

# Novel Ductile FRP System for Concrete Reinforcement: Concept and Experimental Characterization

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## Abstract

This paper presents a novel design concept for fiber reinforced polymer (FRP) composites consisting of three-dimensional (3D) printed cores and FRP helical skins as a means of ensuring adequate ductility, compared to the brittle FRP systems conventionally used for internal reinforcement. The experiment demonstrated that when the FRP skins were loaded in tension, the core—which was 3D printed using acrylonitrile butadiene styrene or polylactic acid—was gradually compressed, thereby leading to plastic deformation. This behavior ensured a nonlinear load response while eliminating the unfavorable brittle failure of the FRPs. The results also indicated that the proposed FRP composite system ensured that no premature debonding/delamination occurred between the skin–skin and skin–core. The results of the parametric experimental study indicated that design parameters such as the FRP amount, core height, core span, core shell thickness, core material, core brace, and core number (i.e., the number of inner cores used for the composite) may be optimized to realize the expected design load capacity and ductility.

Keywords: 3D-print; ABS core; PLA core; Ductility; FRP; Nonlinear; Strength

## Introduction

29

30 Corrosion of steel in reinforced concrete causes concrete cracking, loss of bond strength, reduction in the  
31 steel cross section, and loss of serviceability (Cabrera 1996). It has been reported that the corrosion of steel  
32 in reinforced concrete (RC) requires over \$8 billion annually for repairing RC bridges in the United States  
33 (Behnam and Eamon 2013; US Federal Highway Administration 2001). Although the corrosion of steel in  
34 RC may be treated by improving the concrete mix design, increasing the thickness of the concrete cover  
35 (Faustino et al. 2015), and employing cathodic protection and epoxy-coating methods, such methods fail to  
36 completely eliminate the corrosion.

37 Over the last few decades, the use of FRP as reinforcement in concrete members has gained interest among  
38 the researchers and designers owing to the corrosion resistance, high strength and low-weight characteristics  
39 of the materials. (Achintha et al. 2018; Achintha M 2009; Lou et al. 2016, 2017a; b; Lou and Karavasilis  
40 2018; Sun et al. 2017a, 2018; Sun 2018; Sun et al. 2016, 2017b; Sun and Ghannoum 2015). Despite FRPs  
41 being more expensive than steel on a unit weight basis, it is anticipated that the innovative use of the material  
42 together with its long-term benefits such as low maintenance and high durability may enable FRP  
43 reinforcement systems to be a viable alternative to steel reinforcement. Nevertheless, the brittle failure and  
44 premature debonding of the material when used as reinforcement in concrete has limited the more widespread  
45 use of FRP reinforcement in concrete. Although the provision of anchorages could prevent debonding failures  
46 in some applications, the ductility of the FRP reinforced concrete systems needed to be improved. The use of  
47 stainless steel fibers was noted to enhance the ductility (Allaer et al. 2014); however, the relatively high  
48 density of steel led to limitations, in particular, in lightweight applications.

49 Thin-ply hybrid laminates consisting of high (e.g., GFRP) and low (e.g., CFRP) strain materials have been  
50 developed for safety-critical applications such as motor-sports, aerospace and pressure vessels. Such

51 laminates can realize pseudoductility by stably pulling the low strain material out of the composite (Czél et  
52 al. 2017; Czél and Wisnom 2013; Jalalvand et al. 2015). However, the ductile behavior significantly depends  
53 on the interfacial bond (Jalalvand et al. 2015), which might be degraded under the effects of UV light (Zhai  
54 et al. 2016) and/or heat (Mohan 2013) in the field. Other approaches involved providing fibers with excess  
55 length for developing further extensions. Possible solutions to produce such excess lengths involved  
56 fabricating fabrics with diagonally oriented fibers (Grace et al. 2004) or corrugated fibers (Yokozeki et al.  
57 2006), which could be re-oriented or unfolded to produce further extensions in the process of adapting to the  
58 loading direction. However, these fabrics required adequate matrices to resist the fiber rotation; otherwise,  
59 significant deformations could be developed on the initiation of loading, thereby torpedoing the candidacy of  
60 the fabrics to replace steel reinforcements.

61 More sophisticated approaches to ensure adequate strength and deformation involved the use of composites  
62 shaped as stiff skins on the wavy surface of soft cores (Pimenta & Robinson, 2014; Quon et al., 2013;  
63 Winkelmann et al., 2010). Notable extensions have been achieved by unfolding these shaped skins via  
64 premature hardening responses (Pimenta and Robinson 2014) or reloading processes (Quon et al. 2013;  
65 Winkelmann et al. 2010). Moreover, the core was generally used to shape the composite profile, and its  
66 contributions to the composite behavior has not been fully explored.

67 To summarize, the current approaches—which involve limitations such as notably increased weight,  
68 UV/heat degradation, possible instability on the initiation of loading, unstably reloading responses, and  
69 unfavorable hardening responses—are incapable of being an attractive replacement for steel reinforcements.  
70 This study aims to overcome these limitations by proposing an innovative concept for a low-density, high-  
71 yield-strength, large-deformation and stable-loading-behavior composite consisting of FRP helical skins and  
72 three-dimensional (3D) print cores. The skin and core materials were carefully selected to achieve lightweight

73 and high corrosion-resistance composites. By using the helically braiding technique proposed for this  
74 composite, the skins were expected to tightly attach on the core surface with minimum UV/heat impacts. It  
75 is expected that the proposed composite would be particularly suitable for use in

76 (1) internal reinforcements, taking advantage of the high-yield strength, large-deformation and stable-  
77 loading-behavior potentials of the composites.

78 (2) near-surface mounted reinforcements because of the lightweight and noncorrosion characteristics of  
79 the composites, allowing for easy installation to externally strengthen the concrete elements.

## 80 **Proposed FRP composites: FRP helical skin and 3D-printed core**

81 FRP helical skins and 3D-printed cores are developed for the proposed composites to achieve the desired  
82 composite stress–strain relations. 3D-printed cores are used to shape the FRP skins. The helical structures  
83 ensure a tight skin–core bond by the twisting of skins under tensile loading (Ling 2002). The twisted skins  
84 are expected to effectively resist the skin–core delamination and the opening generated due to the core  
85 stiffness producing significant stress concentrations at the core edges. Subsequently, various cores with the  
86 corresponding stiffness and deformability achieved by 3D printing technology (Dalaq et al. 2016; Wang et  
87 al. 2011) can be applied for the composites. Compressing these cores tends to unfold the shaped skins to the  
88 amount corresponding to the elastic or plastic core deformations, thereby developing stable nonlinear tensile  
89 responses with a considerable stress and strain at skin fracture. As shown in Fig. 1, the initial loads produced  
90 slight core deformations that unfolded a limited amount of shaped skins, resulting in stiff composite  
91 responses. Subsequently, notable composite extensions could be developed through further compression of  
92 the inner cores producing plastic core deformations. To the best of the authors' knowledge, this could be a  
93 pioneering study in the use of inner cores to control composite behaviors. The available composites (Pimenta

94 & Robinson, 2014; Quon et al., 2013; Winkelmann et al., 2010) relying exclusively on epoxy bond or stitches  
95 are incapable of resisting the opening due to the initial stiffness of the soft cores. Thus, 3D-printed cores with  
96 various stiffness have been rarely used to achieve tensile-behavior-designable composites. The use of FRP  
97 helical skins and 3D-printed cores can help realize stable nonlinear responses with a considerable strength  
98 and extension at skin fracture. It is expected that the proposed composite has potential applications as internal  
99 and near-surface mounted reinforcements in concrete structures, owing to its high-strength, large-  
100 deformation, low-density, noncorrosion, and tensile-behavior-designable properties.

## 101 **Concept and mechanical behavior of the composite**

102 The composite relied on three components (i.e., FRP helical skins, 3D-printed cores and bridges) to develop  
103 the nonlinear responses. The FRP helical skins were the main elements that carried the tensile loads (see Fig.  
104 2 (a)). The helical system was expected to effectively resist skin–core delamination by twisting the skins  
105 around the core under tensile loading (see Fig. 2 (a)). By using epoxy resin, the profile of the helical skins  
106 was shaped by the inner core, as shown in Fig. 2 (b). To prevent FRP skins being cut off by sharp cores,  
107 circular-arc cores were applied to define the profiles. The bridges were the linking regions consisting of inner  
108 columns with a diameter of  $d_b$  and helical skins (see Fig. 2 (a) and Fig. 3 (a)). Compressing the core  
109 gradually unfolded the shaped skins to develop the corresponding composite extensions to elastic/plastic core  
110 deformations through the entire loading process, as shown in Fig. 1. By altering the core configurations, the  
111 composite was expected to develop various stress–strain relations under tensile loading. The core  
112 configurations included the shell thickness  $t_s$ , brace thickness  $t_b$ , core height  $h_c$ , core span  $l_c$  and brace  
113 angle  $\theta_b$ , as shown in Fig. 3 (a). The cores consisted of outer shells with or without inner braces, as shown  
114 in Fig. 3 (b)–(c). Braces were used to improve the strength and/or stiffness of the inner cores, thereby leading

115 to stiffer stress–strain responses. Enlarging the brace angles was expected to improve the deformability of the  
116 brace-reinforced cores, as shown in Fig. 3 (c)–(d). It should be noted that one-core composites were first  
117 constructed to efficiently isolate the impacts of core configurations. Next, the composites with multiple cores  
118 were prepared to explore the possibility of extending the composite by using multiple cores. As shown in Fig.  
119 3 (e), these cores were connected by columns having a diameter of  $d_b$  and a length of  $l_b$ .

