoms with the immediately adjacent host molecules, the influence of neighbouring cages with xenon is expected to be negligible.⁷⁷

The spectrum of *m*-fluorophenol (Figure 2) resembles that of β -hydroquinone,⁸ β -phenol and *p*-fluorophenol with one notable difference; it has the largest observed value of ^{129}Xe NMR CSA among known inclusion compounds: $\Delta\delta = 183.3$ ppm vs. 161.9, 171 and 164 ppm for hydroquinone,⁷⁸ phenol⁵ and *p*-fluorophenol,⁷⁹ respectively. Small additional features in the observed spectrum of *m*-fluorophenol are attributed to amorphous host substance and xenon atoms adsorbed in this amorphous phase.

The ^{129}Xe NMR of *o*-fluorophenol shows negative CSA (Figure 2), $\Delta\delta = -47.5$ ppm, in contrast to all previously known solid state ^{129}Xe NMR powder patterns of phenol clathrates for which positive CSA is observed.^{5–10} Changes in the sign of the CSA can be observed in a group of porous channel-like dipeptides,⁸⁰ where CSA changes from positive to negative with increasing xenon gas pressure and therefore with pronounced $\text{Xe}\cdots\text{Xe}$ interactions playing a greater role. No changes of the NMR spectrum of *o*-fluorophenol clathrate have been observed when repeating the experiment with samples of different pressures. The small negative CSA may result from an axially symmetric but oblate environment around the xenon atom in this clathrate.

3.2 Powder X-ray diffraction results

A PXRD pattern could not be obtained from *m*-fluorophenol clathrate, which transformed too quickly to the known high density form when transferred to the diffractometer. This is confirmed by comparison of the measured PXRD pattern to that simulated from the known crystal structure. For *o*-fluorophenol, we were able to obtain a broad powder pattern that does not correspond to either known crystal structures (see ESI†). The pattern is too broad to index, so it could not be used to determine the structure.

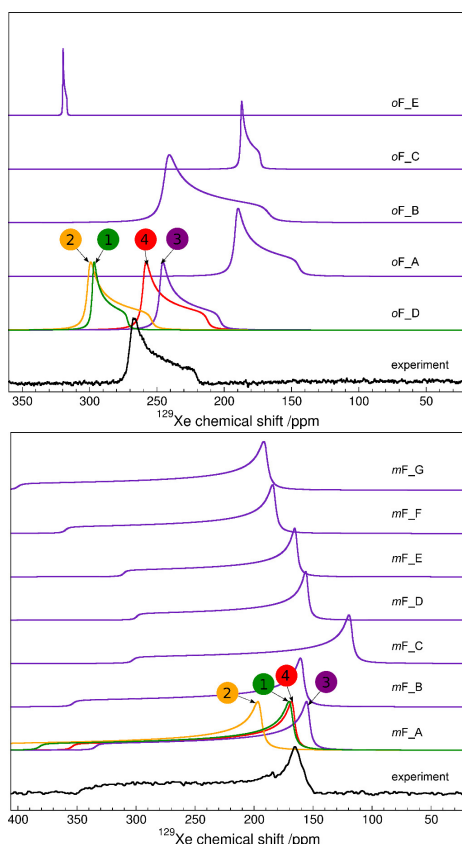


Figure 2: Experimental (black) and calculated (indigo) ^{129}Xe NMR powder spectra of *o*-fluorophenol (top) and *m*-fluorophenol (bottom) with ^{129}Xe gas reference at zero. Calculated spectra for all structures were obtained from BHandHLYP-scaled periodic PBE NMR parameters at PBE-TS optimized geometry (method/structure = BHandHLYP/PBE-TS). For the most probable candidates, *o*F_D and *m*F_A, results from different periodic modeling levels are displayed: (1, green) PBE/CSP, (2, orange) PBE/PBE-TS, (3, indigo) BHandHLYP/PBE-TS and (4, red) BHandHLYP/PBE-TS with effects due to Xe dynamics at $T = 300\text{ K}$ (see text for details).

3.3 Crystal structure prediction results

CSP resulted in an exceptionally large number of low energy crystal structures for each molecule (Figures 3 and 4).

For *o*-fluorophenol, $Z' = 2$ crystal structures with both molecules in the *cis*-conformation are energetically favoured. The known⁸¹ high-pressure polymorph II (CSD refcode QAMWEH01) is located in the search, 0.8 kJ/mol above the global minimum (Figure 3) and with a slightly higher density, as expected for a high pressure polymorph. The known low-temperature polymorph I⁸¹ (CSD refcode QAMWEH) is a disordered $Z' = 1.5$ structure and could not be predicted with the methods employed here.

One crystal structure of *m*-fluorophenol is known⁸¹ (CSD refcode QAMTUU) and corresponds to the global lattice energy minimum from the search. The structure is geometrically well reproduced (with a deviation in atomic positions in a 15-molecule

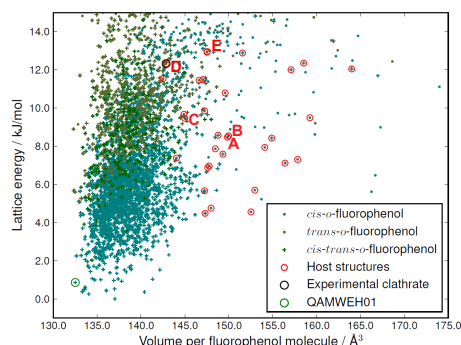


Figure 3: The crystal energy landscape of *o*-fluorophenol. Each point represents one crystal structure and is coloured according to the molecular conformation (*cis-trans-o*-fluorophenol refers to $Z'=2$ structures containing both conformers). Structures with a lattice energy within 13 kJ/mol from the lowest energy structure and having cavities of suitable size for xenon absorption (see text) are encircled in red. The experimentally known high-pressure polymorph (CSD refcode QAMWEH01) is encircled in green. The labels A–E correspond to structures *oF*_A to *oF*_E in the text.

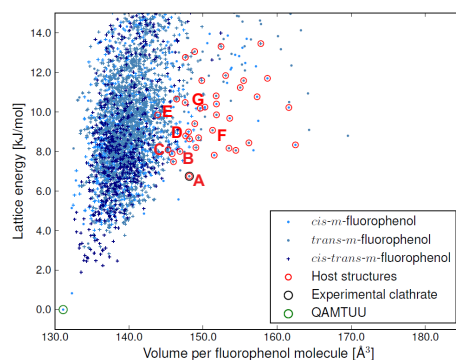


Figure 4: The crystal energy landscape of *m*-fluorophenol. Each point represents one crystal structure and is coloured according to the molecular conformation (*cis-trans-m*-fluorophenol refers to $Z'=2$ structures containing both conformers). Structures with a lattice energy within 13 kJ/mol from the lowest energy structure and having cavities of suitable size for xenon absorption (see text) are encircled in red. The experimentally known stable polymorph (CSD refcode QAMTUU) is encircled in green. The labels A–G correspond to structures *mF*_A to *mF*_G in the text.

cluster taken from predicted and X-ray crystal structures of $\text{RMSD}_{15} = 0.203 \text{ \AA}$.

