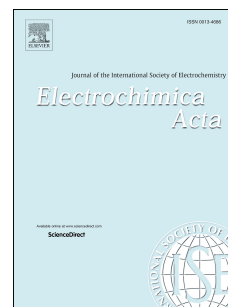


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Edmundo S.V. Neto, Caio V.S. Almeida, Andrea E. Russell, Giancarlo R. Salazar-Banda, Katlin I.B. Eguiluz



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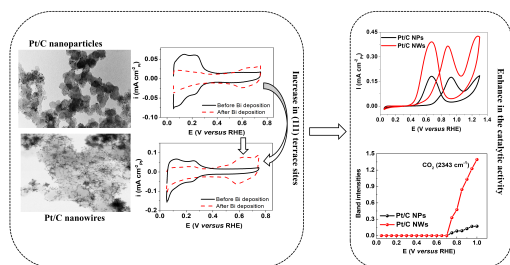
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Katlin I. B. Eguiluz: Project administration, Supervision, Funding acquisition, conceptualization and planning of the work and Writing - Review & Editing.



Journal Pre-proof

Realising the activity benefits of Pt preferential (111) surfaces for ethanol oxidation in a nanowire electrocatalyst

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1 Abstract

2 The commercialisation of direct ethanol fuel cells (DEFCs) is hindered by the low activity
3 of the currently available electrocatalysts that are mainly carbon-supported nanoparticles
4 (NPs). Here, we synthesised carbon-supported Pt nanowires (Pt/C NWs) by chemical
5 reduction of metallic precursors at room temperature without using surfactants, templates, or
6 stabilising agents. The synthesised Pt NWs were supported on high surface area carbon
7 (Vulcan XC-72R) with 40 wt%. of metal loading concerning the support. The electroactivity
8 of the synthesised NWs catalyst towards ethanol and CO (the dominant intermediate species)
9 oxidation is investigated and tested in comparison with a commercial nanoparticle Pt/C
10 electrocatalyst (named: Pt/C NPs). The electrooxidation of ethanol was investigated in acid
11 media by CO-stripping, cyclic voltammetry, derivative voltammetry, chronoamperometry,
12 steady-state polarisation curves, and *in situ* Fourier transform infrared spectroscopy (FTIR)
13 experiments. The results showed that the synthesised Pt NWs are much more active than the
14 Pt/C NPs catalyst, for both ethanol and CO-monolayer oxidation in acidic media. *In situ*
15 FTIR data revealed that Pt NWs catalyst favours the formation of CO₂ and acetic acid. This
16 activity is mainly due to the existence of extended terraces, resulting in enhanced mobility of
17 OH_{ads} and CO_{ads}, facilitating the removal of CO_{ads} from the catalyst surface and allowing
18 better ethanol adsorption for further oxidation. Hence, 5-fold higher current density for
19 ethanol oxidation is obtained. The NW morphology of Pt nanocatalysts results in very active
20 materials towards ethanol oxidation and may provide a promising means of increasing the
21 performance of anodes for DEFCs.

22

23

24 **Keywords:** Nanowires; Ethanol electro-oxidation; Current density; Adsorption.

25

26

27 1. Introduction

28 Increasing environmental protection concerns and the increasing world energy consumption
29 have resulted in a major quest for the development of cleaner and alternative energy
30 production systems, preferentially those not working with fossil fuels. In this context, small-
31 scale generation systems, such as wind turbines, solar energy, and fuel cells (FCs), play an
32 essential role in meeting global needs through distributed generation [1]. FCs are very
33 attractive devices able to obtain directly the electric energy contained in fuels (commonly H₂
34 or liquid fuels like methanol and ethanol), releasing water as a principal by-product [1,2].

35 In FCs, the direct conversion nature avoids mechanical frictional losses, increasing the
36 fuel overall conversion efficiency. Moreover, these systems are not limited by the
37 thermodynamic constraints derived from the Carnot cycle [3]. Most of the studies have been
38 focused on using hydrogen as fuel; however, difficulties in transportation, stocking, and
39 distribution hinder the FCs further application. Thus, by using liquid fuels, such as methanol
40 or ethanol, the need to adapt or re-establish the existing infrastructure (as required for
41 hydrogen) can be reduced. As methanol is a toxic fuel, ethanol appears as a viable alternative
42 [4].

43 Ethanol is a C-2 containing alcohol with important features, such as non-toxicity,
44 large scale production from biomass, easy transportation and storage, high energy density
45 (8.0 kWh kg⁻¹), and high solubility in aqueous electrolytes [2,5] that make it attractive for
46 sustainable energy applications. In addition, ethanol possesses approximately two-thirds of
47 the energy density of pure gasoline [6,7]. In the USA and Brazil, ethanol has a well-
48 established distribution network, being available in many gas/petrol stations, unlike methanol
49 [8].

50 The energy delivered from a direct ethanol fuel cell (DEFC) is theoretically higher
51 than at a direct methanol fuel cell (DMFC, 8.0 kWh kg⁻¹ versus 6.1 kWh kg⁻¹, respectively),

52 thus reducing the fuel consumption [6]. However, the complete ethanol oxidation reaction
53 (EOR) is still a challenge due to the complexity of the reaction. The total oxidation of ethanol
54 involves the cleavage of the C–C bond and the transfer of 12 electrons (e^-) per ethanol
55 molecule [9]. There is a consensus that the partial ethanol oxidation (with parallel or
56 consecutive reactions) generates intermediates such as carbon monoxide (CO). CO strongly
57 adsorbs at the catalyst surface, poisoning its active sites, and decreasing the FC efficiency
58 [10].

59 Conventionally, alcohol-fuelled FCs use acid proton-exchange membranes and
60 platinum-based catalysts [11]. The catalysts (Pt or other highly active materials) develop a
61 critical role in the cost and durability of the FC system [3]. The Pt-based anodes and cathodes
62 represent 35–42% of the total FCs cost. Therefore, since the partial oxidation pathway
63 (without the release of 12 e^-) dominates on pure Pt catalysts, the greatest challenge in
64 developing catalysts for ethanol oxidation is to facilitate C–C bond cleavage and thereby total
65 oxidation at low potentials [9]. For this purpose, some strategies have been widely applied,
66 such as (i) the incorporation of more oxyphilic materials, (ii) the modification of the catalyst
67 morphology, and (iii) the change/modification on the support material.

68 DEFCs still suffer from the limited activity and stability of Pt NPs, even for the best
69 catalysts. These problems are derived from the loss of Pt electrochemical surface area (ESA)
70 over time, caused by the corrosion of carbon support and Pt dissolution/aggregation. Pt NPs
71 possess high surface energies that induce severe Ostwald ripening and/or grain growth during
72 the operation of the FC [12]. One-dimensional (1D) structures (such as NWs or nanotubes)
73 may exhibit greater stability than NPs, whilst retaining high mass activities owing to (i) their
74 high surface-area-to-volume ratio, (ii) the presence of large areas of smooth crystal planes
75 and (iii) lower number of surface defect sites [13].

76 In this paper, we studied the electrocatalytic activity of carbon-supported Pt NWs
77 towards ethanol and CO-monolayer oxidation in acidic media. The Pt/C NWs were
78 synthesised without the use of any surfactant, template, or stabilising agent, and were
79 compared with a commercial Pt/C reference catalyst. The comparison enabled the study of
80 the effects caused by the morphology modification previously suggested by the study of
81 single-crystal surfaces [14] to develop further highly active electrocatalysts, which may be
82 successfully applied as anodes for DEFCs. Notably, we demonstrate that the NW morphology
83 has a positive effect on the catalytic activity of the catalyst showing almost 2-fold higher
84 mass activity in comparison with the commercial NPs Pt catalyst.