120 The FRP composite extension resulted from the unfolded length of the skin and the skin elongation.  
121 Compared to the unfolded length, the contribution of the skin elongation to the FRP composite extension was  
122 limited. The FRP composite therefore was expected to develop nonlinear stress–strain responses through  
123 gradually compression of the inner cores to allow an unfolding corresponding to the elastic/plastic core  
124 deformation, as shown in Fig. 1. In the elastic stage, the inner cores were slightly deformed, allowing limited-  
125 shaped skins to be unfolded, thereby resulting in stiff composite responses. Notable composite extensions  
126 resulted from the plastic core deformations through further compression of the inner cores.

127 The proposed concept was validated with experimental results, as discussed in the following sections.  
128 Based on the experimental results, this study involved the investigation of the impacts of core configurations  
129 and composite materials (i.e., FRP amounts and core materials) on the composite behavior.

## 130 **Materials**

131 A unidirectional and dry Tyfo® SCH 11-UP strip (Fyfe Co.LLC 2015) was used to make the helical skins.  
132 This material has been widely used for strengthening and repairing concrete structures owing to its low weight  
133 and easy-installation properties (Sun et al. 2016). The manufacturer-provided density  $\rho_f$  and thickness  $t_f$   
134 were  $1.8 \text{ g/cm}^3$  and  $0.51 \text{ mm}$ , respectively. The selected material was a typical FRP strip with linear-elastic  
135 behavior, although it required improvement in terms of its deformability and energy dissipation. Acrylonitrile

136 butadiene styrene (ABS) and polylactic acid (PLA) having considerable stiffness and deformability were used  
137 for the inner cores to prevent large core deformations at the initiation of loading and to develop considerable  
138 ultimate extensions. For the ABS material, the manufacturer-provided density  $\rho_p$ , modulus  $E_p$ , strength  $f_p$   
139 and ultimate strain  $\varepsilon_p$  of the core material were 1 g/cm<sup>3</sup>, 1.95 GPa, 41 MPa and 0.21, respectively. The  
140 manufacturer-provided PLA properties were  $\rho_p=1.2$  g/cm<sup>3</sup>,  $E_p=2.50$  GPa,  $f_p=63$  MPa and  $\varepsilon_p=0.04$ .

## 141 **Fabrication**

142 The fabrication process of the proposed composites tended to be simple and uniform, resulting in minimum  
143 difference among nominally identical composites. Based on this fabrication concept, the composites were  
144 fabricated by the following steps, as shown in Fig. 4:

- 145 i. The cores and the inner columns were printed using the JG AURORA A8 3D printer and selected core  
146 materials (i.e., either ABS or PLA materials). The printer could print structures with a layer thickness  
147 of 0.3 mm and accuracy of 0.1 mm. As shown in Fig. 4 (a), the 3D printing technology provided a  
148 feasible, robust and formwork-free method for producing the inner cores having various  
149 configurations. The impacts of core configurations (i.e.,  $h_c$ ,  $l_c$ ,  $\theta_b$ ,  $t_s$ , and  $t_b$ ) could be effectively  
150 explored, as discussed in the following sections.
- 151 ii. The cores and inner columns for all specimens were helically wrapped with three strands of FRP  
152 material evenly and carefully separated from a given width of FRP strips having negligible FRP  
153 material loss during the fabrication, as shown in Fig. 4 (b). The three-stranded structure was inspired  
154 by fishing ropes in which three-stranded polyethylene ropes are helically wound together to resist large  
155 axial loading. In this procedure, the FRP material was clipped at one end and the inner system was

156 clipped at both ends. The FRP skins were tightly and helically wrapped around the inner cores and  
157 columns by rotating the three FRP strands at the other end, as shown in Fig. 4 (c).

158 iii. The wrapped composite was saturated with epoxy resin, which was the material provided by the  
159 manufacturer for the installation of the selected FRP strip (Fyfe Co.LLC 2015). The wrapped  
160 composite was held manually and fully saturated in the epoxy for 2 min, as shown in Fig. 4 (d). Next,  
161 the ends of the saturated composite were clipped by three sticks for strengthening the tensile ends with  
162 additional FRP material (Fig. 4 (e)).

163 iv. The tensile ends were strengthened using FRP strips. FRP strips with dimensions of 50×50 mm were  
164 prepared for the tensile ends. These FRP strips were saturated with epoxy before wrapping both the  
165 ends. The 50-mm-wide strips were selected to ensure a good bond by overwrapping the ends by at  
166 least five laps, resulting in more than 6-mm-diameter sections for the tensile ends (see Fig. 4 (e)).  
167 Next, the completed composites were cured for 72 hours at 60°C before testing, according to the  
168 manufacturer's advice (Fyfe Co.LLC 2015).

## 169 Experimental program

170 To validate the proposed design concept, tensile tests were conducted. Specimens were clipped using the  
171 clamps of the displacement-controlled machine for tensile testing (DNS100), in which they were loaded at a  
172 rate of 2 mm/min as shown in Fig. 5. The specimens were expected to fail in the form of FRP skin fracture  
173 when the tensile strength of the FRP skins was reached, or in the form of the inner core crush caused by a  
174 considerable core deformation leading to the core failure before skin fracture. A sudden core crush is expected  
175 to fracture the skin as a simultaneous result. The nominal tensile stress  $f_{com}$  was therefore calculated as

$$176 \quad f_{com} = P/(w_f t_f) \quad (1)$$



177 where

178  $P$  = the applied load measured by the load cell of the testing machine, N.

179  $w_f$  = the measured width of the FRP strip used to make the helical skins, mm.

180  $t_f$  = the nominal thickness (0.51 mm) of the FRP strip.

181 The nominal strains were obtained from the relative displacements of two points at the composite ends.

182 The composite ends represented the last points on the axis of the composite, and they were next to the tensile

183 ends, as shown in Fig. 2 (a). The displacements were measured using a high-resolution digital image

184 correlation (DIC) system, as shown in Fig. 5. The nominal stress and strain were used to demonstrate the

185 tensile behavior of the proposed composite. Moreover, the energy dissipation was obtained from integrating

186 the applied loads with respect to the corresponding displacements between two composite ends.

187 The impacts of FRP amount, core height, core span, shell thickness, core material, core brace and core

188 number were investigated in this study. As the main tensile element, the amount of FRP material determined

189 the ultimate load of the composite and the ultimate deformation of the inner cores. Changes in the core

190 materials or configurations were expected to lead to various core deformations for unfolding the

191 corresponding amounts of shaped skins under tensile loading. Moreover, the composite was also expected to

192 be extendable owing to the use of several cores. To validate those proposed concepts, six groups (A, B, C, D,

193 E and F) of specimens were manufactured as per the details listed in Table 1. Group A consisted of three sets

194 of specimens with identical core configurations wrapped by FRP strands obtained from 10-, 15- and 20-mm-

195 wide FRP strips. This group was aimed to demonstrate the impacts of the FRP amount. Group B had another

196 three sets of specimens to reveal the impact of  $h_c/l_c$ . All the core parameters except  $h_c/l_c$  were kept

197 constant. Various values of  $h_c/l_c$  were obtained from varying core heights  $h_c$  (from 2–5 mm) over a

198 constant core span  $l_c$  (=16 mm). According to the findings obtained from Group A, a reasonable proportion

199 of the FRP amount (in terms of the strip width to the perimeter of the core section at the middle span) was  
200 applied to evenly distribute the FRP material on the core surface. Two more sets of tests were performed for  
201 Group C to investigate the impacts of shell thickness  $t_s$  (from 0.5–2 mm). All parameters except  $t_s$  were  
202 kept constant. Group D was aimed to explore the impacts of the core materials. In this group, the PLA material  
203 was used to print the same core configurations wrapped with an amount of FRP material identical to that of  
204 the corresponding ABS composite. As shown in Fig. 3 (c)-(d), inner braces were applied to strengthen the  
205 inner cores in Group E. These braces were solid elements having a given thickness  $t_b$ , and they provided the  
206 inner cores with support at the middle section. Moreover, the brace could be angled with respect to the vertical  
207 axis. Changing the brace angle  $\theta_b$  was expected to develop various compressive responses. To investigate  
208 the angle impacts, the total thickness of the brace was kept constant (2 mm). The cores were either supported  
209 by one 2-mm-thick brace with an angle  $\theta_b$  of  $0^\circ$  or two 1-mm-thick braces with angles  $\theta_b$  of  $60^\circ$ . Group  
210 F was used to demonstrate the performance of the composite consisting of several cores, which can be a good  
211 reference for extending the proposed composite to a desired length. Another five FRP coupons were also  
212 tested; these specimens were 240 mm long and 15 mm wide FRP strips with 40 mm rectangular FRP end-  
213 tabs. The labeling system applied to identify the specimens involved the group number-set number-unique  
214 test number.