Overlays of both matches are included in the electronic supporting information (ESI[†]). The successful reproduction of the known crystal structures of both molecules amongst the lowest energy predicted structures provides confidence in the sampling of crystal structures and of their lattice energy rankings.

3.4 Selection rules for clathrate structures

Since the inclusion of guest molecules in clathrates can significantly stabilize the structure, using the lattice energy of the empty host alone is not useful for selection of promising structures. Our analysis of known clathrates found that the guest-to-host volume ratio R_g is normally distributed with mean 59% and a standard deviation of 8 percentage units, in good agreement with Rebek's "55% solution"⁸² and providing further evidence for the empirical rule^{82,83} that R_g in observed inclusion structures should fall in this limited range.

Of the predicted crystal structures within 13 kJ/mol of the global minimum for *o*-fluorophenol and *m*-fluorophenol, 230 and 223 are porous (having cavities $> 10 \text{ \AA}^3$). 33 (*o*-fluorophenol) and 32 (*m*-fluorophenol) of the predicted structures have cavities of a suitable volume to accommodate xenon as a guest, which we took as those with R_g within three standard deviations ($\pm 3\sigma$) for observed clathrates. These structures are encircled in red in Figs 3 and 4. The lowest energy of these structures are 4.5 and 6.7 kJ/mol above the global minima for *o*- and *m*-fluorophenol, respectively.

3.5 Computational NMR results

3.5.1 Screening of structures by cluster models

Examples of cluster models used for initial screening are shown in Fig. 5. As can be seen in Figures 6 and 7, even for a set of structures with similar void volumes and packing energies, the computed NMR tensor parameters for ^{129}Xe vary widely in our initial screening using cluster models of the predicted structures. This demonstrates the exceptional sensitivity of the ^{129}Xe NMR parameters to small differences in cavity size and shape.

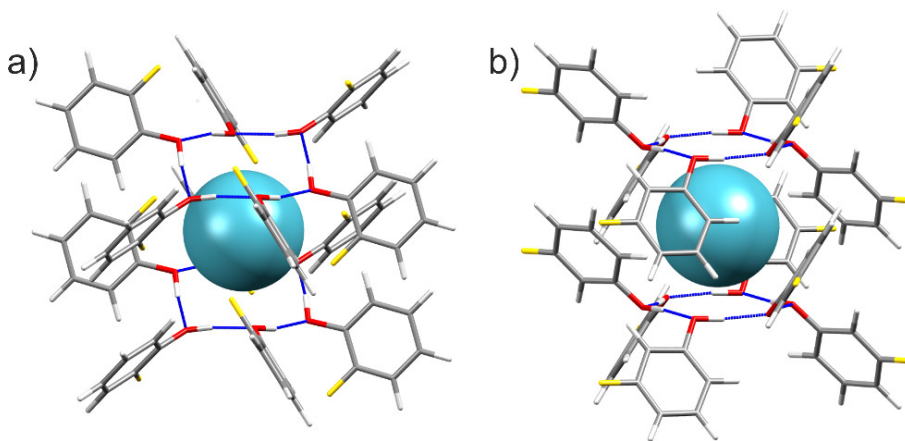


Figure 5: Examples of clusters used in the initial screening NMR calculations of the predicted clathrate structures. The clusters shown correspond to the structures selected as best models of the observed clathrates: a) *o*-fluorophenol structure *oF_D*, b) *m*-fluorophenol structure *mF_A*. Atoms are colored by element: carbon (grey); oxygen (red); hydrogen (white); fluorine (yellow).

Calculated NMR parameters for the 33 *o*- and 32 *m*-fluorophenol clathrate structures obtained with nonrelativistic DFT cluster models are tabulated in the ESI†, see

Tables S4 and S5. Many of these structures, including all predicted $Z' = 2$ structures, have chemical shift tensors with significant asymmetry; these voids are clearly not compatible with the experimental spectra, in which $\eta < 0.04$. At this stage, only the structures with axially symmetric chemical shift tensors (green squares in Figures 6 and 7) are kept as possible candidates for the experimental structures.

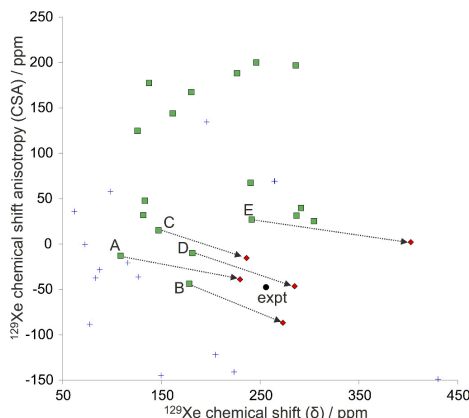


Figure 6: ^{129}Xe NMR parameters computed at screening (nonrelativistic DFT/BHandHLYP/MHA) level for cluster models of the predicted clathrate structures of *o*-fluorophenol. Structures with asymmetry $\eta > 0.04$ are shown as blue crosses, while structures with symmetric chemical shift tensors ($\eta \leq 0.04$) are shown as green squares. The experimental data (expt, black filled circle) is included for comparison. Structures that were considered for further study are connected with the black arrows to the periodic GIPAW/PBE calculation of Xe shielding from periodic PBE-TS optimized structures of the corresponding crystal (red diamonds). The labels A–E correspond to structures *oF_A* to *oF_E* in the text.

Optimisation of the clathrate structure after insertion of the xenon into the host allowed the structures to relax in response to guest insertion. Despite very small structural changes, this significantly affected calculated NMR parameters, see Figure S7 in the ESI†.

Since the screening level of theory is expected to underestimate both CS and CSA by a few tens of ppm, we focused on structures with both properties of the correct sign and underestimated, also including a few structures with overestimated CS for *m*-fluorophenol and slightly positive CSA for *o*-fluorophenol for higher level calculations. This excluded all but five *o*-fluorophenol and seven *m*-fluorophenol structures; hereafter, we will refer to these as structures *oF_A* to *oF_E* (in order of increasing energy, Fig. 3) and *mF_A* to *mF_G* (Fig. 4). All of these candidate clathrate structures have space group symmetry $R\bar{3}$.