85

86 **2. Experimental**

87 *2.1 Nanowires synthesis*

88 Pt NWs were synthesised by the chemical reduction of the metallic precursor by formic acid
89 ^[12,14] without the utilisation of surfactants, template, or stabilising agents. For the preparation
90 of 100 mg of catalyst, 5.31 mL of hexachloroplatinic acid solution ($0.03861 \text{ mol L}^{-1}$) (Sigma-
91 Aldrich[®], 37.5% of Pt) was dissolved in 4 mL of ultrapure water and left in an ultrasonic bath
92 (Ultronique QR500) for ~30 min. The hexachloroplatinic acid solution was diluted to a total
93 volume of 80 mL with ultrapure water, and 60 mg carbon black Vulcan XC-72R powder was
94 added to the solution (to give a loading of 40 wt% Pt in the resulting catalyst) and stirred for
95 15 min. Then, 7.24 mL of pure formic acid (CH_2O_2 , Sigma-Aldrich[®], 98–100%) was slowly
96 added, and the suspension was stirred for more 15 min. The resulting solution was stored for
97 72 h to allow the NWs growth, during which the solution changed from black to colourless
98 (indicating the successful reduction of metallic precursors), and the resulting catalyst powder
99 dropped down to the bottom of the reaction flask. The resulting suspension was vacuum-
100 filtered, washed several times with ultrapure water and dried in an oven for 30 min at $60 \text{ }^\circ\text{C}$,

101 and finally stored for further electrochemical and physical characterisations. This catalyst is
102 referred to as Pt/C NWs in the text below.

103 A commercial nanoparticle Pt/C (Pt on carbon black, from Johnson Matthey™) was
104 used as a reference catalyst and is referred to as Pt/C NPs in the text below.

105

106 ***2.2 Physical characterisation***

107 The XRD patterns were obtained using a Bruker D8 Advance diffractometer operating with
108 Cu K α radiation ($\lambda = 0.15406$ nm) machine. The patterns were generated at 50 kV and 100
109 mA with a scanning speed of 2° min^{-1} and sweep range between 20° and 90° . The
110 diffractograms were compared with the JCPDS (Joint Committee of Powder Diffraction
111 Standard) crystallographic cards. The morphology of the catalysts and their sizes were
112 observed by transmission electron microscopy (TEM) using an FEI Tecnai 12 at an 80-kV
113 accelerating voltage.

114

115 ***2.3 Electrode preparation and electrochemical measurements***

116 The electrochemical measurements were performed at room temperature in 0.5 mol L^{-1}
117 H_2SO_4 aqueous solution saturated with high purity N_2 (White Martins, 99.9 %) to eliminate
118 dissolved O_2 . All experiments were carried out in a conventional one compartment Pyrex®
119 glass three-electrode electrochemical cell, using an Autolab Model PGSTAT 302N
120 potentiostat/galvanostat. A Pt spiral wire was used as the counter electrode, a reversible
121 hydrogen electrode (RHE) prepared at the same electrolyte solution immersed in a Luggin
122 capillary was used as the reference electrode, and a glassy carbon disk (GC, 3 mm in
123 diameter) was used as working electrode. The supporting electrolyte was used to produce
124 hydrogen in the reference electrode, by applying a constant negative potential of -6.0 V. All

125 aqueous solutions used at this work were prepared with ultrapure water (Gehaka model MS
126 2000 system, 18.2 M Ω ·cm).

127 Before the measurements, the GC electrode was polished with a 0.3 μ m α -alumina
128 slurry, then, washed with 2-propanol and sequentially with ultrapure water in an ultrasonic
129 bath, to remove the surface residues and renew the electrode surface, before each experiment.
130 Electrocatalyst inks were prepared by adding 2.9 mg of the catalyst to a solution composed of
131 1000 μ L of 2-propanol and 30 μ L of 5 wt. % Nafion[®] (~5% lower aliphatic alcohols, Sigma
132 Aldrich[®]). The resulting ink was sequentially submitted to an ultrasonic bath for 20 min for
133 the dispersion and homogenization of the powder catalysts in the solution. 5 μ L of the
134 catalyst ink was deposited dropwise to the GC electrode surface and dried at room
135 temperature.

136 For all electrochemical measurements, the catalyst ink modified GC electrodes were
137 first submitted to 500 voltammetric cycles between 0.05 and 0.80 V at 500 mV s⁻¹ to obtain
138 reproducible voltammograms in 0.5 mol L⁻¹ H₂SO₄ aqueous solution.

139 Cyclic voltammograms were recorded over 0.05 V to 1.30 V (2 cycles) performed at
140 20 mV s⁻¹. The electrochemically active surface area (ECA) was determined by integration of
141 the monolayer hydrogen desorption peaks of the voltammograms (0.0–0.3 V) using a charge
142 density of 0.21 mC cm⁻² [15] and all electrochemical results presented below are normalised
143 by the ECA.

144 The irreversible/adsorbed Bi process was used to quantify the number of {111}
145 surface sites present in the catalysts, according to the method established by Feliu and co-
146 workers [16, 17]. Thus, a cyclic voltammogram was obtained in the potential range of 0.05–
147 0.75 V at 50 mV s⁻¹ in 0.5 mol L⁻¹ H₂SO₄ aqueous solution. The glassy carbon electrode was
148 removed from the solution and washed with ultra-pure water. Then, the CV electrode was
149 immersed in a solution containing Bi₂O₃ (10⁻⁴ mol L⁻¹ of Bi) in HClO₄ (0.6 mol L⁻¹) for 1

150 min (enough time for the total covering of the sites (111) of the Pt). Bi₂O₃ was previously
151 calcined at 500 °C for 45 min. Finally, the electrode was immersed into a 0.5 mol L⁻¹ H₂SO₄
152 aqueous solution, and a new voltammogram was collected.

153 For the study of the EOR, ethanol was added (C₂H₆O, Sigma Aldrich, 99.5 %) until a
154 0.5 mol L⁻¹ concentration and additional cyclic voltammograms were recorded. Derivative
155 voltammetry was used to obtain more precise information regarding the EOR mechanism.
156 The derivative voltammograms were constructed from the derivative of the first cycle of
157 ethanol oxidation voltammogram, between 0.05 and 1.2 V *versus* RHE.

158 CO stripping experiments were carried out in 0.5 mol L⁻¹ H₂SO₄ by first bubbling CO
159 gas into the electrochemical cell for 5 min applying 0.05 V *versus* RHE, followed by purging
160 with high purity N₂ for 25 min to remove the CO remaining in the solution. Sequentially, the
161 CO stripping voltammogram was obtained (1 cycle) between 0.05 V and 1.30 V *versus* RHE
162 at 20 mV s⁻¹.

163

164 **In situ infrared spectroscopic measurements**

165 *In situ* Fourier Transform Infrared (FTIR) measurements were carried out on a Bruker
166 VERTEX 70v FTIR spectrometer equipped with an MCT detector cooled with liquid
167 nitrogen. The reflectance spectra were collected with an R/R₀ ratio, in which R represents the
168 spectra at a given potential, and R₀ is the collected spectra where no reaction of interest
169 happens, in our case, 50 mV. Besides, the spectra were calculated from the average of 128
170 interferograms, with the spectral resolution set at 4 cm⁻¹. The *in situ* spectroelectrochemical
171 system was equipped with a CaF₂ window, and the spectra were scanned between 3000 and
172 1000 cm⁻¹ wavelength range.

173 The catalytic suspension was supported on a gold substrate previously polished with
174 alumina (0.05 μm), to obtain a surface with high reflectivity. The catalyst was characterised

175 in an H_2SO_4 0.5 mol L^{-1} solution and transferred to the spectroelectrochemical cell to
176 investigate the EOR by *in situ* FTIR (in real-time). Then, ethanol (0.5 mol L^{-1}) was added,
177 and the spectra were collected as a function of the applied potential. The scan rate and the
178 number of interferograms were chosen to allow the spectra to be collected at intervals of
179 about 0.05 V between $0.05\text{--}1.0 \text{ V}$ of potential *versus* RHE.

180

181 **3. Results and discussion**

182 **3.1. Physical characterisation**

183 Fig. 1 shows the TEM micrographs of the commercial Pt/C NPs (Fig.1A) and the synthesised
184 Pt/C NWs catalyst (Fig.1B). Pt NWs were successfully synthesised by the formic acid as the
185 reducing agent methodology, as demonstrated by the presence of well-defined multi-armed
186 star-like Pt NWs in the images. In Fig. 1B, Pt NWs are composed of several short arms
187 approximately 20 nm in length and 4 nm diameter. The good dispersion of the Pt NWs over
188 the carbon support may indicate a favourable interaction between the metallic atoms and the
189 exposed crystal planes of the carbon [18]. The selected-area diffraction (SEAD) pattern (Fig.
190 1B inset) reveals a series of bright concentric rings, which are attributed to face-centred cubic
191 (fcc) structure of crystal Pt, which is similar to that of bulk Pt [19,20]. These bright rings are
192 assigned to $\{111\}$ (inner ring), $\{200\}$, $\{220\}$ and $\{311\}$ (outer ring) planes, confirming that
193 this method produces highly crystalline NWs [18-20].