## 215 **Results**

216 The section describes the experimental results obtained from sixty-five tests to validate the proposed  
217 concept and identify the key parameters.

### Typical test behavior

218

219 Five FRP coupons were first tested to obtain the ultimate stresses and strains of the FRP material. Based  
220 on the average values, the strength  $f_{fu}$  and ultimate strain  $\varepsilon_{fu}$  of the FRP material were 1043 MPa and  
221 0.011, respectively. The tensile modulus  $E_f$  obtained from the strength and ultimate strain was 94.55 GPa.

222 Two major failure modes were observed for the proposed composites, i.e., skin fracture when the tensile  
223 strength of the FRP material was reached; and core crush caused by a considerable core deformation resulting  
224 in the fracture of skins. It should be noted that in this study, every core crush simultaneously resulted in skin  
225 fracture (See Fig. 6). The inner cores were designed to resist bending moment. After core crush, the bending  
226 moment was exclusively introduced on FRP skins, simultaneously resulting in the cracking of resin matrices  
227 and skin fracture. No other failures were observed. Specimen A-ii-5 and C-ii-3 were selected as the typical  
228 test specimens. All parameters except for the shell thickness were kept constant for these two specimens (see  
229 Table 1). The shell thickness of Specimen A-ii-5 and C-ii-3 were 1 mm and 0.5 mm, respectively. Fig. 6  
230 shows the stress–strain responses of the selected tests. It can be seen that both the composites developed  
231 notably larger deformations than those of the typical test coupons. Further, improving the deformability tends  
232 to deteriorate the elastic modulus and ultimate strength. As shown in Fig. 6, both the composites failed to  
233 achieve an elastic modulus and ultimate strength comparable to those of the typical coupon tests.

234 The skin fracture for Specimen A-ii-5 occurred at a larger ultimate stress (737 MPa) than the stress at  
235 which Specimen C-ii-3 failed in core crush (at 622 MPa). Although the specimens failed at 71% and 60% of  
236 the FRP strength, both of them successfully developed considerable ultimate strains (0.036 and 0.056)  
237 representing 3.3 and 5.1 times the ultimate strain of the FRP material. This demonstrated that the proposed  
238 composites were able to develop notable deformations by compromising their tensile strengths.

239 It should to be noted that both the stress–strain curves had almost linear and stiff responses prior to 180  
240 MPa, suggesting that linear-elastic core responses occurred, resulting in slight core deformations. The tensile  
241 moduli obtained at 180 MPa were 61.18 GPa and 51.36 GPa for Specimens A-ii-5 and C-ii-3, respectively;  
242 that is, the selected specimens achieved approximately 65% and 54% of the elastic moduli obtained from the  
243 FRP coupon tests. Next, both the stress–strain curves experienced a gradual softening due to the occurrence  
244 of increasingly plastic core deformations allowing more shaped skins to be unfolded. After softening, both  
245 the tests produced softer curves in which the test using the thicker core shell performed almost linearly prior  
246 to skin fracture, and the other test experienced the second softening at approximately 550 MPa before core  
247 crush occurred. These observations demonstrated that the proposed composite could exhibit nonlinear tensile  
248 behavior by producing plastic core deformations. The composites were expected to fail in skin fracture,  
249 achieving a considerable ultimate load. Otherwise, core crush was observed at a lower ultimate load than that  
250 required to fracture the FRP skins.

251 Moreover, the study used the value of energy/weight to demonstrate the performance of composites in  
252 terms of energy dissipation. The energy refers to the energy dissipation and the weight stands for the  
253 composite weight without the contribution of tensile ends. Based on the experimental results, the proposed  
254 composites exhibited improved values of the energy/weight from 3.28 J/g (average of five coupon tests) to  
255 6.17 J/g (skin fracture) and 10.85 J/g (core crush). This indicates immense potential of the proposed composite  
256 in terms of energy dissipation.

### 257 ***Impacts of FRP amount***

258 The results of fifteen tests (from three sets i.e., A-i, A-ii, and A-iii) were compared, as shown in Table 2 and  
259 Fig. 7, to explore the impacts of FRP amount. It should be noted that the ultimate stresses were obtained from

260 inputting ultimate loads (as listed in Table 2) into Eq. (1). All specimens failed in the form of skin fracture.

261 As listed in Table 1, all the parameters except the FRP amount were kept constant. Given that the height of

262 the cores in this section was 3 mm, the mid-section of the cores had perimeters of 19 mm to be covered with

263 FRP skins. The FRP skins consisted of three strands evenly separated from 10 mm-, 15 mm- and 20 mm-

264 wide strips (with a constant thickness of 0.51 mm), resulting in a strip-width/mid-section-perimeter ratio of

265 1.1, 0.79 and 0.53, respectively. It should be noted that all cores were fully and evenly covered by the FRP

266 material. Thus, the composite made by a wider strip was expected to have a larger skin thickness than that of

267 the corresponding composite resulted from a narrower strip.

268 Figs. 7 (a)-(c) illustrate the stress–strain curves for comparable tests using skins separated via FRP strips

269 having various widths. The comparisons among the typical tests are shown in Fig. 7 (d). When the strip width

270 was increased from 10 mm to 20 mm, the composites attained more notable nonlinear responses. Moreover,

271 increasing the strip width resulted in limited modulus improvement at the initial stresses up to 200 MPa, as

272 shown in Fig. 7 (d). Then, the composite having a wider strip tended to produce more core deformation,

273 resulting in a larger composite deformation. As shown in Table 2, increasing the strip width from 10 mm to

274 15 mm improved the average ultimate stress from 693 MPa to 768 MPa, the average ultimate strain from

275 0.019 to 0.039, and the average energy dissipation from 1.64 J/g to 7.00 J/g. Table 2 also provides the standard

276 deviations of these average values. A continually increasing width from 15 mm to 20 mm, however, produced

277 less improvements in terms of the ultimate stresses, ultimate strains and energy dissipation. Thus, the FRP

278 amount influenced the composite responses in terms of the stress–strain shape, energy dissipation, and

279 ultimate stress and strain. Since the composites using 15-mm-wide strips produced more convergent and

280 improvable stress–strain curves, a strip-width/mid-section-perimeter ratio of 0.79 was used to explore the

281 impacts of core configuration ( $h_c/l_c$ ).

### ***Impacts of core height and span***

282

283 Composites with  $h_c/l_c$  of 2/16, 3/16, 4/16 and 5/16 (i.e., B-i, A-ii, B-ii, and B-iii) were compared to  
284 investigate the impacts of core height and span. All parameters except  $h_c/l_c$  were kept the same as listed in  
285 Table 1. Various  $h_c/l_c$  ratios resulted from a constant span (=16 mm) and varying heights (from 2–5 mm).  
286 To effectively wrap the inner cores with various heights, a previously recommended strip-width/mid-section-  
287 perimeter ratio of 0.79 was used to calculate the required FRP amount. The required strip widths for cores  
288 with a height of 2 mm, 3 mm, 4 mm, and 5 mm were, therefore, 10 mm, 15 mm, 20 mm and 25 mm,  
289 respectively. All composites having cores with a height of 5 mm failed in core crush. The other composites  
290 failed at their ultimate stresses because of skin fracture. Increasing the height  $h_c$  from 2 mm to 4 mm  
291 produced more notable nonlinear stress–strain responses, as shown in Fig. 8 (a), Fig. 7 (b) and Fig. 8 (b).  
292 Composites with a height of 5 mm failed prematurely in core crush, producing the least ultimate stresses, as  
293 shown in Fig. 8 (c). This suggested that composites with a larger height had greater potentials for developing  
294 nonlinear tensile responses, but they could be much more vulnerable to core crush. The comparisons among  
295 the typical tests are shown in Fig. 8 (d). It can also be seen that increasing the height increased the composite  
296 deformability and reduced the stiffness at both the elastic and plastic stages. As listed in Table 2, increasing  
297 the value of  $h_c/l_c$  from 2/16 to 4/16 resulted in lower average stresses at skin fracture (from 822 MPa to  
298 687 MPa), larger average ultimate strains (from 0.012 to 0.09) and larger average energy dissipation (from  
299 1.49 J/g to 11.39 J/g). Composites with a  $h_c/l_c$  of 5/16 failed at the minimum stresses (average=369 MPa)  
300 but developed considerable average ultimate strains (0.054) and average energy dissipation (3.15 J/g). Given  
301 a constant span, a larger height allowed a larger deformation to be developed. Meanwhile, the cores with  
302 larger heights were much more vulnerable to crush failure. A  $h_c/l_c$  of 3/16 produced stable and improvable  
303 tensile behavior, and this value was used to explore the impacts of shell thickness.