3.5.2 NMR chemical shift parameters from periodic DFT calculations

NMR shielding tensors were calculated by periodic DFT for the five plausible *o*-fluorophenol and seven *m*-fluorophenol clathrate structures. We found that periodic modeling is necessary and that the geometry relaxation with periodic PBE-TS shifts the CS and CSA towards larger magnitudes, corresponding to more elliptic, prolate ($\Delta\delta > 0$) and oblate ($\Delta\delta < 0$) *m*- and *o*-fluorophenol cavities, respectively.

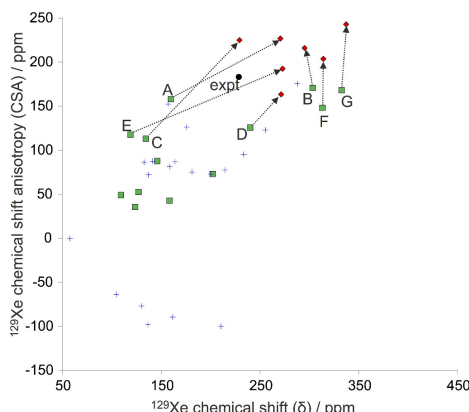


Figure 7: ^{129}Xe NMR parameters computed at screening (nonrelativistic DFT/BHandHLYP/MHA) level for cluster models of the predicted clathrate structures of *m*-fluorophenol. Structures with asymmetry $\eta > 0.04$ are shown as blue crosses, while structures with symmetric chemical shift tensors ($\eta \leq 0.04$) are shown as green squares. The experimental data (expt, black filled circle) is included for comparison. Structures that were considered for further study are connected with the black arrows to the periodic GIPAW/PBE calculation of Xe shielding from periodic PBE-TS optimized structures of the corresponding crystal (diamonds). The labels A–G correspond to structures *mF*_A to *mF*_G in the text.

NMR parameters from SR GIPAW/PBE calculations at the PBE-TS optimized geometries are tabulated in Tables S6 and S7 in ESI† as well as displayed in Figures 6 and 7 as red diamonds connected with arrows to the corresponding screening cluster result (green squares). Here, we recall that the periodic results set upper limits on ^{129}Xe CS and CSA, as the PBE functional (and all pure GGA DFT functionals) overestimates the magnitudes of both parameters.^{29,32,61–63} Although the disagreement in CS and CSA of some of the candidate structures now makes them unlikely (such as *oF*_C, *oF*_E and *mF*_G), all were kept for further analysis with more detailed NMR calculations.

3.5.3 Results of detailed NMR calculation of the most likely candidates

Typically, the crystal lattice effect, *i.e.* the difference of ^{129}Xe NMR parameters between the periodic and cluster models of the same PBE-TS optimized clathrate, increases both CS and CSA as seen in Tables 1 and 2. However, the extent of the change is very case-specific and unforeseeable, which makes periodic modeling essential, in one form or another, in the present search of a best candidate of an unknown clathrate structure.

The ADF/SR-ZORA results for PBE-TS optimized cluster models were used to extract correction factors (see details in ESI†) by which the periodic, SR GIPAW/PBE results were scaled in order to obtain periodic estimates of ^{129}Xe NMR parameters with BLYP, B3LYP and BHandHLYP functionals. The estimated periodic data for the five and seven candidates of *o*-fluorophenol and *m*-fluorophenol clathrates are listed in Tables 1 and 2 as well as displayed in Figures 8 and 9, respectively. The best BHandHLYP-scaled periodic estimates bring one clathrate structure for each of the fluorophenol isomers into excellent agreement with the experimental NMR: *oF*_D and *mF*_A. For *o*-fluorophenol, the other CSP candidates in the PBE-TS crystal geometry have quite different periodic BHandHLYP ^{129}Xe CS and/or CSA values as compared

to the experimental ones, while for *m*-fluorophenol the closest alternative candidates overestimate either the CS (*mF_F*) or CSA (*mF_B*). We expect the BHandHLYP functional to provide reasonable approximation for both quantities due to benchmarking against ab initio calculations.⁶¹⁻⁶³

Table 1: Theoretically modelled ^{129}Xe NMR chemical shift (δ) and anisotropy (CSA, $\Delta\delta$) in ppm for the five most likely *o*-fluorophenol CSP candidates in their PBE-TS optimised structures, unless stated otherwise.

Method	oF_A		oF_B		oF_C		oF_D		oF_E	
	δ	$\Delta\delta$	δ	$\Delta\delta$	δ	$\Delta\delta$	δ	$\Delta\delta$	δ	$\Delta\delta$
<i>Cluster model</i> ^a										
PBE	218.7	-14.3	245.0	-79.3	215.1	4.6	261.5	-38.6	370.5	7.7
BLYP	226.5	-14.6	253.4	-81.8	222.5	4.1	270.9	-40.2	381.5	7.9
B3LYP	195.9	-23.1	223.1	-75.6	195.9	0.9	240.8	-36.6	338.1	13.1
BHandHLYP	165.8	-21.3	190.7	-68.3	162.6	-1.2	209.9	-35.8	288.5	18.3
BHandHLYP ^b	145.8	-20.5	175.0	-61.8	143.2	0.9	189.4	-31.6	262.0	14.4
BHandHLYP ^c	174.9	-11.6	192.3	-71.2	171.3	6.7	216.7	-31.6	292.0	31.1
<i>Periodic model</i> ^d										
PBE ^e	195.1	-28.1	272.1	-66.4	238.7	7.0	288.8	-26.2	362.1	23.1
PBE	229.8	-38.9	273.2	-86.7	236.4	-15.4	285.0	-46.5	402.7	2.0
BLYP ^f	237.3	-39.1	281.5	-89.3	243.6	-15.9	294.3	-48.1	413.6	2.1
B3LYP ^f	206.3	-48.0	250.8	-82.9	216.7	-19.3	263.8	-44.5	369.5	7.5
BHandHLYP ^f	175.6	-46.3	217.7	-75.4	182.6	-21.5	232.3	-43.6	318.8	12.7
BHandHLYP ^g			236.2	-82.4			243.7	-46.8		
BHandHLYP ^h			237.8	-85.4			250.5	-42.6		
Experimental							256.0	-47.5		

^a Scalar relativistic SR-ZORA calculations of a cluster model of one clathrate cavity with ADF code^{73,74} using Xe/other=jepl/TZP basis sets. ⁷⁶ ^{129}Xe chemical shift ($\delta = \sigma_{\text{Xe-atom}} - \sigma$) with respect to free Xe atom shielding values ($\sigma_{\text{Xe-atom}}$): PBE/BLYP/B3LYP/BHandHLYP = 5752.2/5752.6/5752.1/5752.0 ppm. $\Delta\delta$ (CSA) defined in principal axis system (PAS) along the unique axis (perpendicular to the plane of O-H rings). Asymmetry parameter $\eta = 0$ in all cavities due to the cylindrical symmetry.