194 The powder XRD patterns for the commercial Pt/C NPs and the synthesised Pt/C
195 NWs catalysts are shown in Fig. 2. The XRD patterns were compared with the JCPDS (Joint
196 Committee of Powder Diffraction) crystallographic cards. Similar peak positions are
197 observed for both catalysts. The diffraction peak at $2\theta = 26.54^\circ$ is ascribed to the graphite
198 carbon (PDF card no. 001-0646). For both catalysts, the peaks at $2\theta = 39.91^\circ$, 46.44° , 67.75° ,
199 81.60° , and 86.21° are assigned to *fcc* phase of Pt reflections planes (111), (200), (220), (311)

200 and (222), respectively, in accordance with PDF card no. 004-0802, in agreement with TEM
201 images shown in Fig.1.

202 The Pt(220) diffraction peak (inset) shifted slightly to higher 2θ values, attributed to
203 the NWs anisotropic structure[21]. As previously reported by Sun et al. [22], the reduction
204 reaction used to form the NW at room temperature proceeds slowly, favouring the anisotropic
205 growth according to the lowest energy principle. For *fcc* structures, the order of surface
206 energies is $(111) < (100) < (110)$, and therefore, the relative growth rate along the closed-
207 packed $\langle 111 \rangle$ direction is increased. However, the exact growth mechanism is not fully
208 known [21].

209 In order to evaluate the metallic loading of the catalysts, EDX measurements were
210 recorded (Fig. S1). The EDX analysis confirmed that the metallic loading of the as-prepared
211 Pt/C NWs catalyst (40.7 ± 0.53 wt%) and for Pt/C NPs (19.5 ± 0.10 wt%) is close to the
212 nominal values, respectively, 40 and 20 wt%, as expected.

213

214 3.2. Electrochemical characterisation

215 Fig. 3 shows the cyclic voltammograms (second scan), recorded in $0.5 \text{ mol L}^{-1} \text{ H}_2\text{SO}_4$
216 between 0.05 V and 1.30 V *versus* RHE at 20 mV s^{-1} , for the synthesised NWs catalyst and
217 commercial NP Pt/C. Both catalysts show typical profiles of carbon-supported Pt-based
218 catalysts in acidic media [23]. The voltammograms display pairs of well-defined peaks in the
219 hydrogen adsorption/desorption region (0.05 to 0.30 V *versus* RHE), with differences in the
220 desorption peaks attributed to preferential exposure of multiple crystallographic planes [24]
221 for the NWs catalyst.

222 Comparison to the published cyclic voltammograms of single-crystal Pt electrodes
223 [24], suggests that for the Pt/C NWs catalyst, the main peaks at ~ 0.11 V and ~ 0.27 V (0.21 V
224 for the commercial Pt/C catalyst) *versus* RHE show can be attributed to H adsorbed on (110)

225 steps and (100) step sites on (111) terraces, respectively. For the Pt/C NWs, the peak
226 corresponding to (100) step sites on (111) terraces, which arise from (111) x (111) junction,
227 is more pronounced. The commercial catalyst showed a broader second peak, which is
228 attributed to contributions from short (110) terraces or kinks, indicative of the NP
229 (cubooctahedral or icosahedral) shape. The presence of a peak at ~ 0.55 V for the synthesised
230 Pt/C NWs catalyst may also be an indication of the presence of (111) terraces contribution
231 [24], although this peak overlaps with region often attributed to the quinone/hydroquinone
232 couple on the carbon support [25].

233 Fig. 4 shows the voltammetric profiles recorded in $0.5 \text{ mol L}^{-1} \text{ H}_2\text{SO}_4$ between 0.05 V
234 and 0.75 V *versus* RHE at 50 mV s^{-1} before and after Bi adsorption for Pt/C NPs (Fig 4a) and
235 Pt/C NWs (Fig. 4b) catalysts. In order to estimate the fraction of (111) ordered domains
236 present on the catalysts surface, bismuth irreversible adsorption was carried out (red dashed
237 lines at Fig. 4a and 4b). The charge measured under the peak at 0.62 V was used to determine
238 the charge associated with (111) terrace sites present on the surface of the catalyst [16, 17].
239 The Pt/C NPs and Pt/C MWs yield charge density values of 16.3 and $50.2 \mu\text{C cm}^{-2}$,
240 respectively, which corresponds, using the calibration equation (Equation 1) [17], to a charge
241 associated with (111) terrace sites of 25.5 and $78.4 \mu\text{C cm}^{-2}$.

242

$$243 \quad q_{Bi} = 0.64q_{\{111\}}^{t-1} \quad (1)$$

244

245 These values indicate that the surface of Pt/C NPs and Pt/C MWs have a ratio,
246 respectively, of 11.6 and 35.6 % of (111) ordered domains. These results indicate that NWs
247 morphology enhanced the existence of (111) terrace sites.

248 The monolayer CO stripping peaks have previously been investigated at single crystal
249 Pt surfaces, and their positions and shapes related to different symmetries and size domains

250 over the catalyst surfaces reported [26]. For example, Lai et al. [27] employed a combination
251 of electrochemical and theoretical techniques to study the CO and methanol electro-oxidation
252 on stepped single-crystal Pt and Rh electrodes. They studied the mechanism and kinetics of
253 CO electro-oxidation on Pt electrode on a number Pt[$n(111) \times (111)$] vicinal surfaces. For an
254 *fcc* metal, these surfaces are composed of terraces of (111) orientation, which are $(n - 1)$
255 atoms wide, separated by monoatomic steps of (110) orientation. In the CO oxidation study, a
256 pronounced catalysing effect of steps was observed. Higher step density lowers the
257 overpotential of CO oxidation with peak potentials difference for Pt(553), and Pt(111) equals
258 0.17 V between them. This effect was attributed to the preferential formation of O-containing
259 species at the steps sites compared to the terrace sites.

260 Fig. 5 shows the CO-monolayer stripping voltammograms. The hydrogen desorption
261 region (0.0–0.20 V *versus* RHE) is entirely suppressed on both catalysts, confirming the
262 saturation of the surfaces of the catalyst with adsorbed CO (CO_{ads}) [28]. The voltammogram
263 for the Pt/C NPs is in excellent agreement with those presented in the literature, for carbon-
264 supported Pt NPs recorded at 20 mV s^{-1} [29]. This catalyst shows CO oxidation only above
265 0.60 V with a single peak (peak III) centred at ~ 0.82 V. In contrast, the voltammogram for
266 the Pt/C NWs has three peaks, peak I at 0.45 V, peak II at 0.73 V, and peak III overlapping
267 with that of the Pt/C NPs.

268 The pre-peak (peak I) is commonly known as pre-ignition or pre-oxidation wave [30]
269 and has been the subject of many studies, with several explanations proposed, including (i)
270 the shift of bridge-bonded CO (CO_{b} , weakly adsorbed) to linear-bonded CO (CO_{L} , strongly
271 adsorbed), (ii) preferential oxidation of CO_{b} , (iii) oxidation of a weakly adsorbed or
272 kinetically unstable CO state, (iv) preferential oxidation at or near defect sites and (v) the
273 oxidation been driven by a rearrangement of CO-induced by the potential [31].

274 However, based on Farias and co-worker's publication [31], the occurrence of the CO
275 peroxidation peak is controlled by four interconnected and simultaneous conditions, such as
276 CO adsorption at potentials lower than about 0.2 V; on surfaces saturated with CO_{ads}; in the
277 presence of both, traces of CO solution and surface steps. If any of these four conditions is
278 not satisfied, the CO preoxidation pathway does not appear. For example, in Farias et al.'s
279 paper, they compare the CO oxidation in the presence and absence of small amounts of
280 dissolved CO in the electrolyte. In these experiments, the potential was fixed at 0.1 V, and
281 CO was bubbled into the solution for 5 min to cover the electrode surface fully. After that, Ar
282 was bubbled into the solution for 5, 10, and 60 min. Finally, when the solution was purged
283 for 60 min, the CO peroxidation signal became utterly absent. Thus, based on our
284 experimental setup, were CO adsorption was hold at 0.05 V for 5 min, and N₂ was bubbled
285 for 25 min; we believed that these four conditions are satisfied.