### ***Impacts of shell thickness of inner cores***

304

305 Composites with shell thicknesses of 2 mm, 1 mm and 0.5 mm were tested to investigate the impacts of  
306 shell thickness  $t_s$ . The experimental results of fifteen tests from three sets (i.e., C-i, A-ii, and C-ii) were  
307 compared, as given in Table 2. All the parameters except shell thickness were kept constant. All the specimens  
308 having a shell thickness of 0.5 mm failed in core crush. The other tests failed at their ultimate stresses because  
309 of skin fracture. As shown in Figs. 9 (a)-(b), reducing the shell thickness tended to produce softer nonlinear  
310 stress–strain curves and less nonlinear responses. The comparisons among the typical tests are shown in Fig.  
311 9 (c); the results suggest that the shell thickness has limited impact on the initial modulus. Instead, the core  
312 with a thinner thickness tended to develop more notable nonlinear stress–strain responses at a lower stress  
313 level. Composites having a shell thickness of 2 mm developed the least deformations (average ultimate  
314 strain=0.017) and energy dissipation (average energy/weight=1.65 J/g), as listed in Table 2. The slightly  
315 deformed cores fractured skins at relatively lower ultimate stresses (average=643 MPa). Reducing the  
316 thickness from 2 mm to 1 mm increased the ultimate stresses and strains (i.e., for Specimens C-i and A-ii  
317 listed in Table 2). Continually reducing the thickness from 1 mm to 0.5 mm resulted in much larger  
318 deformations and energy dissipation but inner core crush occurred at a relatively lower ultimate stress  
319 (average=625 MPa). These observations indicated the trends of less notable nonlinear responses and larger  
320 deformations with increasing shell thickness.

### ***Impacts of core material***

321

322 Ten composites (from two sets i.e., B-ii and D-i) with a  $h_c/l_c$  of 4/16, core thickness of 1 mm, strip width  
323 of 20 mm and different core materials (ABS or PLA) were compared, as given in Table 2. The use of PLA  
324 material aimed to produce nonlinear stress–strain responses at higher “yield” stresses. When applying the

325 proposed composite to strengthen RC, a higher “yield” stress would be more effective in controlling the  
326 concrete cracks.

327 In the evaluation described in this section, all specimens failed in skin fracture. As shown in Fig. 10 (a),  
328 most composites using the PLA material remained elastic at higher stresses (approximately 300 MPa). When  
329 the ABS material was used, all the composites produced plastic responses before 300 MPa (see Fig. 8 (b) and  
330 Fig. 10 (b)). The composites using PLA material eventually failed at approximately 487 MPa with an average  
331 strain of 0.046, while the comparable composites with ABS material developed ultimate stresses more than  
332 670 MPa and ultimate strains no less than 0.084. Nevertheless, the PLA composites produced much more  
333 convergent loading responses than those of the ABS composites, demonstrating their merits of printing  
334 quality. The comparisons among the typical tests using ABS and PLA materials to print cores are shown in  
335 Fig. 10 (b). It can be seen that the material properties have limited impact on the initial modulus. Instead, a  
336 stiffer material (with a higher  $E_p$ , and  $f_p$ ) tended to develop nonlinear strain-stress responses at a higher  
337 stress than the comparable composite that used a softer material to print the inner cores did.

### 338 ***Impacts of core brace***

339 Fifteen tests (from three sets i.e., E-i, E-ii, and A-ii) were compared to investigate the impacts of core brace.  
340 All the parameters except brace arrangements were kept constant. The brace arrangements included one 2  
341 mm-thick, 0 ° brace and two 1 mm-thick 60 ° braces. This selection was made to provide braces with an  
342 equivalent overall thickness, i.e., one 2-mm-thick brace or two 1-mm-thick braces. Moreover, the brace-  
343 reinforced tests were also compared with corresponding brace-free tests (Specimen A-ii) to isolate the brace  
344 impacts.



345 All the tests failed in skin fracture. With the braces, the composites developed stiffer stress–strain responses  
346 than those of the comparable tests in which the braces were not used (see Fig. 11 and Fig. 7 (b)). The  
347 composites using braces achieved a stress of 400 MPa at a strain less than 0.01, as shown in Fig. 11 (a)-(b).  
348 The only exception (E-ii-5) developed a stress of 400 MPa at a strain of approximately 0.012. For the  
349 composites not using braces, the stresses at the strain of 0.01 were much less than 400 MPa, as shown in Fig.  
350 7 (b). Similarly, the stresses of brace-reinforced composites at the strain of 0.02 were approximately 600  
351 MPa, which was no less than the stresses obtained from their comparable tests that did not use braces. The  
352 comparisons among the typical tests using various braces and no braces are shown in Fig. 11 (c). The findings  
353 also suggest that the use of the brace and reducing the brace angle allowed the development of stiffer nonlinear  
354 responses and reduction of the deformability.

355 Increasing the brace angle from 0 ° to 60 ° resulted in limited increases in the average ultimate stresses (from  
356 670 MPa to 692 MPa) but considerable improvements in terms of the average ultimate strains (from 0.024 to  
357 0.030) and average energy dissipation (from 3.3 J/g to 4.2 J/g). Compared to those in the tests with braces,  
358 the composites that did not use braces achieved much greater average ultimate stress (768 MPa), average  
359 ultimate strain (0.039) and average energy dissipation (7.00 J/g).

360 As discussed above, the brace-reinforced cores tended to produce stiffer stress–strain relations with lower  
361 ultimate stresses and strains as well as less energy dissipation. Increasing the brace angle from 0 ° to 60 °  
362 improved the ultimate strain and energy dissipation.

### 363 ***Performance of composites with multiple cores***

364 Ten tests (from two sets, i.e., F-i and F-ii) were performed to investigate the impacts of core number. The  
365 cores were connected by 10-mm-long inner columns. Next, the inner cores and columns were helically

366 wrapped by three strands separated from 15-mm-wide FRP strips. The core configurations were kept constant  
367 for the brace-reinforced and brace-free composites, respectively (see Table 1).

368 As shown in Fig. 12 (a), the brace-reinforced composites developed nonlinear stress–strain responses,  
369 which were reasonably similar to those of the comparable tests with one single core (see. Fig. 11 (a)) in terms  
370 of the curve shapes and ultimate stresses. The ultimate strains corresponding to three-core composites were  
371 a little less than those of the comparable composites with one core (i.e., Specimen E-i listed in Table 2). The  
372 reason could be that the stiff bridges compromised the entire deformability of the composite with multiple  
373 cores, resulting in less ultimate strains. For the brace-free composites, tests having multiple cores produced  
374 much less ultimate strains (see Fig. 12 (b)) compared to those of the specimens with one core (i.e., Specimen  
375 A-ii shown in Fig. 7 (b)), thereby indicating that the stiff bridges had a more considerable impact on the  
376 composites with brace-free cores. Moreover, all tests having three cores failed in skin fracture. This suggested  
377 that the proposed helical system effectively resisted the skin–core delamination and the opening stresses at  
378 the core–bridge connections. Otherwise, the composites would have failed in premature delamination owing  
379 to the development of much lower ultimate stresses. Thus, the composite can be extended by using multiple  
380 cores. Fig. 12 (c) illustrates the typical tests of composites having multiple-braced or brace-free cores.  
381 Similarly, the brace had limited impact on the initial modulus but tended to reduce the composite  
382 deformability. This trend is the same as that observed from the corresponding tests using a single core. This  
383 suggests that the parametric studies of composites having one-single core can be used for predicting the trend  
384 of the corresponding composites with multiple cores.