^b As footnote *a* but at nonrelativistic (NR) level of theory with $\sigma_{\text{Xe-atom}}(\text{BHandHLYP}) = 5643.4$ ppm.

^c As footnote *a* but at relativistic spin-orbit SO-ZORA level of theory with $\sigma_{\text{Xe-atom}}(\text{BHandHLYP}) = 6609.3$ ppm.

^d Scalar relativistic GIPAW results obtained with CASTEP code.^{71,72} All periodic ^{129}Xe chemical shifts referenced to CASTEP/PBE value: $\sigma_{\text{Xe-atom}} = 5926.3$ ppm.

^e In the CSP optimised crystal geometry.

^f As footnote *d* but estimated by scaling GIPAW/PBE PAS components with factors obtained at SR-ZORA level for cluster model resulting in data in footnote *a*.

^g Correction added to result of footnote *f* due to thermal averaging (AVE) over Xe motion at $T = 300\text{K}$ by MC-NVT simulation (see text for details).

^h Relativistic spin-orbit (SO) corrections obtained as difference of SO-ZORA and SR-ZORA calculations with ADF code added to result of footnote *g*.

Table 2: Theoretically modelled ^{129}Xe NMR chemical shift (δ) and anisotropy (CSA, $\Delta\delta$) in ppm for the seven most likely m -fluorophenol CSP candidates in their PBE-TS optimised structures, unless stated otherwise.

Method	mF_A		mF_B		mF_C		mF_D		mF_E		mF_F		mF_G	
	δ	$\Delta\delta$	δ	$\Delta\delta$	δ	$\Delta\delta$	δ	$\Delta\delta$	δ	$\Delta\delta$	δ	$\Delta\delta$	δ	$\Delta\delta$
<i>Cluster model</i> ^a														
PBE	259.6	219.9	272.1	188.5	216.5	198.5	256.3	146.6	256.7	187.1	297.6	187.9	316.1	220.4
BLYP	264.8	222.0	278.7	189.5	224.1	201.3	262.7	147.5	262.0	189.1	305.4	189.3	323.3	222.1
B3LYP	236.9	197.2	240.0	172.1	193.6	182.4	225.6	135.5	233.0	167.1	265.1	173.8	282.1	204.0
BHandHLYP	204.6	174.7	202.4	166.9	167.5	159.2	189.8	127.8	198.4	141.9	227.2	161.4	240.9	188.1
BHandHLYP ^b	182.3	165.8	189.8	159.5	152.2	147.7	176.7	122.6	177.1	135.1	210.6	153.5	223.9	179.9
BHandHLYP ^c	211.2	176.3	197.4	162.1	170.1	156.2	186.4	124.6	204.7	139.0	224.0	157.0	237.2	182.1
<i>Periodic model</i> ^d														
PBE ^e	239.7	213.1	428.8	226.8	198.7	162.1	343.8	166.0	193.2	156.9	436.1	194.5	467.0	225.7
PBE	270.6	226.7	295.2	216.0	229.2	224.8	271.4	163.4	272.9	192.4	314.1	203.7	337.2	242.9
BLYP ^f	275.5	228.9	301.6	217.0	236.7	227.6	277.7	164.3	277.9	194.4	321.7	205.1	344.2	244.5
B3LYP ^f	247.3	203.4	262.3	199.3	205.7	208.3	240.0	152.0	248.6	171.8	280.7	189.4	302.3	226.1
BHandHLYP ^f	214.2	180.2	223.8	194.1	179.1	184.5	203.3	144.2	213.1	145.9	241.9	176.7	260.1	209.9
BHandHLYP ^g	229.3	187.3	229.7	200.4					227.4	152.3	250.2	191.2		
BHandHLYP ^h	235.9	188.9	224.7	195.6					233.8	149.4	247.0	186.7		
Experimental	228.5	183.3												

^a Scalar relativistic SR-ZORA calculations of a cluster model of one clathrate cavity with ADF code^{73,74} using Xe/other=jepl/TZP basis sets. ⁷⁶ ^{129}Xe chemical shift ($\delta = \sigma_{\text{Xe-atom}} - \sigma$) with respect to free Xe atom shielding values ($\sigma_{\text{Xe-atom}}$): PBE/BLYP/B3LYP/BHandHLYP = 5752.2/5752.6/5752.1/5752.0 ppm. $\Delta\delta$ (CSA) defined in principal axis system (PAS) along the unique axis (perpendicular to the plane of O-H rings). Asymmetry parameter $\eta = 0$ in all cavities due to the cylindrical symmetry.

^b As footnote *a* but at nonrelativistic (NR) level of theory with $\sigma_{\text{Xe-atom}}(\text{BHandHLYP}) = 5643.4$ ppm.

^c As footnote *a* but at relativistic spin-orbit SO-ZORA level of theory with $\sigma_{\text{Xe-atom}}(\text{BHandHLYP}) = 6609.3$ ppm.

^d Scalar relativistic GIPAW results obtained with CASTEP code.^{71,72} All periodic ^{129}Xe chemical shifts referenced to CASTEP/PBE value: $\sigma_{\text{Xe-atom}} = 5926.3$ ppm.

^e In the CSP optimised crystal geometry.

^f As footnote *d* but estimated by scaling GIPAW/PBE PAS components with factors obtained at SR-ZORA level for cluster model resulting in data in footnote *a*.

^g Correction added to result of footnote *f* due to thermal averaging (AVE) over Xe motion at $T = 300\text{K}$ by MC-NVT simulation (see text for details).

^h Relativistic spin-orbit (SO) corrections obtained as difference of SO-ZORA and SR-ZORA calculations with ADF code added to result of footnote *g*.

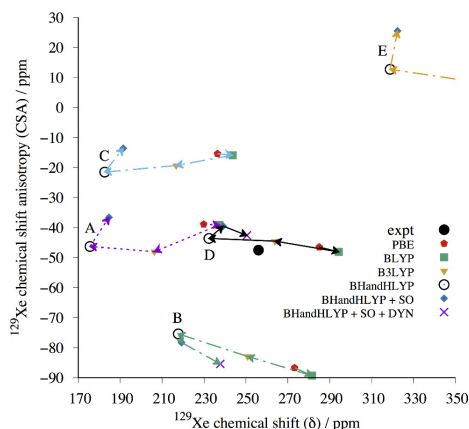


Figure 8: The periodic GIPAW results for the five most likely *o*-fluorophenol structures optimized at the periodic PBE-TS level of theory. The labels A–E correspond to structures *oF_A* to *oF_E* in the text. We expect the correct structure to approach the experimental (expt) ^{129}Xe NMR parameters, when the computed GIPAW/PBE result is scaled with factors obtained using different pure and hybrid DFT functionals with increasing amount of exact exchange in the series of PBE→BLYP(0%)→B3LYP(20%)→BHandHLYP(50%). The SO correction is added to BHandHLYP values of all structures (blue diamonds). For structures *oF_B* and *oF_D*, the final points (indigo crosses) include also the effect of Xe dynamics (DYN) at $T = 300\text{ K}$.