286 The sharp peak II in Fig. 5 for the Pt/C NWs catalyst on 0.73 V is comparable with
287 the "butterfly" peak, which is observed in CO oxidation in perchloric solution [32]. This peak
288 appears in 0.1 mol L⁻¹ HClO₄ solution and corresponds to the reversible adsorption of OH
289 (OH_{ads}). In this sense, this peak, in 0.5 mol L⁻¹ H₂SO₄ solution, may be attributed to a filling-
290 in process by OH_{ads} on the Pt defects in the closed-packed adlayer [33]. However, the
291 formation of a full OH_{ads} adlayer on the H₂SO₄ solution is inhibited due to the strong
292 adsorption [34].

293 For instance, the effect of Pt agglomeration on CO stripping must be considered.
294 According to Mailard et al. [35], platinum agglomeration can cause peak II, observed for the
295 Pt/C NWs catalyst. Also, according to Urchaga et al. [36], the multiple peaks on CO stripping
296 could be related to the presence of different crystal facets at the Pt surface. It was indicated
297 that larger particles have a higher number of exposed faces Pt(111) on their surface than in
298 small Pt nanoparticles. It is also known that the Pt(111) face is more thermodynamically

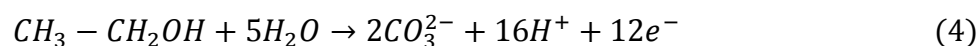
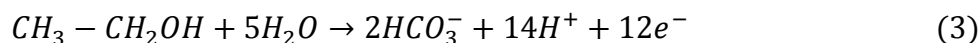
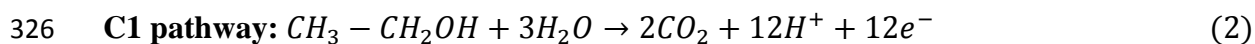
299 stable and, therefore, it would be the most preferably exposed surface in the case where the
300 particles agglomerate [37]. Thus, it is likely that peak II for the Pt/C NWs catalyst is due to
301 the oxidation of weakly adsorbed CO on Pt(111) faces, present in large quantities in both
302 agglomerated Pt particles and nanowires, compared to isolated Pt nanoparticles. This fact
303 becomes evident from the CO stripping profile (Fig. 5) for the catalyst with nanoparticle
304 morphology, Pt/C NPs, in which peak II is absent. Thus, the Peak III present in both
305 catalysts, centred at 0.82 V and 0.83 V for Pt/C NPs and Pt/C NWs, respectively, can be
306 attributed to the oxidation of CO adsorbed in isolated Pt nanoparticles.

307 Fig. 6 shows the cyclic voltammograms for EOR obtained at room temperature and 20
308 mV s^{-1} scan rate. Both catalysts presented the characteristic profile of the ethanol oxidation in
309 acidic media on carbon-supported Pt. The hydrogen adsorption region was fully inhibited,
310 indicating strong adsorption of ethanol molecules at the catalysts surface active sites at the
311 initial potential of 0.05 V. Significant voltammetric changes were observed with the
312 morphology change from NPs to NWs. The Pt/C NPs catalyst showed smaller EOR currents
313 at all potentials, with an onset potential of ~ 0.73 V, chosen as the potential at which the
314 current density was 5 mA cm^{-2} . In contrast, the synthesised Pt NWs catalyst displayed
315 approximately 5-fold greater EOR current densities, with a much lower ethanol oxidation
316 onset potential (~ 0.57 V). The negatively shifted onset potential is comparable with that
317 reported for state-of-art bimetallic catalysts [38].

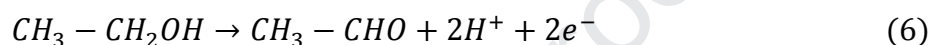
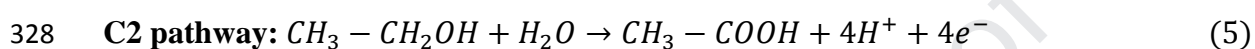
318 Fig. 7 shows the first derivatives of the anodic sweep for Pt/C NPs and Pt/C catalysts,
319 using the method described by Murthy and Manthiram [39]. Some insights regarding the
320 ethanol oxidation mechanism can be obtained by using derivative voltammetry; however, up
321 to now, a detailed ethanol oxidation mechanism remains unclear, since many parallel and
322 consecutive reactions can happen, resulting in more than forty possible volatile and adsorbed

323 species [40,41]. Currently, a dual-pathway mechanism on Pt catalysts is broadly accepted for
 324 either acid or alkaline environments: [14,42]

325



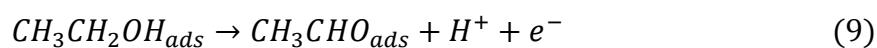
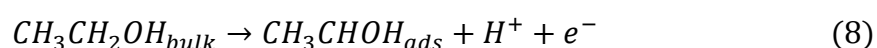
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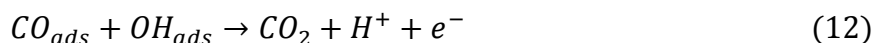


329

330 The C1 pathway leads to the complete ethanol oxidation to CO_2 through the CO_{ads}
 331 intermediate delivering $12 e^-$. Hence, the C2 pathway represents the partial oxidation of
 332 ethanol to acetate or acetic acid (both delivering $4 e^-$) and to acetaldehyde (delivering $2 e^-$)
 333 without the C–C bond cleavage [43]. At low potential ($< 0.4 V$), the ethanol oxidation starts
 334 with the adsorption of ethanol at the Pt surface (equation 7). Furthermore, adsorbed ethanol
 335 can suffer oxidation to acetaldehyde or CO (derived from the dissociative adsorption,
 336 equations 8 to 12). Besides, reaction 5 occurs preferentially at potentials higher than $0.8 V$, in
 337 which water molecule is activated, forming O-containing species. Reaction 6 illustrates
 338 acetaldehyde production. This reaction happens preferentially at potentials below $0.6 V$,
 339 where the dissociative adsorption of water occurs (reaction not shown) [44].

340





341

342 The derivative curves in Fig. 7 display two positive peaks, which are more defined for
 343 the Pt/C NWs catalyst. The electrocatalytic behaviour is similar for both catalysts between
 344 0.2 and 0.4 V, which is the region where dissociative adsorption and the subsequent hydrogen
 345 removal occurs, resulting in CO_{ads} . Then, the derivative current starts to grow (for both
 346 catalysts) and stops at around 0.82 V (Peak II). In this region, the oxidation of CO_{ads}
 347 (equation 12) to CO_2 can be the rate-determining step (*rds*) for the catalysts. Peak I in Figure
 348 7 (~ 0.73 V) is related to EOR through the C1 pathway (total oxidation); consequently, the
 349 higher this peak, the greater the amount of ethanol being oxidised. If this pathway did not
 350 happen, this peak would not exist (dashed arrow); hence, the C2 pathway would be
 351 dominating.

352 The Pt/C NWs catalyst shows the highest peak I, which can be associated with the
 353 ethanol's adsorption on the exposed crystal planes or defects of the NWs. The further
 354 increase of the potential around 0.83 V leads to more coverage of Pt sites with OH, blocking
 355 the ethanol adsorption and consequently decreasing the EOR, resulting in the third peak at
 356 both catalysts.

357 The Pt/C NPs catalyst showed the lowest ethanol oxidation, which can be explained
 358 by the high affinity of Pt for intermediary carbonaceous species, evidenced by the higher
 359 amount of LCAs. The derivative current reaches its minimum value at ~0.96 V for both
 360 catalysts, since, at this point, OH species are strongly adsorbed at the catalyst's sites,
 361 blocking the adsorption of ethanol from the bulk solution. In Fig.7, the ethanol oxidation
 362 onset potential for Pt/C NPs and Pt/C NWs catalysts are 0.67 and 0.60 V, respectively, which
 363 are very close to the values obtained at cyclic voltammetry tests (Fig. 6).