## Conclusion

385

386 This study proposed a novel concept to achieve high-strength, large-deformation, ductile, tensile-behavior-  
387 designable FRP composites. The composite consists of three components: (i) inner cores (providing  
388 designable stiffness, deformation and profile to shape skins), (ii) helical skins (providing primary resistance  
389 for tensile loads and openings at core edges), and (iii) bridges (providing connections to achieve a desired  
390 length). The experimental results from sixty-five tests validated the capability of (1) the helical system to  
391 resist the opening stress and (2) the proposed composite to develop nonlinear stress–strain responses.  
392 Moreover, the impacts of FRP amount, core configurations (i.e., height  $h_c$ , span  $l_c$ , shell thickness  $t_s$ , inner  
393 braces) and core material within the investigated ranges are clarified as follows.

394 1. Increasing the FRP amount had a limited impact on the initial modulus but tended to develop increasingly  
395 pronounced nonlinear responses and larger ultimate strains. Moreover, increasing the core height tended  
396 to develop more notable nonlinear responses, reduce the stiffness at both elastic and plastic stages and  
397 transfer the failure modes from skin fracture to core crush. To develop considerable nonlinear responses,  
398 the recommended core material is ABS and the other parameters are as follows: FRP strip width = 20  
399 mm,  $t_s = 1$  mm, and  $h_c/l_c = 4/16$ .

400 2. The shell thickness is another important parameter that can influence the composite behavior. Although  
401 reducing the shell thickness has limited contributions to the initial modulus, a thinner shell tends to  
402 develop more notable nonlinear stress–strain responses at a lower stress level, and the specimen is thus  
403 more vulnerable to core crush. In this study, all the composites with a 0.5 mm-thick shell,  $h_c/l_c = 3/16$ ,  
404 strip width = 15 mm and ABS printed cores failed in core crush.

405 3. Alternatively, various core materials can be selected to control the tensile behavior of the proposed  
406 composite. A stiffer material (e.g., PLA with a higher  $E_p$ , and  $f_p$ ) tended to develop nonlinear strain–

407 stress responses at a higher stress than the comparable composite that employed a softer material (e.g.,  
408 ABS) to print inner cores did. Moreover, a stiffer PLA material allowed less core deformation, resulting  
409 in less notable nonlinear responses and fracturing skins at lower stresses than those of comparable tests  
410 using a softer ABS material to print cores.

- 411 4. Core braces can also be used to composite behaviors. Composites supported by core braces tended to  
412 produce stiffer stress–strain relations with lower ultimate stresses and strains than the comparable one  
413 that did not use braces did. Increasing the brace angle from 0 ° to 60 ° improved the ultimate strain.
- 414 5. The proposed composite could be extended by using multiple cores.

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421

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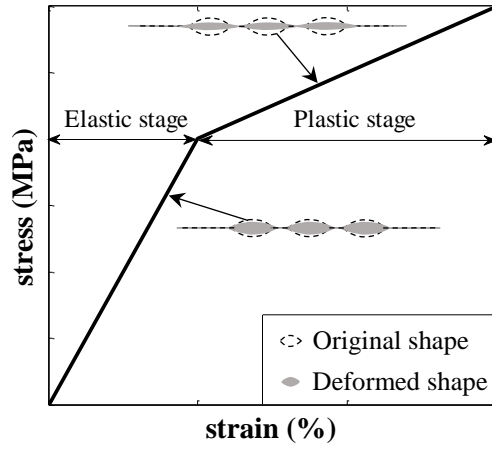
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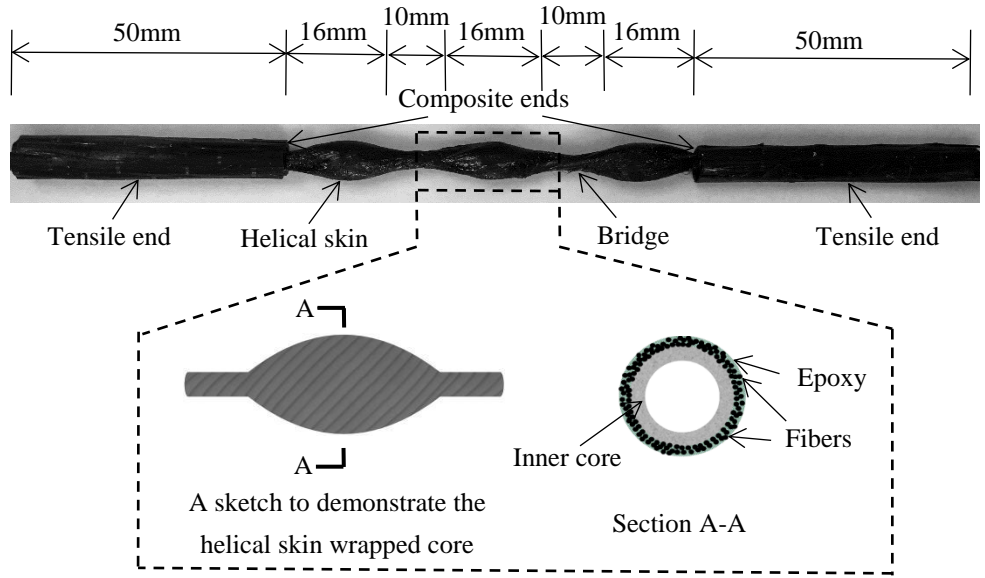


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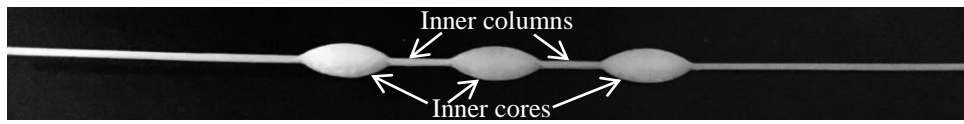


**Fig. 1.** Typical tensile responses of the proposed composite

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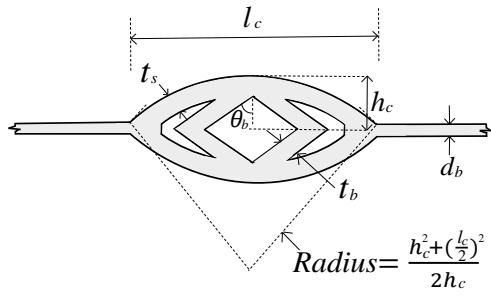


(a) A typical composite



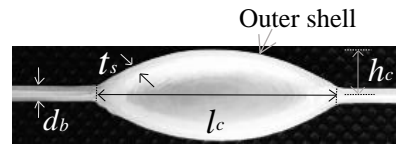
(b) Inner system consisting of cores and columns

**Fig. 2.** Proposed composite



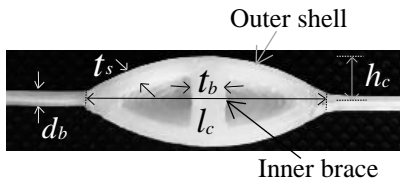
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533 (a) Geometry of an inner core with two braces



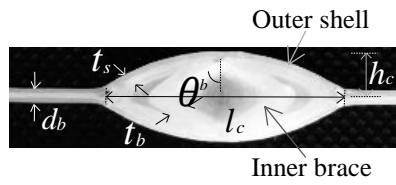
(b) A typical core without braces

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(c) A typical core with a 0° brace

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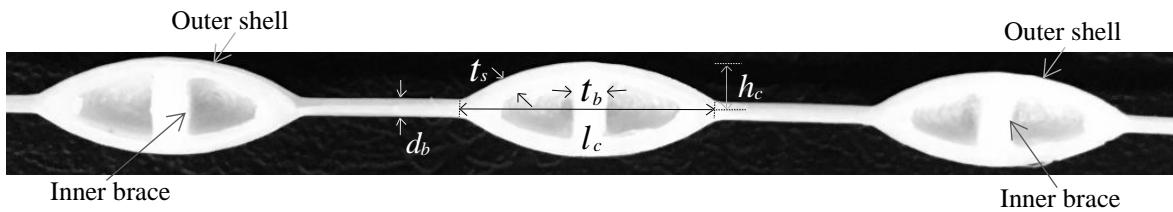


(d) A typical core with two 60° braces

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(e) A typical composite having three cores

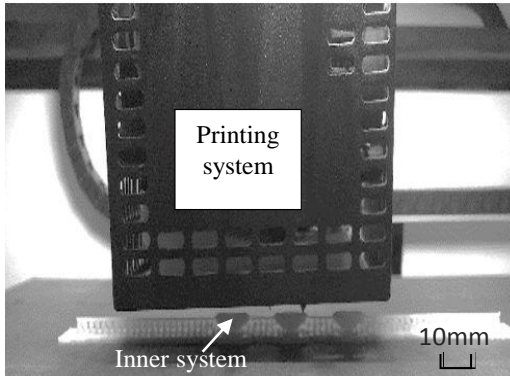
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**Fig. 3. Core configurations**