In addition, the relativistic SO correction, obtained as the difference between SO- and SR-ZORA cluster calculations with ADF (see Tables 1 and 2), is added to the static BHandHLYP results of all structures (Figures 8 and 9). Due to its different physical origin, the SO effect is case-specific and may either increase or decrease the CS and CSA, while the latter is affected slightly more. The magnitude of the SO correction is, however, smaller than the dynamical correction (see below) and, hence, does not alter the identification of the best clathrate candidates.

The temperature effect of Xe dynamics at $T = 300\text{ K}$ was modeled for the two (*oF_B* and *oF_D*) and four (*mF_A*, *mF_B*, *mF_E*, and *mF_E*) most relevant clathrate candidates. The dynamical (DYN) correction is added to the SO corrected BHandHLYP results (BHandHLYP + SO). In all cases, the thermal averaging increases the magnitudes of both CS and CSA of ^{129}Xe . As seen in Figures 8 and 9, the inclusion of DYN correction confirms the most probable clathrate structure of both isomers. The effect on the simulated spectra is also shown in Figure 2.

It is evident from Figures 8 and 9 (see also Tables S6 and S7 in ESI†) that in all probability the crystal structures for *o*- and *m*-fluorophenol clathrates are *oF_D* and *mF_A*. For them, the ^{129}Xe NMR isotropic chemical shift and chemical shift anisotropies at different levels of theory encompass the experimental data. For both structures, the experimental NMR parameters are approached as the computational level is improved. The approximated periodic BHandHLYP results at PBE-TS optimized structures only slightly underestimate the chemical shift, leaving room for improvements by dynamics and relativistic SO corrections. Astonishingly, the most accurate level of theoretical modeling almost quantitatively reproduces the two NMR parameters that are clearly specific for a given clathrate. The remaining differences between

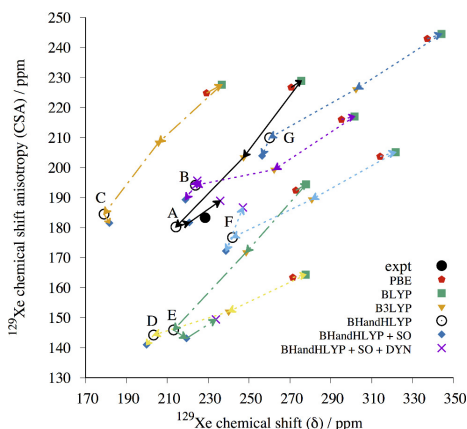


Figure 9: The periodic GIPAW results for the seven *m*-fluorophenol structures optimized at the periodic PBE-TS level of theory. The labels A–G correspond to structures *mF_A* to *mF_G* in the text. We expect the correct structure to approach the experimental (expt) ^{129}Xe NMR parameters, when the computed GIPAW/PBE result is scaled with factors obtained using different pure and hybrid DFT functionals with increasing amount of exact exchange in the series of $\text{PBE} \rightarrow \text{BLYP}(0\%) \rightarrow \text{B3LYP}(20\%) \rightarrow \text{BHandHLYP}(50\%)$. The SO correction is added to BHandHLYP values of all structures (blue diamonds). For structures *mF_A*, *mF_B*, *mF_E*, and *mF_F*, the final point (indigo crosses) include also the effect of Xe dynamics (DYN) at $T = 300$ K.

experimental and computational data may be attributed to deficiencies in the structure as well as treatments of electron correlation and thermal averaging of the whole system.

Our PXRD results provide further evidence for the CSP- ^{129}Xe NMR determined structure of the *o*-fluorophenol clathrate; the simulated diffraction pattern from *oF_D* is similar to the PXRD obtained from the clathrate sample (see Figure S13 in ESI†). However, we cannot use PXRD of the present quality to unambiguously distinguish between the candidates structures.

3.6 Description of the proposed clathrate structures

The crystal structures that we propose as the fluorophenol xenon clathrates belong to space group $R\bar{3}$, with lattice parameters shown in Table 3. Both structures have 3-fold screw axes parallel to the *c* lattice vector, resulting in 6-membered rings of host molecules held together with strong hydrogen bonds formed by practically ideal $\text{OH} \cdots \text{O}$ interactions. The hydrogen bonding forms $R_6^6(12)$ graphs,⁸⁴ which seems to be a characteristic feature in clathrates of phenol derivatives. Packing diagrams are displayed in Figures 10 and 11. CIF files are included in the ESI†. Both structures have three cavities per unit cell, with volumes 61.4 and 88.5 Å³ each for *o*- and *m*-fluorophenol respectively, resulting in R_g values of 68.7 and 47.7% for xenon, both within 1.5 standard deviations from the ideal ratio of 59%.

Despite the same hydrogen bonding and crystal symmetry, the voids in the two structures are of quite different geometry. The void in *oF_D* has an oblate shape with ratios of the principal moments of the free volume (calculated from a grid sampling in PLATON⁸⁵) of 1.00:1.00:0.85, with the short dimension oriented along the crystallo-

Table 3: Unit cell parameters (after PBE-TS geometry optimisation) for the proposed xenon clathrate structures of *o*- and *m*-fluorophenol

Structure	space group	a	b	c	γ
<i>o</i> F_D	$R\bar{3}$	22.5225 Å	22.5225 Å	5.6879 Å	120°
<i>m</i> F_A	$R\bar{3}$	22.5526 Å	22.5526 Å	5.6998 Å	120°

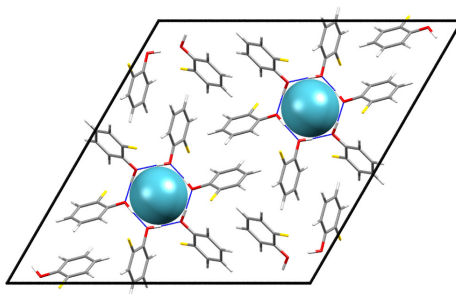


Figure 10: Packing diagram of the PBE-TS optimized *o*-fluorophenol xenon clathrate structure *o*F_D viewed down the **c**-axis. Space group $R\bar{3}$ and with $R_6^6(12)$ hydrogen bonds encasing the xenon atoms. See also Figure 5a.