364 Chronoamperometric curves in Fig. S2 shows that the Pt/C NWs catalyst displays the
 365 highest anodic current density ($0.0062 \text{ mA cm}^{-2}$) that is more than 5-fold higher the current
 366 density presented by the commercial Pt/C NPs ($0.0012 \text{ mA cm}^{-2}$) at the end of the
 367 experiment. Those data agree with the results in Fig. 6, indicating a better EOR and less
 368 coverage of the surface by intermediate by-products for the synthesised catalyst, which
 369 derives from the unique NW morphology.

370 The polarisation curves (Fig. S3) confirms the trend in both Figs. 5 and 7, indicating
 371 enhanced ethanol oxidation activity for the synthesised Pt/C NWs. The change in catalysts
 372 the morphology (NPs to NWs) shifts the oxidation onset potential to more negative values
 373 (Table 1).

374

375 **Table 1.** CO and ethanol oxidation onset potential and Tafel slopes for all catalysts
 376 synthesised in this study

Catalyst	CO oxidation E_{onset} [V]	Ethanol oxidation E_{onset} [V] ^a	Current density at 900 s [mA cm^{-2}] ^b	Tafel slopes [mV dec^{-1}]	Mass activity ($\text{A mg}^{-1}_{\text{Pt}}$) @ [0.8 V <i>versus</i> RHE ^a]
Pt/C NPs	0.77	0.41	0.0012	180.32	0.007
Pt/C NWs	0.68	0.36	0.0062	134.15	0.013

377 ^a Values obtained by Tafel plots.

378 ^b Values obtained by chronoamperometric experiment.

379

380 The Tafel slopes were 180.32 and $134.15 \text{ mV dec}^{-1}$ for Pt/C NPs and Pt/C NWs
 381 catalyst, respectively, indicating that different rate-determining steps (*rds*) can be happening
 382 for both catalysts [20,45]. Tafel slopes around 120 mV dec^{-1} have been obtained on single-

383 crystal carbon-supported Pt catalysts. Generally, the Tafel slope allows the observation of
 384 whether the *rds* is the transfer of the first or second electron and if a chemical reaction is
 385 involved [46]. Thus, for the synthesised Pt/C NWs catalyst, the obtained slope value (close to
 386 120.0 mV dec⁻¹) is in agreement with a mechanism that involves Temkin-type adsorption for
 387 both OH_{ads} and ethoxy at low potentials [47]. For the commercial Pt/C NPs catalyst, the value
 388 of 180.32 mV dec⁻¹ indicates that the *rds* is controlled by the ethanol dehydrogenation or
 389 acetic acid formation or a combination of both [48]. Further, different Tafel slopes can also
 390 be obtained depending on the distribution of by-products [49].

391 FTIR experiments were carried out to understand the formation of products and
 392 intermediates and the effect of the catalyst morphology on the EOR. The main functional
 393 groups related to ethanol oxidation products are listed in Table 2.

394
 395 **Table 2:** Assignment of the bands observed in the spectra obtained by FTIR

Wavenumber (cm ⁻¹)	Functional group or chemical species	Deformations	References
2980, 2903	CH ₃ , CH ₂	C–H, stretch	[50]
2621	COOH	O–H, stretch	[50]
2343	CO ₂	O=C=O, asymmetrical stretch	[51]
2050	CO _L	–C–O, stretch	[52]
1720	COOH ou CHO	–C=O, carbonyl stretch	[51]
1650	H ₂ O	H–O–H, angular deformation	[53]
1400	Acetate	–O–C–O, stretch	[54]
1367	Acetaldehyde	–CH ₃ , symmetrical stretch	[55, 56]
1280	Acetic acid	–C=O, stretch	[51]
1200	HSO ₄ ²⁻	S–O, symmetrical and asymmetrical stretch	[57]
1113	Acetaldehyde	C–H, vibration	[56]
1045	Ethanol	C–O, axial deformation	[58]

396

397 Fig. 8 shows the *in situ* FTIR reflectance spectra acquired over a potential range of
398 0.05–1.0 V *versus* RHE, in which the spectra were obtained every 0.05 V in H₂SO₄ 0.5 mol
399 L⁻¹ + ethanol 0.5 mol L⁻¹ solution on the Pt/C NPs and Pt/C NWs catalysts. Pointing down
400 bands (negative bands), indicate the species formation between the thin layer of the working
401 electrode and the CaF₂ window, on the other hand, pointing upwards bands (positive bands),
402 represent the consumption/decrease of another species. The presence of CO₂, acetaldehyde,
403 and acetic acid was observed for both catalysts, indicating that both pathways are present (C1
404 and C2 pathway).

405 The upward band located at 1650 cm⁻¹ is assigned to the bending mode H–O–H of
406 water [49]. This band is more intense for the Pt/C NWs catalyst (Fig. 8a) than for the Pt/C
407 NPs catalyst. A more intense positive water band for Pt/C NWs can implicate that more water
408 is consumed or that more water is leaving the thin layer region. Water is oxygen donor
409 species, necessary for alcohols oxidation to CO₂, also competes with the organic molecules
410 for adsorption sites [55]. In this sense, as the Pt/C NWs catalyst showed the highest
411 intensities in the CO₂ bands, more considerable amounts of water are needed to complete the
412 reaction, which may justify the higher positive water bands for the catalyst with nanowire
413 morphology. Parallel to this, as the by-products are formed, water is expelled from the thin
414 layer to permit the diffusion of the new products generated. Since the Pt/C NWs catalyst
415 generated the largest bands of CO, CO₂, and CH₃COOH, a higher amount of water is
416 expelled from the thin layer.

417 Three upwards bands detected at 1045 cm⁻¹, 2903 cm⁻¹, and 2980 cm⁻¹ are ascribed to
418 axial deformation of C–O, and ethanol asymmetric CH₂, and CH₃ vibrations, respectively [59]
419 indicates the consumption of ethanol from the thin layer.

420 The negative band at 2343 cm^{-1} is attributed to asymmetrical stretch $\text{O}=\text{C}=\text{O}$ of CO_2 .
421 The first band of CO_2 appears from 0.75 V for Pt/C NPs and 50 mV earlier for Pt/C NWs,
422 which are in good agreement with CO monolayer oxidation (Fig. 4), and the signal of the
423 band increases with increasing potential. It is worth to note that the band at 2343 cm^{-1} for
424 Pt/C NWs catalyst (Fig. 7b) is much more intense than Pt/C NPs catalyst, indicating that a
425 more significant amount of CO_2 may be produced. As previously mentioned, the existence of
426 extended terraces or defects on the NW catalyst could result in enhanced mobility of OH_{ads}
427 and CO_{ads} , facilitating the CO oxidation from the catalyst surface [60].

428 The band at 2050 cm^{-1} is attributed to the C–O stretch of linearly adsorbed CO (CO_L)
429 [52]. CO_L band was initially detected at 0.45 V for Pt/C NPs and 0.4 V for Pt/C NWs. The
430 appearance of these bands indicates that the breaking of the C–C bond of the ethanol
431 molecule occurs between 0.05 and 0.45 V , resulting in the formation and accumulation of
432 adsorbed CO. The intensity of this band reaches the maximum between 0.6 and 0.65 V for
433 both catalysts. Subsequent decay of CO_L band intensity may be a consequence of both
434 increased oxidation of adsorbed CO and decreased CO_L formation. However, for Pt/C NWs
435 the CO_L band intensity is still present up to 1.0 V , which differs from data obtained during
436 CO stripping measurements (Fig. 5), where all CO molecules have been oxidised to CO_2 at
437 1.0 V in both catalysts. These discrepancies will be explained further below.

438 The band located at 1720 cm^{-1} is associated with the stretching vibration of the
439 carbonyl group C=O [51] present in acetic acid and acetaldehyde. For the Pt/C NPs catalyst,
440 the first band was observed at 0.75 V , and for Pt/C NWs catalyst, the band was identified at
441 0.65 V . From these potentials, the band continuously increases until reaching 1.0 V and the
442 formation of intermediates without the C–C bond breaking seems to be predominant [61].

443 Acetic acid formation becomes evident from the bands located at 2621 and 1280 cm^{-1} ,
444 attributed to the $-\text{OH}$ stretching deformation and $-\text{C}=\text{O}$ stretching groups, respectively. These

445 bands appear only at high potentials, 0.65 V for Pt/C NWs, and higher than 0.75 V for Pt/C
446 NPs. Another band located at 1400 cm^{-1} at high potential values was identified as the
447 stretching of adsorbed acetate ($-\text{O}-\text{C}-\text{O}$), indicating that acetate can be an intermediate of
448 ethanol oxidation reaction [53].