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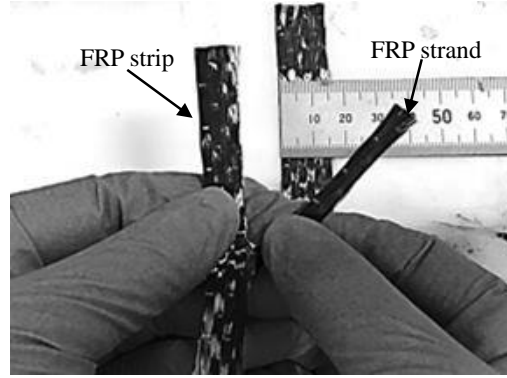


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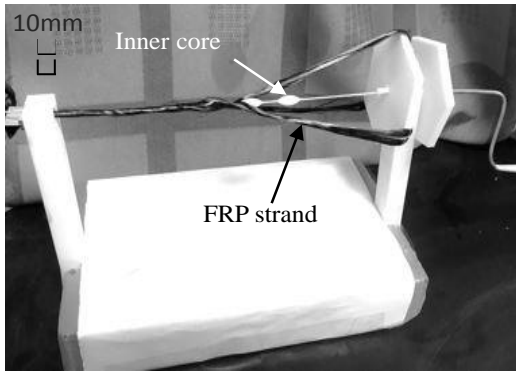
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(a) Printing inner cores and inner columns



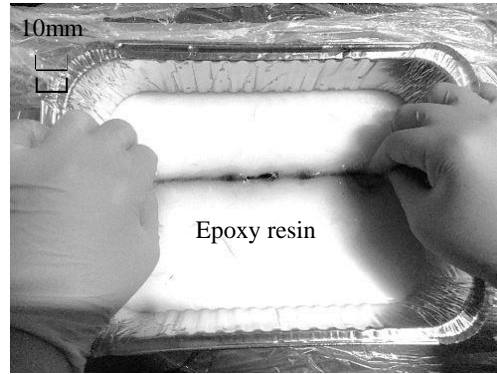
(b) Separating strands from a given width of an FRP strip



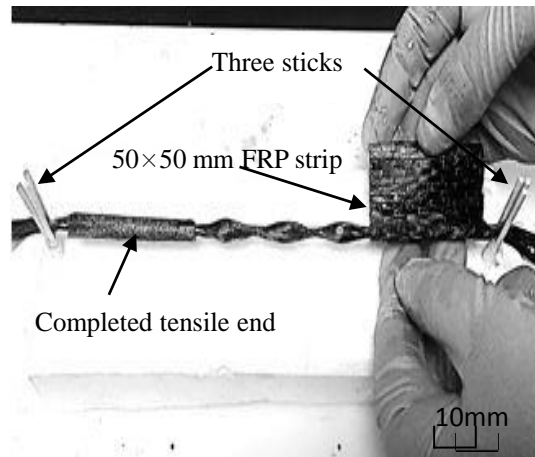
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(c) Helically wrapping the printed system



(d) Saturating the wrapped composite



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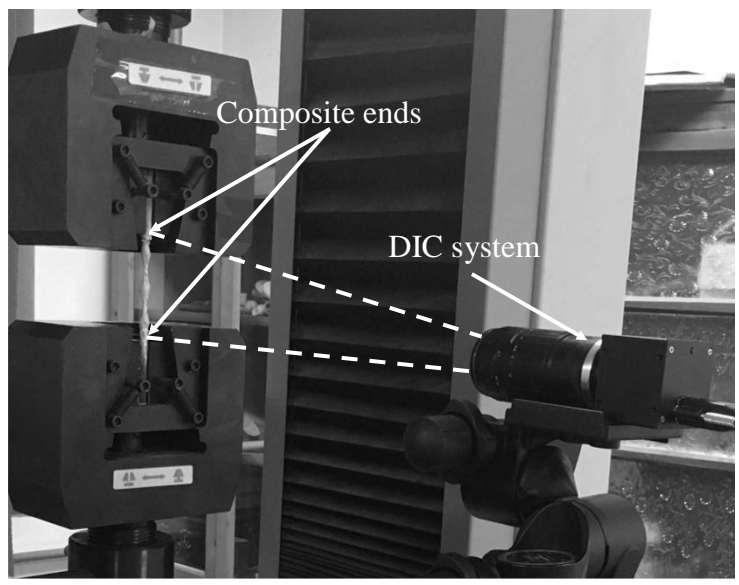
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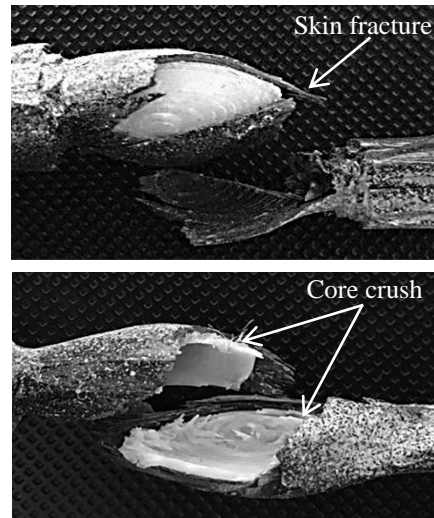
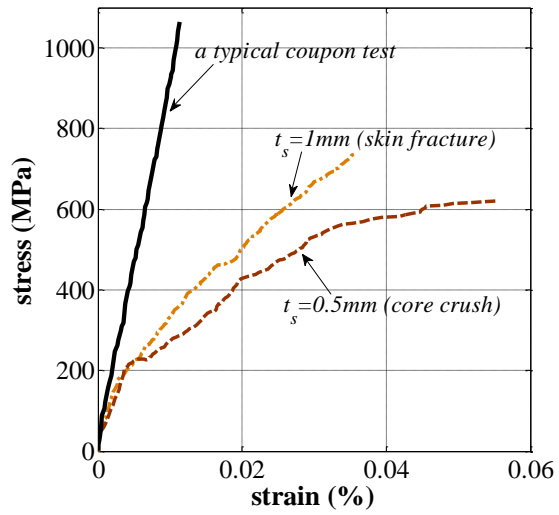
(e) Wrapping both ends for tensile testing

**Fig. 4.** Fabrication of the proposed composite

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**Fig. 5.** Test setup



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573 **Fig. 6.** Stress–strain plots for typical composites using 15-mm-wide FRP strips, ABS material,  $h_c/l_c$  of 3/16,

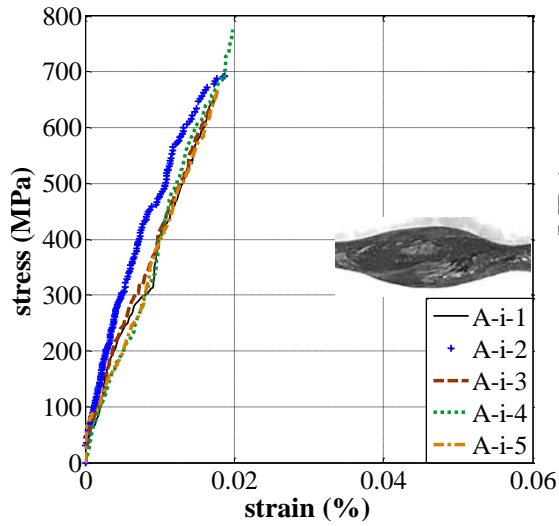
574 various shell thicknesses and failure types of skin fracture (A-ii-5) or core crushing (C-ii-3)