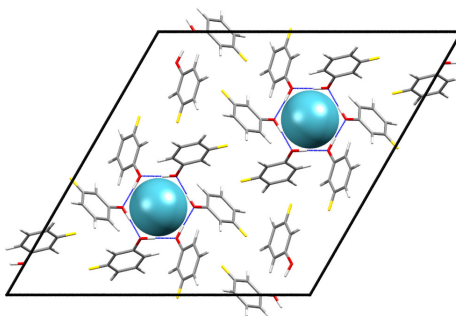


Figure 11: Packing diagram of the PBE-TS optimized *m*-fluorophenol xenon clathrate *m*F_A viewed down the **c**-axis. Space group $R\bar{3}$ and with $R_6^6(12)$ hydrogen bonds encasing the xenon atoms. See also Figure 5b.

graphic **c**-axis. In contrast, *m*F_A has a prolate shape with calculated ratio of dimensions of 1.00:1.00:1.20, elongated along the crystallographic **c**-axis. Thus, we find that the sign of the CSA relates to the shape of the void, as expected, with an oblate cavity leading to a negative CSA and prolate cavity yielding a positive CSA.

The proposed structure of the *o*-fluorophenol Xe clathrate (*o*F_D) is one of the highest energy and one of the densest structures that we considered as possible clathrates (Figure 3). The stabilisation of this structure must derive from the interactions of Xe with the host structure, which should be large, due to the tight fit of Xe to the host cavities.

In contrast, the most likely structure of the *m*-fluorophenol Xe clathrate (*m*F_A) corresponds to the most stable structure on the crystal structure landscape (Figure 4) that contains cavities suitable for Xe enclathration; this structure's stability relates, in part, to the stability of the host framework.

4 Conclusions

We have used detailed first principles NMR calculations on candidates from CSP to propose structures for the xenon clathrates of *o*- and *m*-fluorophenol. The exceptional sensitivity of the ^{129}Xe chemical shift tensor to its local environment allows comparisons between observed and calculated NMR chemical shift parameters that have been used to directly confirm or reject hypothetical clathrate structures. Based on these comparisons, we propose likely crystal structures for the two clathrates. The proposed structures strongly resemble a previously known β -hydroquinone xenon clathrate⁶ and have similar $R_6^6(12)$ hydrogen bonding motifs.⁸⁴

The unusual *o*-fluorophenol ^{129}Xe NMR powder spectrum – with its negative CSA – was initially thought to suggest a structural motif different from the known and common $R_6^6(12)$ hydrogen-bonded double sandwich. Our results, however, confirm that the $R_6^6(12)$ motif is an important feature for clathrates of phenol derivatives that is present in both materials studied here.

Static solid-state NMR has clear benefits over high-frequency magic-angle spinning NMR, that would only result in isotropic chemical shifts rather than the complete chemical shift tensors. In this study, using isotropic shifts only would not have allowed us to identify the experimental clathrate structures among the predicted candidates.

We believe that the method presented here is a powerful approach for structure determination of porous materials, which is particularly useful in cases where powder diffraction patterns either cannot be obtained, or are insufficient for structure determination. The success of the method is related to the high sensitivity of the Xe NMR chemical shift and chemical shift anisotropy to minute details of the cavity geometry, such that even structurally very similar clathrates can have vastly different chemical shift tensors. However, this sensitivity also requires highly detailed NMR calculations; in addition to the proper treatment of electron correlation and relativistic phenomena the inclusion of explicit crystal lattice effects as well as Xe dynamics was necessary in order to precisely reproduce experimental NMR values. Developments in this area, along with progress in CSP algorithms, open up new possibilities for the prediction and characterization of new porous materials by combining structure prediction with computational and experimental ^{129}Xe NMR spectroscopy.

5 Acknowledgements

The work reported here has been financially supported by the European Research Council under the European Union’s Seventh Framework Programme (FP/2007-2013)/ERC Grant Agreement no. 307358, ERC-stG-2012-ANGLE (JN, GD), by the Magnus Ehrnrooth Foundation and NGS-NANO (MS), and by the Academy of Finland project nos. 125316, 218191, 255641, and 285666 (PL). Computational resources for the NMR part of the article were partially provided by CSC – IT Center for Science Ltd, Espoo and Kajaani, Finland. Parts of the calculations have been carried out using resources provided by Wrocław Centre for Networking and Supercomputing (<http://wcss.pl>). The crystal structure prediction was carried out using the IRIDIS high performance computing facility at the University of Southampton. MS acknowledges great help of Anu M. Kantola in performing solid state NMR measurements.

References

- [1] M. Baías, J.-N. Dumez, P. H. Svensson, S. Schantz, G. M. Day and L. Emsley, *J. Am. Chem. Soc.*, 2013, **135**, 17501–17507.
- [2] M. Baías, C. M. Widdifield, J.-N. Dumez, H. P. G. Thompson, T. G. Cooper, E. Salager, S. Bassil, R. S. Stein, A. Lesage, G. M. Day and L. Emsley, *Phys. Chem. Chem. Phys.*, 2013, **15**, 8069–8080.
- [3] Y. N. Kazankin, A. A. Palladiev and A. M. Trofimov, *Zh. Obshch. Khim.*, 1972, **42**, 2368–70.
- [4] Y. N. Kazankin, A. A. Palladiev and A. M. Trofimov, *Zh. Obshch. Khim.*, 1973, **43**, 2670–1.
- [5] J. A. Ripmeester, *J. Am. Chem. Soc.*, 1982, **104**, 289–290.
- [6] T. Birchall, C. S. Frampton, G. J. Schrobilgen and J. Valsdóttir, *Acta Crystallogr. C*, 1989, **45**, 944–946.
- [7] L. Mandelcorn, *Chem. Rev.*, 1959, **59**, 827–839.
- [8] M. Ilczyszyn, M. Selent and M. M. Ilczyszyn, *J. Phys. Chem. A*, 2012, **116**, 3206–3214.
- [9] S.-P. Kang and H. Lee, *J. Chem. Eng. Data*, 1997, **42**, 467–469.
- [10] R. M. Barrer and V. H. Shanson, *J. Chem. Soc. Faraday Trans.*, 1976, **72**, 2348–2354.
- [11] W. Proctor and F. Yu, *Phys. Rev.*, 1951, **81**, 20–30.
- [12] J. Ripmeester and D. Davidson, *J. Mol. Struct.*, 1981, **75**, 67 – 72.
- [13] T. Ito and J. Fraissard, Proceedings of the 5th International Conference on Zeolites, 1981, p. 510.
- [14] B. Bartik, P. Choquet, A. Constantinesco, G. Duhamel, J. Fraissard, J. N. Hyacinthe, G. E. Pavlovskaya, J. Jokisaari, E. Locci, T. J. Lowery, M. Luhmer, T. Meersmann, I. L. Moudrakovski, K. L. Pierce, A. Pines, J. A. Ripmeester, V.-V. Telkki and W. S. Weeman, *Actual Chimique*, 2005, **287**, 16.
- [15] D. Raftery, in *Xenon NMR Spectroscopy*, ed. G. Webb, Academic Press, 2006, vol. 57, pp. 205 – 270.
- [16] M. M. Spence, S. M. Rubin, I. E. Dimitrov, E. J. Ruiz, D. E. Wemmer, A. Pines, S. Q. Yao, F. Tian and P. G. Schultz, *Proc. Natl. Acad. Sci.*, 2001, **98**, 10654–10657.
- [17] J. P. Mugler, T. A. Altes, I. C. Ruset, I. M. Dregely, J. F. Mata, G. W. Miller, S. Ketel, J. Ketel, F. W. Hersman and K. Ruppert, *Proc. Natl. Acad. Sci.*, 2010, **107**, 21707–21712.
- [18] D. W. Davidson and J. A. Ripmeester, in *Inclusion Compounds*, ed. J. L. Atwood, J. E. D. Davies and D. D. MacNicol, Academic Press: London, 1984, vol. 3, pp. 69–128.