449 The bands at 1367 and 1113 cm^{-1} are associated with symmetric deformation ($-\text{CH}_3$)
450 and vibration ($\text{C}-\text{H}$), respectively, of acetaldehyde [55]. The 1113 cm^{-1} band was initially
451 observed at 0.65 V for Pt/C NPs and 0.70 V for Pt/C NWs, and its intensity continuously
452 increased to 1.0 V for both catalysts. Acetaldehyde formation occurs first, then the formation
453 of acetic acid, which confirms that acetaldehyde is an intermediate for the production of
454 acetic acid, as previously demonstrated by Iwasita et al. [62] and Weaver et al. [63].

455 For all materials, it is possible to observe the stretching bands of the HSO_4^- ion at
456 1200 cm^{-1} , due to the H_2SO_4 used as the electrolyte [53]. The HSO_4^- band is due to its
457 accumulation in the thin layer in order to compensate for the positive increase of electrode
458 potential [55].

459 For a more detailed analysis of infrared spectra, the absorption area of the bands of
460 CO (2050 cm^{-1}), CO_2 (2343 cm^{-1}), and acetic acid (1280 cm^{-1}) was integrated. After
461 integration, these areas were normalised by the electroactive area of each electrode obtained
462 during the CO oxidation experiments (Fig. 5). The dependences of the integrated band
463 intensities of Pt/C NPs and Pt/C NWs catalysts on the potential are shown in Fig. 8. It should
464 be noted that these graphs indicate at which potential the ethanol oxidation reaction products
465 are generated, they do not allow the quantification of these products. However, the results
466 suggest that CO, CO_2 , and CH_3COOH production for Pt/C NWs is much higher than Pt/C
467 NPs, which can explain the increased oxidation current density obtained during the cyclic
468 voltammograms (Fig. 6).

469 Fig. 9a shows the behaviour of the intensity of integrated CO absorption bands. The
470 CO band intensities for the catalysts increases as the potential increases up to 0.6 and 0.65 V
471 for Pt/C NPs and Pt/C NWs, respectively. At potentials higher than 0.6 V for Pt/C NPs and
472 0.65 V for Pt/C NWs, the CO_L band starts to decrease. For the commercial Pt/C NPs, the CO_L
473 band is no longer visible in the spectra at potentials higher than 0.85 V. These outcomes may
474 indicate the total elimination of adsorbed CO or a slow rate of C–C bond breaking. However,
475 this band is still present for the Pt/C NWs until 1.0 V. Del Colle *et al.* [64] suggested that the
476 presence of the CO_L band at high potentials means that the CO is being produced at a rate
477 higher than the oxidation rate and, consequently, some CO remains on the surface.
478 Nevertheless, it seems more plausible to assume that some CO is present from both, the
479 reference (0.05 V) and the sampling potentials (0.1–1.0 V), thus generating a bipolar band
480 with some contribution of the CO_L bands at higher potentials [64], thereby justifying the
481 presence of CO_L bands at potentials as high as 1.0 V.

482 Note in Fig. 9b that the onset potential for CO₂ production was 0.7 V for both
483 catalysts. However, the production of CO₂ was higher on Pt/C NWs catalyst than on
484 commercial Pt/C NP catalyst in all potentials studied (Fig. 9b). This fact may be due to the
485 existence of extended terraces or defects on NWs morphology [60]. Moreover, a surface
486 contraction may occur for the Pt nanowires. According to Nørskov and Hammer's theory, [65,
487 66] a surface contraction strain can cause a down-shift of the d-band centre of Pt, which will
488 decrease the absorption strength of reactants. Wang *et al.* [67] reported a down shift of the Pt
489 4f d-band for Pt NWs compared to Pt NPs supported on carbon. Thus, this structure-
490 dependent electronic change of the Pt/C NWs may improve the CO surface diffusion rate due
491 to the weaker binding of CO, leading to an enhanced oxidation rate.

492 Another comment about the effect of the morphology of the nanowire is related to
493 O/OH adsorption. At high potentials, there is a competition on the catalyst surface between

494 O/OH and ethanol adsorption. Similar to the CO poisoning effect, the strong interaction of
495 O/OH and the surface of the catalyst could block the active sites and decrease the ethanol
496 oxidation rate. Therefore, a surface with a weaker OH binding strength can provide more
497 sites for ethanol oxidation [55]. Observing the displacement, for higher potentials, of the Pt–
498 O reduction peak for the nanowire morphology catalyst (Fig. 3), the lower binding energy
499 between Pt and the oxygenated species can be suggested for the nanowire morphology
500 catalyst, justifying the higher catalytic activity compared to the catalyst with nanoparticle
501 morphology.

502 Acetic acid has been formed at potentials higher than 0.6 V in both Pt/C NPs and Pt/C
503 NWs catalysts. For potentials higher than 0.75 V, the acetic acid bands for Pt/C NWs were
504 higher compared to commercial Pt/C NPs. Previous results [68] indicated that the (111) plane
505 is very active in the formation of acetic acid, and an increase in the <111> direction for Pt/C
506 NWs could explain the higher production of acetic acid.

507 In summary, from FTIR data, it was possible to observe that the C1 pathway, with the
508 C–C bond break to CO formation, is dominant in both surfaces, Pt/C NWs and Pt/C NPs, at
509 low potentials, lower than 0.6 V. However, at higher potentials, the production of acetic acid
510 and acetaldehyde took place and the C2 pathway which represents the partial oxidation of
511 ethanol, maintaining the C–C bond intact, is the main path.

512

513 3. Conclusions

514 This paper describes the synthesis of carbon-supported Pt NWs using a simple, low cost,
515 template-free, and surfactant-less chemical reduction of metallic precursors method by formic
516 acid. TEM images confirmed the formation of NWs of approximately 20 nm in length and 4
517 nm in diameter, confirming the success of the applied methodology for the synthesis.

518 The synthesised NWs presented improved catalytic activity compared with the
519 commercial Pt/C Johnson Matthey catalyst, showing higher oxidation current density, as well
520 as, lower onset potentials for ethanol and CO-monolayer oxidation, attributed to the presence
521 of (110) step sites on (111) terraces. The oxidation current density was more than 5-fold
522 higher than the commercial NPs catalyst for EOR measured at 0.90 V *versus* RHE. FTIR
523 results revealed the presence of CO, CO₂, acetic acid, and acetaldehyde for all catalysts,
524 demonstrating that both routes C1 and C2 are active for these catalysts. However, the
525 nanowires produced more significant quantities of CO₂.

526 The results confirm the feasibility of the synthesis method and the importance of the
527 nanowire morphology in the removal of intermediate species from the catalyst surface,
528 producing efficient catalysts to be applied as anodes for DEFCs.