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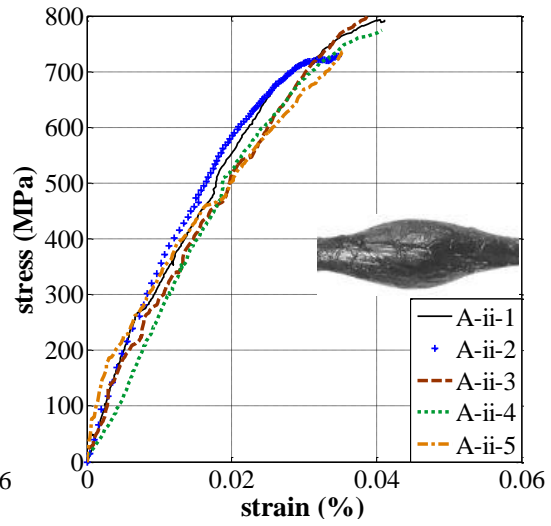
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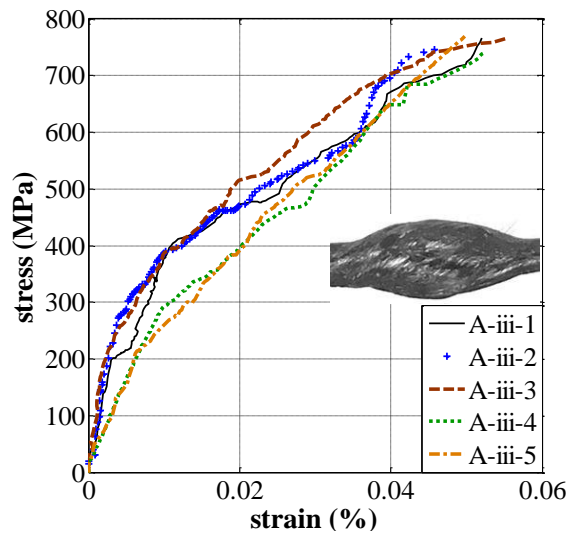
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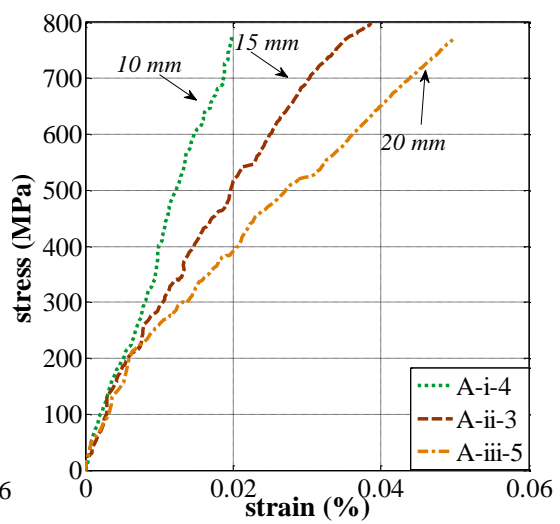
(a) 10-mm-wide FRP strips



(b) 15-mm-wide FRP strips

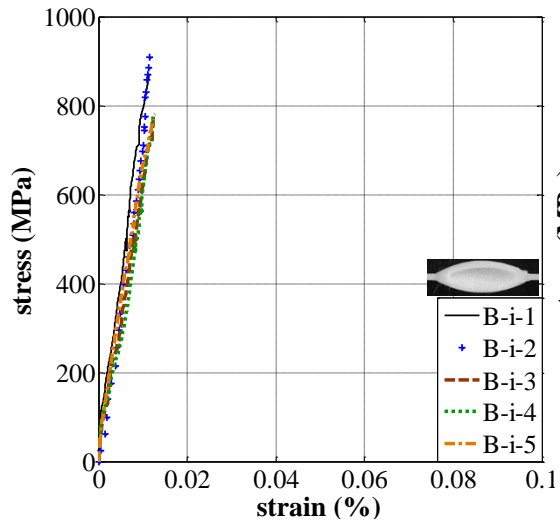


(c) 20-mm-wide FRP strips



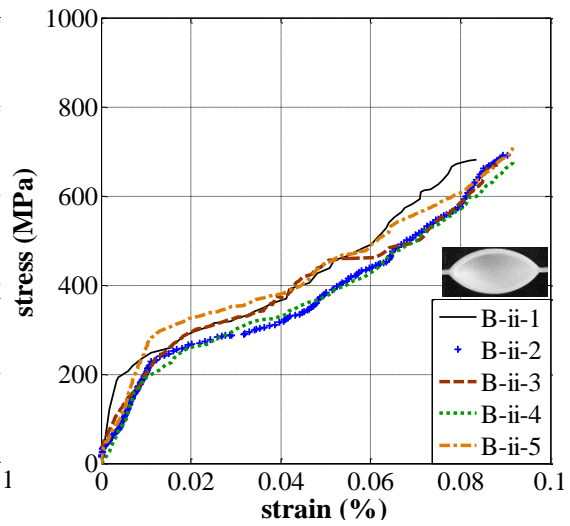
(d) Typical tests with various strip widths

**Fig. 7.** Stress–strain curves for composites involving the use of various strip widths to construct the FRP skins



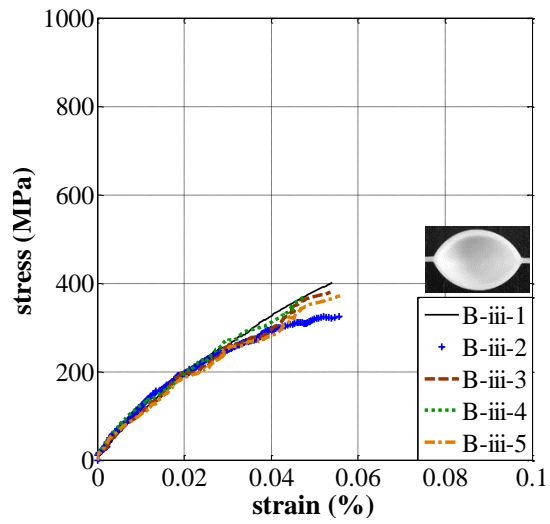
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(a) 2-mm-high cores



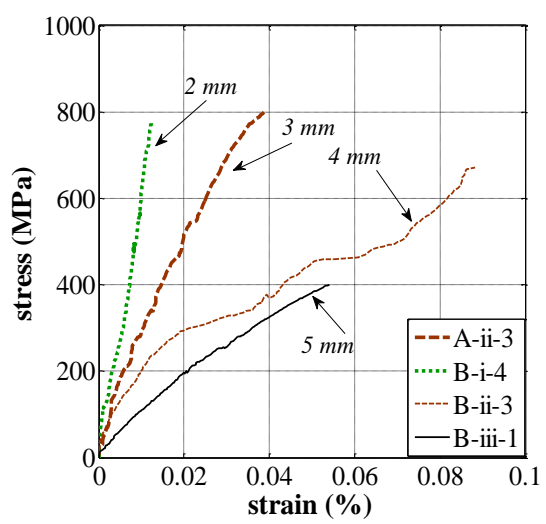
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(b) 4-mm-high cores



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(c) 5-mm-high cores



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(d) Typical tests with various core heights

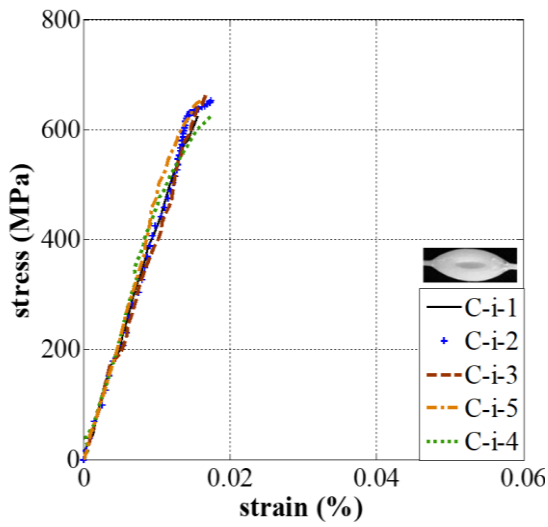
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**Fig. 8.** Stress–strain curves for composites having various core heights

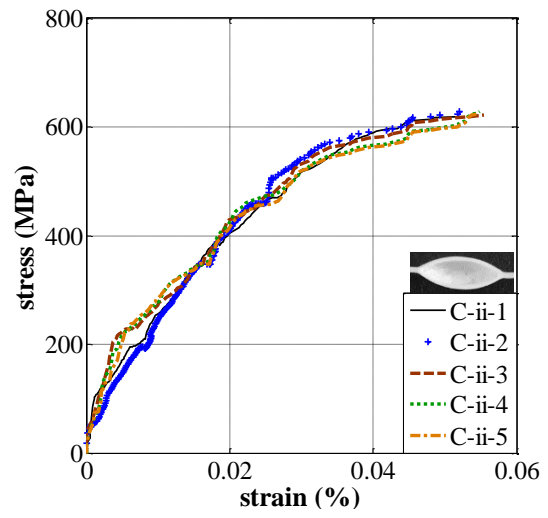
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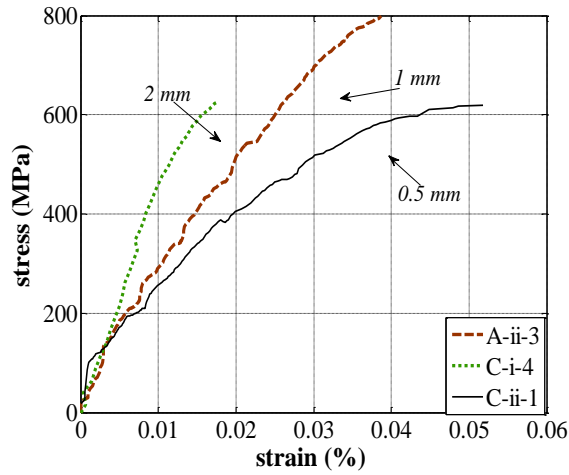
(a) Shell thickness  $t_s = 2$  mm