- [19] C. Dybowski, N. Bansal and D. M. Duncan, in *Annual Review of Physical Chemistry*, Annual Reviews, 1991, vol. 42, pp. 433–464.
- [20] C. I. Ratcliffe, in *Annual Report on NMR Spectroscopy*, Elsevier, 1998, vol. 36, pp. 123–221.
- [21] T. G. Walker and W. Happer, *Reviews of Modern Physics*, 1997, **69**, 629.
- [22] J. Ripmeester, C. Ratcliffe and J. Tse, *J. Chem. Soc. Faraday Trans.*, 1988, **84**, 3731–3745.
- [23] C. J. Jameson, A. K. Jameson, B. I. Baello and H. Lim, *J. Chem. Phys.*, 1994, **100**, 5965–5976.
- [24] D. Stueber and C. J. Jameson, *J. Chem. Phys.*, 2004, **120**, 1560–1571.
- [25] C. J. Jameson and D. Stueber, *J. Chem. Phys.*, 2004, **120**, 10200–10214.
- [26] M. Bühl, S. Patchkovskii and W. Thiel, *Chem. Phys. Lett.*, 1997, **275**, 14 – 18.
- [27] D. N. Sears and C. J. Jameson, *J. Chem. Phys.*, 2003, **118**, 9987–9989.
- [28] J. Autschbach and E. Zurek, *J. Phys. Chem. A*, 2003, **107**, 4967–4972.
- [29] M. Straka, P. Lantto and J. Vaara, *J. Phys. Chem. A*, 2008, **112**, 2658–2668.
- [30] S. Standara, P. Kulhánek, R. Marek and M. Straka, *J. Comput. Chem.*, 2013, **34**, 1890–1898.
- [31] A. Bagno and G. Saielli, *Chem. Eur. J.*, 2012, **18**, 7341–7345.
- [32] J. Roukala, J. Zhu, C. Giri, K. Rissanen, P. Lantto and V.-V. Telkki, *J. Am. Chem. Soc.*, 2015, **137**, 2464–2467.
- [33] J. A. Ripmeester, J. S. Tse, C. I. Ratcliffe and B. M. Powell, *Nature*, 1987, **325**, 135–136.
- [34] L. Yang, C. A. Tulk, D. D. Klug, I. L. Moudrakovski, C. I. Ratcliffe, J. A. Ripmeester, B. C. Chakoumakos, L. Ehm, C. D. Martin and J. B. Parise, *Proceedings of the National Academy of Sciences*, 2009, **106**, 6060–6064.
- [35] J. L. Flippen, J. Karle and I. L. Karle, *J. Am. Chem. Soc.*, 1970, **92**, 3749–3755.
- [36] J. T. A. Jones, T. Hasell, X. Wu, J. Bacsá, K. E. Jelfs, M. Schmidtman, S. Y. Chong, D. J. Adams, A. Trewin, F. Schiffman, F. Cora, B. Slater, A. Steiner, G. M. Day and A. I. Cooper, *Nature*, 2011, **474**, 367–371.
- [37] E. O. Pyzer-Knapp, H. P. G. Thompson, F. Schiffmann, K. E. Jelfs, S. Y. Chong, M. A. Little, A. I. Cooper and G. M. Day, *Chem. Sci.*, 2014, **5**, 2235–2245.
- [38] A. G. Slater, M. A. Little, A. Pulido, S. Y. Chong, L. Chen, C. Morgan, X. Wu, G. Cheng, R. Clowes, M. E. Briggs, T. Hasell, K. E. Jelfs, G. M. Day and A. I. Cooper, *Nat. Chem.*, 2016.
- [39] A. J. Cruz-Cabeza, G. M. Day and W. Jones, *Chem. Eur. J.*, 2008, **14**, 8830–8836.
- [40] A. J. Cruz-Cabeza, S. Karki, L. Fabian, T. Friscic, G. M. Day and W. Jones, *Chem. Commun.*, 2010, **46**, 2224–2226.