529

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536

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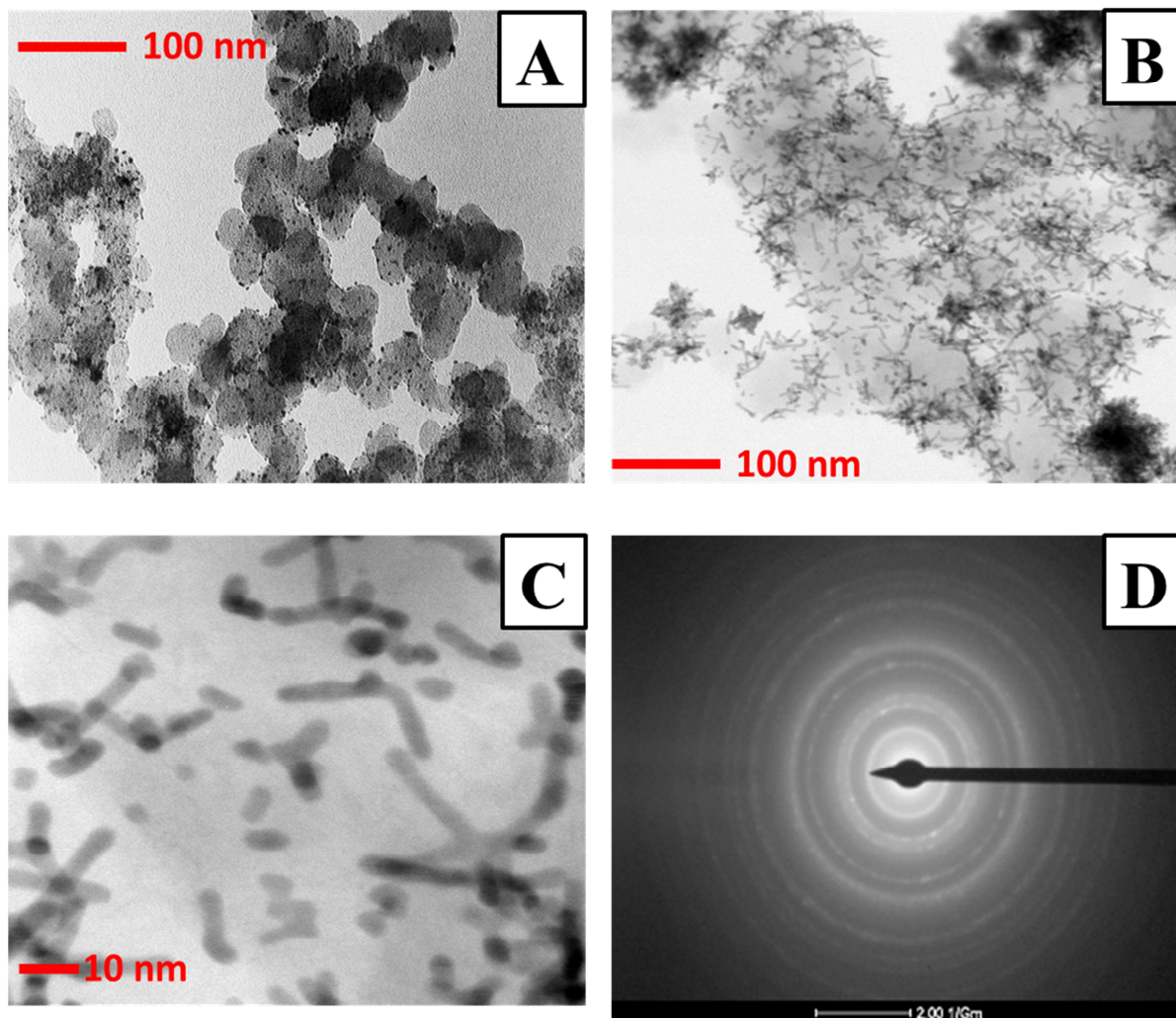
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721

Figures



722

723 **Figure 1.** TEM micrographs obtained for (A) Pt/C NPs, and (B, C) Pt/C NWs catalysts in different

724 magnifications, as well as (D) SAED pattern obtained for the Pt/C NWs catalyst.

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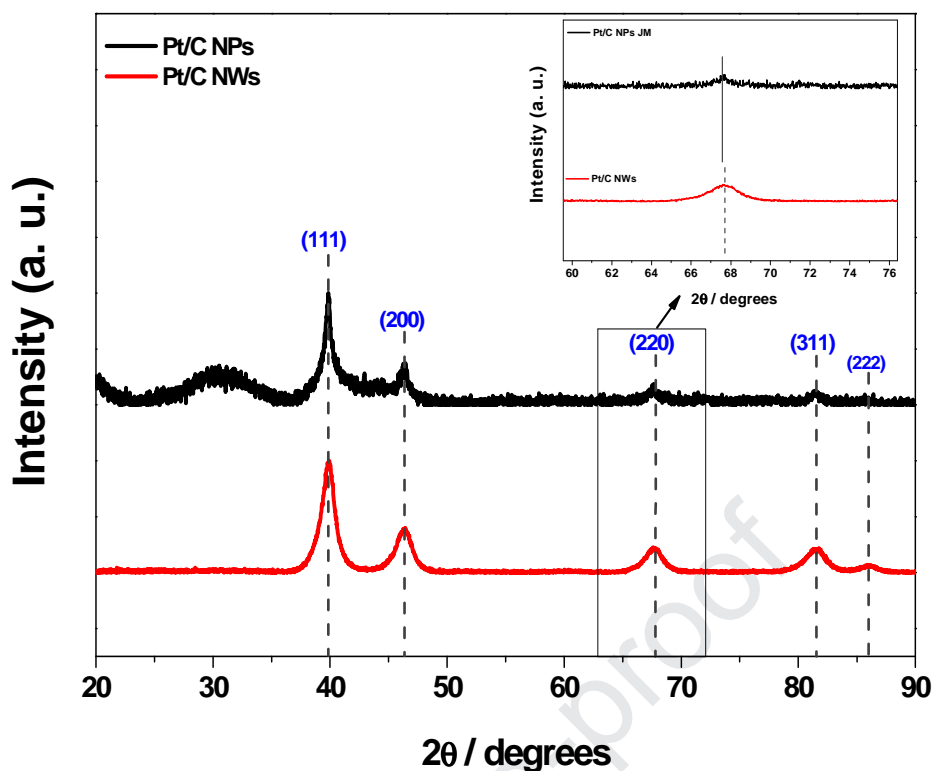


Figure 2. XRD patterns of the Pt/C NPs Johnson Matthey and Pt/C NWs catalysts. The inset is the enlarged Pt(220) diffraction peak for both catalysts.

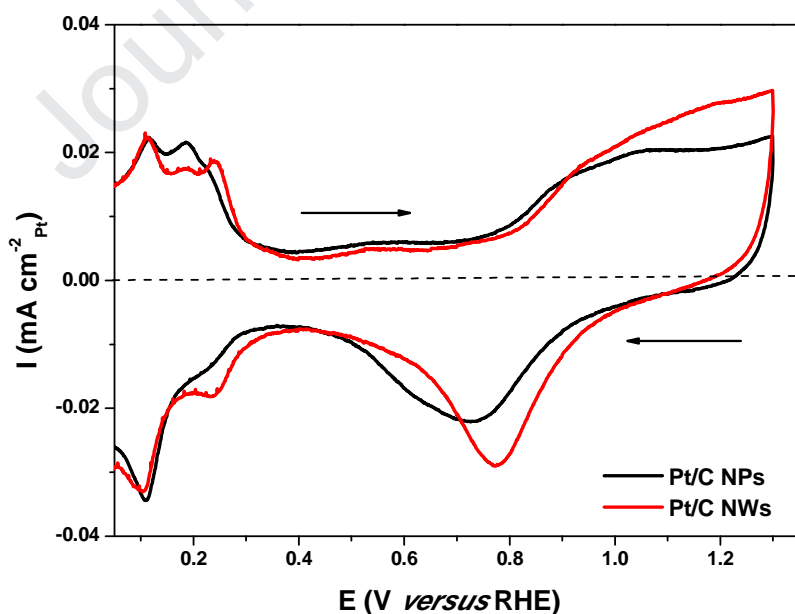
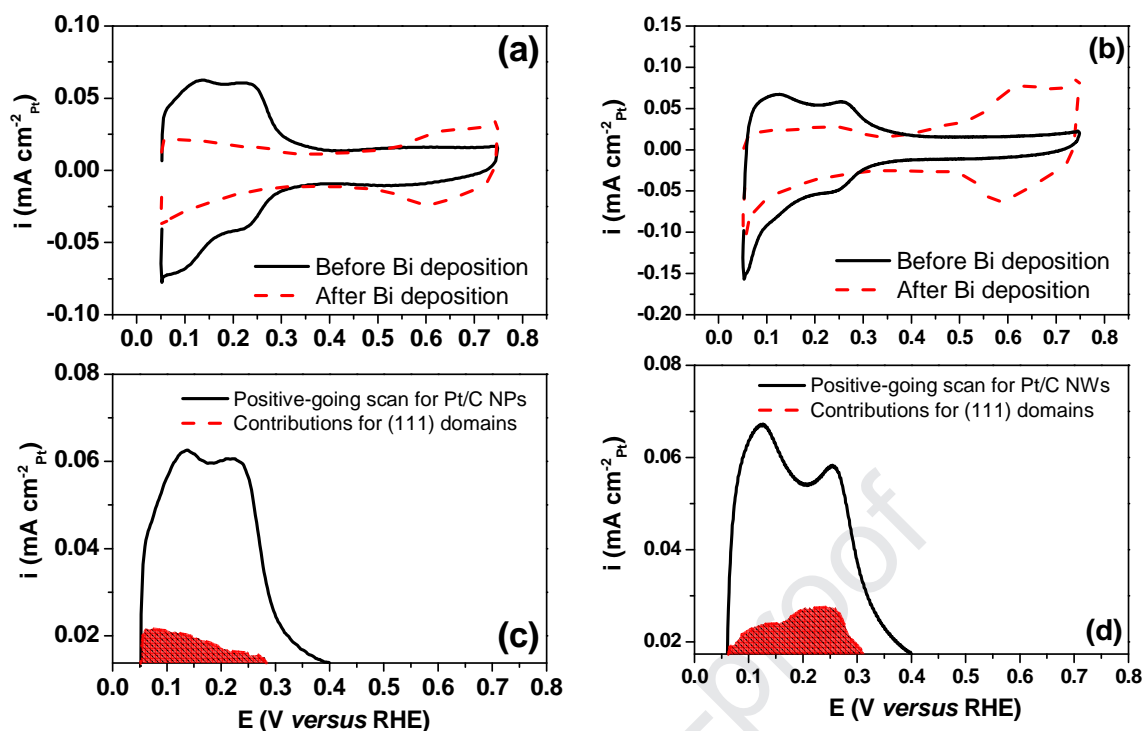