(b) Shell thickness  $t_s = 0.5$  mm

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(c) Typical tests with various shell thicknesses

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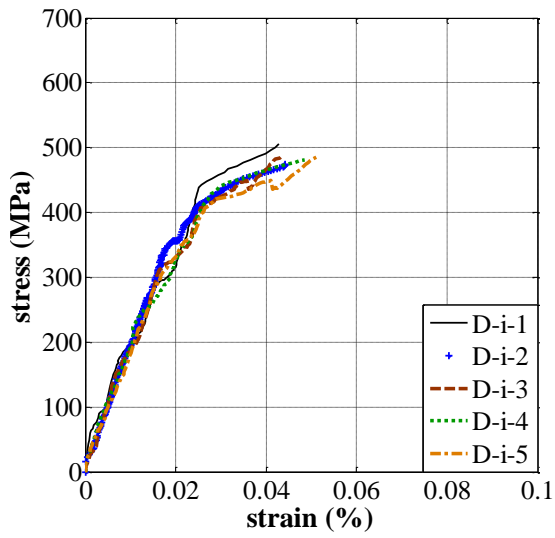
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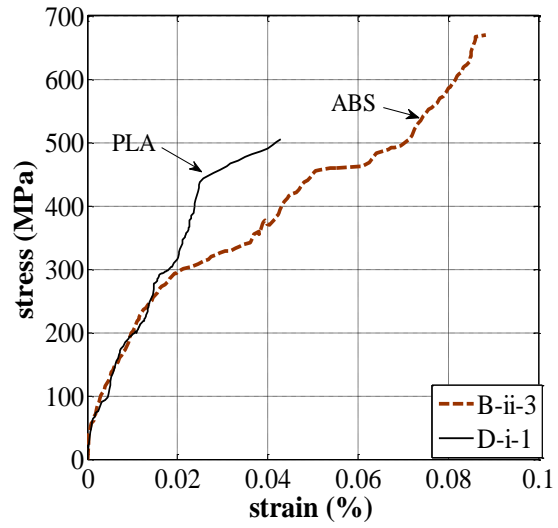
**Fig. 9.** Stress–strain curves for composites having various shell thicknesses

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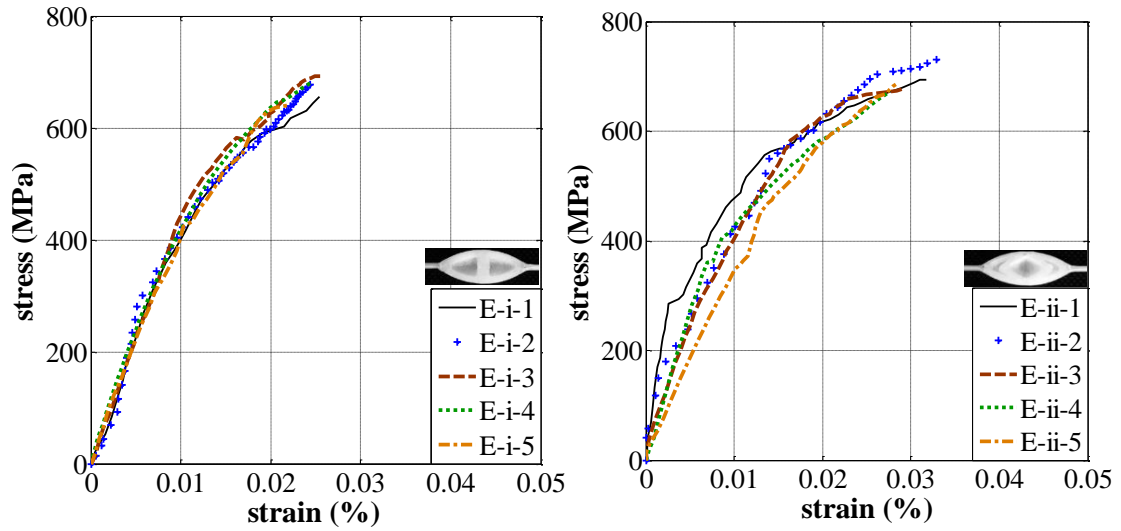
(a) Composites with a PLA core



(b) Typical tests with various core materials

**Fig. 10.** Stress–strain curves for composites involving the use of PLA or ABS material to print cores

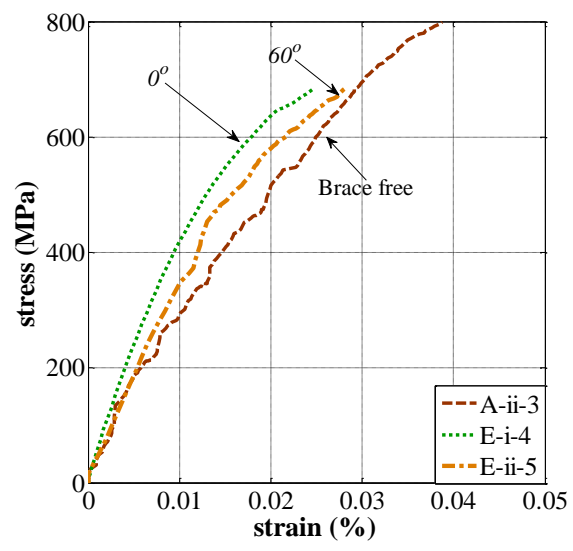
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(a) Brace angle  $\theta_b=0^\circ$

(b) Brace angle  $\theta_b=60^\circ$

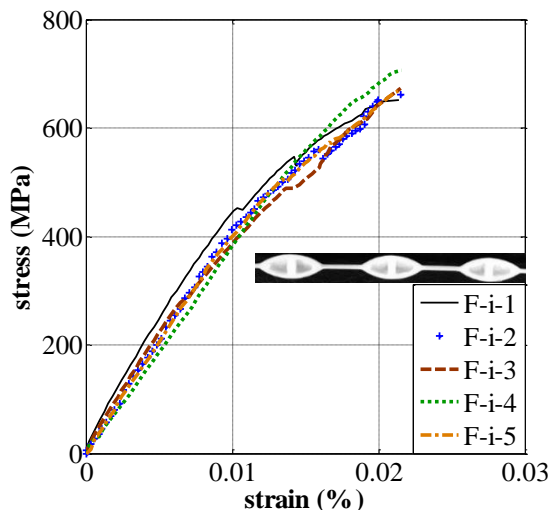
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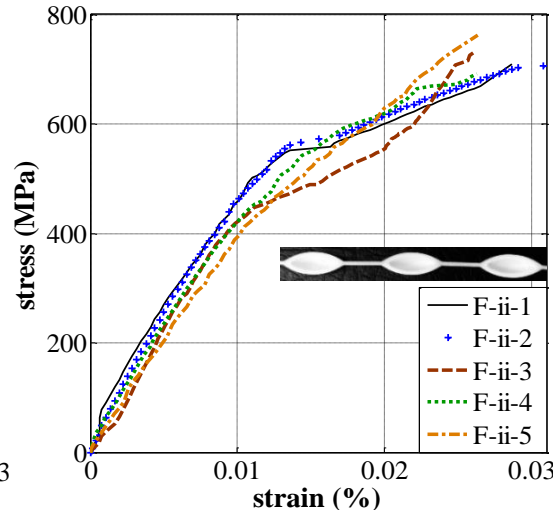
(c) Typical tests with various brace arrangements

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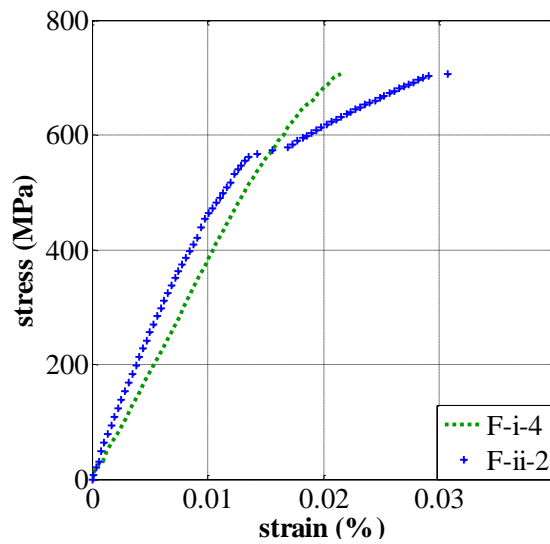
**Fig. 11.** Stress–strain curves for composites having various brace arrangements



(a) Brace reinforced cores



(b) Brace free cores



(c) Typical tests having multiple cores and various brace arrangements

**Fig. 12.** Stress–strain curves for composites having multiple cores