- [41] D. E. Braun, T. Gelbrich, V. Kahlenberg and U. J. Griesser, *CrystEngComm*, 2015, **17**, 2504–2516.
- [42] A. J. Cruz-Cabeza, G. M. Day and W. Jones, *Chem. Eur. J.*, 2009, **15**, 13033–13040.
- [43] A. J. Cruz Cabeza, G. M. Day, W. D. S. Motherwell and W. Jones, *J. Am. Chem. Soc.*, 2006, **128**, 14466–14467.
- [44] A. Webb, *Annu. Rep. NMR Spectro.*, 2002, **45**, 1–67.
- [45] D. Massiot, F. Fayon, M. Capron, I. King, S. Le Calvé, B. Alonso, J.-O. Durand, B. Bujoli, Z. Gan and G. Hoatson, *Magnetic Resonance in Chemistry*, 2002, **40**, 70–76.
- [46] U. Haeberlen, *High Resolution NMR in Solids Selective Averaging: Supplement 1 Advances in Magnetic Resonance*, Elsevier, 2012, vol. 1.
- [47] M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria, M. A. Robb, J. R. Cheeseman, G. Scalmani, V. Barone, B. Mennucci, G. A. Petersson, H. Nakatsuji, M. Caricato, X. Li, H. P. Hratchian, A. F. Izmaylov, J. Bloino, G. Zheng, J. L. Sonnenberg, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, T. Vreven, J. A. Montgomery, Jr., J. E. Peralta, F. Ogliaro, M. Bearpark, J. J. Heyd, E. Brothers, K. N. Kudin, V. N. Staroverov, R. Kobayashi, J. Normand, K. Raghavachari, A. Rendell, J. C. Burant, S. S. Iyengar, J. Tomasi, M. Cossi, N. Rega, J. M. Millam, M. Klene, J. E. Knox, J. B. Cross, V. Bakken, C. Adamo, J. Jaramillo, R. Gomperts, R. E. Stratmann, O. Yazyev, A. J. Austin, R. Cammi, C. Pomelli, J. W. Ochterski, R. L. Martin, K. Morokuma, V. G. Zakrzewski, G. A. Voth, P. Salvador, J. J. Dannenberg, S. Dapprich, A. D. Daniels, Ö. Farkas, J. B. Foresman, J. V. Ortiz, J. Cioslowski and D. J. Fox, *Gaussian 09 Revision D.01*, Gaussian Inc. Wallingford CT 2009.
- [48] A. Becke, *J. Chem. Phys.*, 1993, **98**, 5648.
- [49] P. J. Stephens, F. J. Devlin, C. F. Chabalowski and M. J. Frisch, *J. Phys. Chem.*, 1994, **98**, 11623.
- [50] H. P. Thompson and G. M. Day, *Chem. Sci.*, 2014, **5**, 3173–3182.
- [51] H. Karfunkel and R. Gdanitz, *J. Comput. Chem.*, 1992, **13**, 1171–1183.
- [52] E. O. Pyzer-Knapp, H. P. G. Thompson and G. M. Day, *Acta Crystallogr. B*, 2016, **72**, 477–487.
- [53] J. A. Chisholm and S. Motherwell, *J. Appl. Crystallogr.*, 2005, **38**, 228–231.
- [54] A. V. Kazantsev, P. G. Karamertzanis, C. S. Adjiman and C. C. Pantelides, *J. Chem. Theory Comput.*, 2011, **7**, 1998–2016.
- [55] J. Nyman and G. M. Day, *CrystEngComm*, 2015, **17**, 5154–5165.
- [56] A. Kitaigorodsky, *Molecular Crystals and Molecules*, Elsevier, 2012, vol. 29.
- [57] L. J. Barbour, *Chem. Commun.*, 2006, 1163–1168.

- [58] *TURBOMOLE V6.6 2014, a development of University of Karlsruhe and Forschungszentrum Karlsruhe GmbH, 1989-2007, TURBOMOLE GmbH, since 2007; available from*
<http://www.turbomole.com>.
- [59] A. D. Becke, *Phys. Rev. A*, 1988, **38**, 3098.
- [60] A. D. Becke, *J. Chem. Phys.*, 1993, **98**, 1378.
- [61] P. Lantto and J. Vaara, *J. Chem. Phys.*, 2007, **127**, 084312.
- [62] M. Straka, P. Lantto, M. Rasanen and J. Vaara, *J. Chem. Phys.*, 2007, **127**, 234314.
- [63] P. Lantto, S. Standara, S. Riedel, J. Vaara and M. Straka, *Phys. Chem. Chem. Phys.*, 2012, **14**, 10944–10952.
- [64] J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865.
- [65] C. Lee, W. Yang and R. Parr, *Phys. Rev. B*, 1988, **37**, 785.
- [66] A. Tkatchenko and M. Scheffler, *Phys. Rev. Lett.*, 2009, **102**, 073005.
- [67] C. Pickard and F. Mauri, *Phys. Rev. B*, 2001, **63**, 245101.
- [68] J. Yates, C. Pickard and F. Mauri, *Phys. Rev. B*, 2007, **76**, 024401.
- [69] E. van Lenthe, E. J. Baerends and J. G. Snijders, *J. Chem. Phys.*, 1993, **99**, 4597.
- [70] S. K. Wolff, T. Ziegler, E. van Lenthe and E. J. Baerends, *J. Chem. Phys.*, 1999, **110**, 7689–7698.
- [71] S. J. Clark, M. D. Segall, C. J. Pickard, P. J. Hasnip, M. J. Probert, K. Refson and M. Payne, *Z. Kristall.*, 2005, **220**, 567–570.
- [72] E. R. McNellis, J. Meyer and K. Reuter, *Phys. Rev. B*, 2009, **80**, 205414.
- [73] G. t. Velde, F. M. Bickelhaupt, E. J. Baerends, C. Fonseca Guerra, S. J. A. van Gisbergen, J. G. Snijders and T. Ziegler, *J. Comput. Chem.*, 2001, **22**, 931.
- [74] ADF2016, SCM, Theoretical Chemistry, Vrije Universiteit, Amsterdam, The Netherlands, <http://www.scm.com>.
- [75] E. van Lenthe, E. J. Baerends and J. G. Snijders, *J. Chem. Phys.*, 1994, **101**, 9783.
- [76] E. van Lenthe and E. J. Baerends, *J. Comput. Chem.*, 2003, **24**, 1142.
- [77] C. J. Jameson and A. C. de Dios, *J. Chem. Phys.*, 2002, **116**, 3805–3821.
- [78] I. Moudrakovski, C. Ratcliffe and J. Ripmeester, *J. Am. Chem. Soc.*, 2001, **123**, 2066–2067.
- [79] D. W. Davidson, S. K. Garg, S. R. Gough, Y. P. Handa, C. I. Ratcliffe, J. S. Tse and J. A. Ripmeester, in *Clathrate Compounds, Molecular Inclusion Phenomena, and Cyclodextrins: Proceedings of the Third International Symposium on Clathrate Compounds and Molecular Inclusion Phenomena and the Second International Symposium on Cyclodextrins, Tokyo, Japan, July 23-27, 1984*, ed. J. L. Atwood, J. E. Davis and T. Osa, Springer Science & Business Media, 2012, pp. 231–238.

- [80] D. V. Soldatov, I. L. Moudrakovski, E. V. Grachev and J. A. Ripmeester, *J. Am. Chem. Soc.*, 2006, **128**, 6737–6744.
- [81] I. D. H. Oswald, D. R. Allan, W. D. S. Motherwell and S. Parsons, *Acta Crystallogr. B*, 2005, **61**, 69–79.
- [82] S. Mecozzi and J. Rebek, Jr., *Chem. Eur. J.*, 1998, **4**, 1016–1022.
- [83] K. Nakano, K. Sada, Y. Kurozumi and M. Miyata, *Chem. Eur. J.*, 2001, **7**, 209–220.
- [84] M. C. Etter, *J. Phys. Chem.*, 1990, **95**, 4601–4610.
- [85] A. L. Spek, *J. Appl. Crystallogr.*, 2003, **36**, 7–13.