Figure 3: Cyclic voltammograms (second cycle) recorded in 0.5 mol L⁻¹ H₂SO₄ for the Pt/C NPs Johnson Matthey and Pt/C NWs catalysts. $v = 20 \text{ mV s}^{-1}$ and $T = 25 \text{ }^\circ\text{C}$.



750

751 **Figure 4:** Voltammetric profiles before (black solid line) and after Bi deposition (red dashed

752 line) for (a) Pt/C NPs and (b) Pt/C NWs. Positive going scan for (c) Pt/C NPs and (d) Pt/C

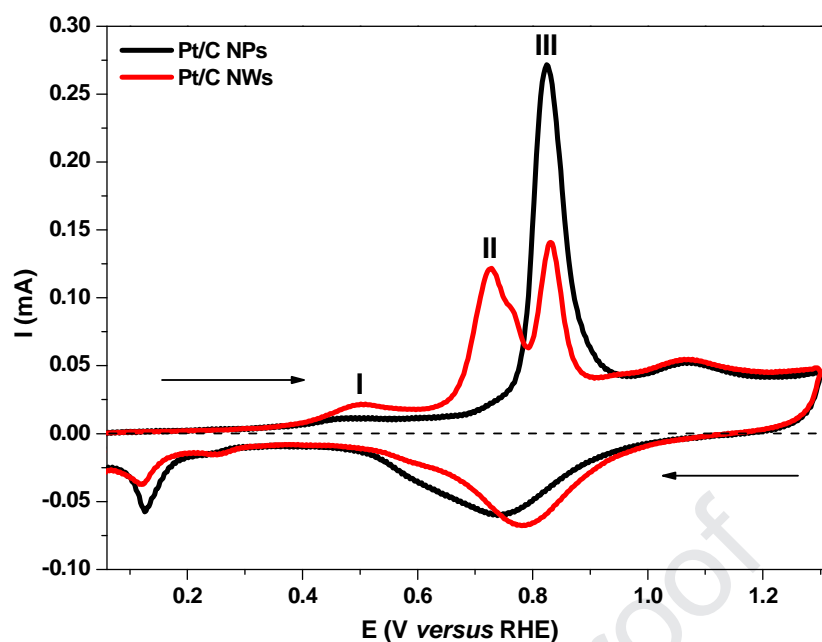
753 NWs in the hydrogen adsorption/desorption region after double-layer subtraction in 0.5 mol

754 L⁻¹ H₂SO₄ and a scan rate of 50 mV s⁻¹. The shadowed area corresponds to the hydrogen

755 contribution of the Pt atoms that were not covered by Bi.

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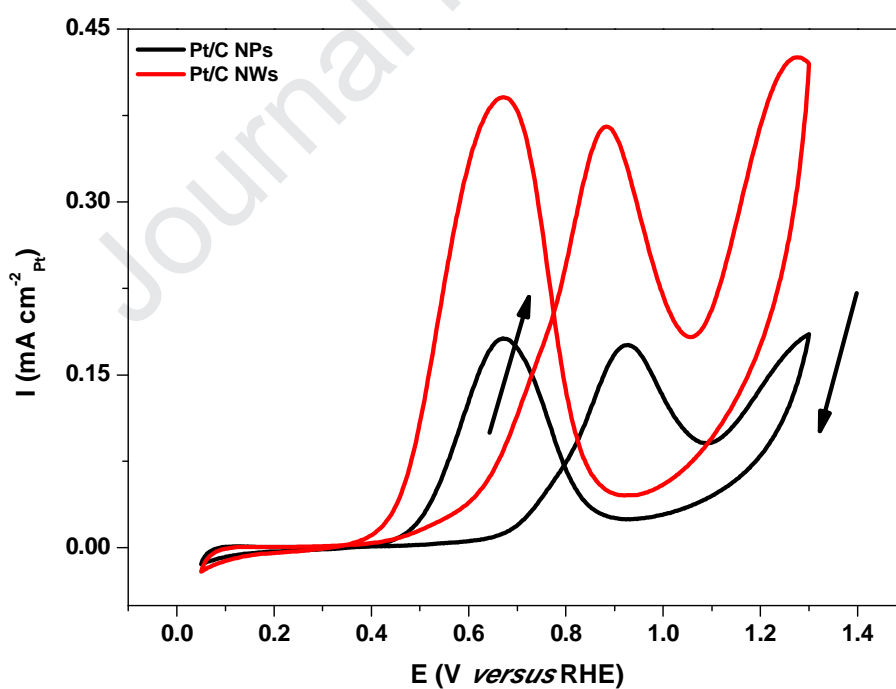
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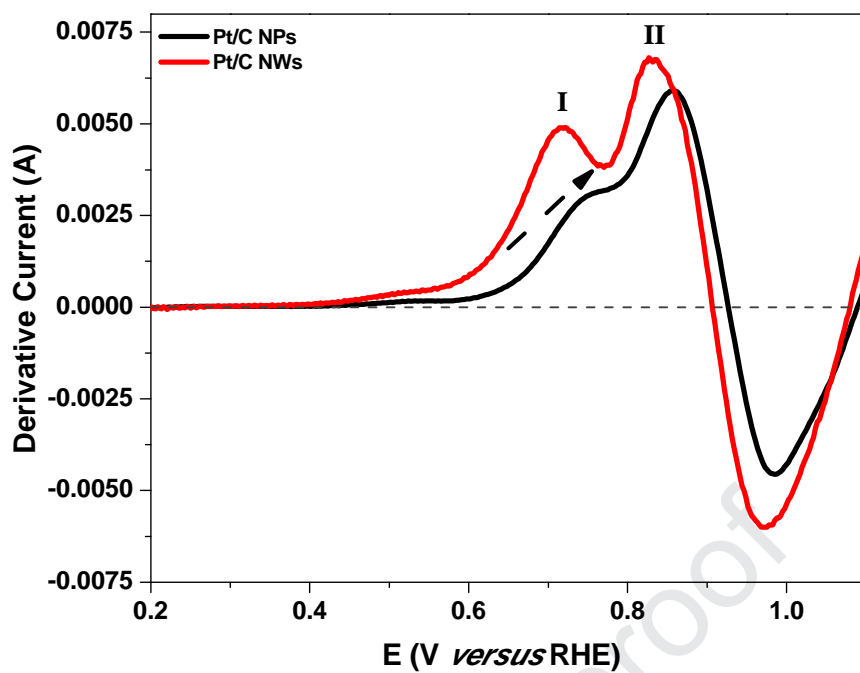
759 **Figure 5.** CO-monolayer stripping voltammetry (first scan) on Pt/C NPs Johnson Matthey
760 and Pt/C NWs catalysts in $0.5 \text{ mol L}^{-1} \text{ H}_2\text{SO}_4$ at 20 mV s^{-1} .

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762

763 **Figure 6.** Cyclic voltammograms (second cycle) for the electrochemical oxidation of ethanol
764 0.5 mol L^{-1} dissolved in H_2SO_4 for Pt/C NPs Johnson Matthey and Pt/C NWs catalysts. $v =$
765 20 mV s^{-1} and $T = 25 \text{ }^\circ\text{C}$.



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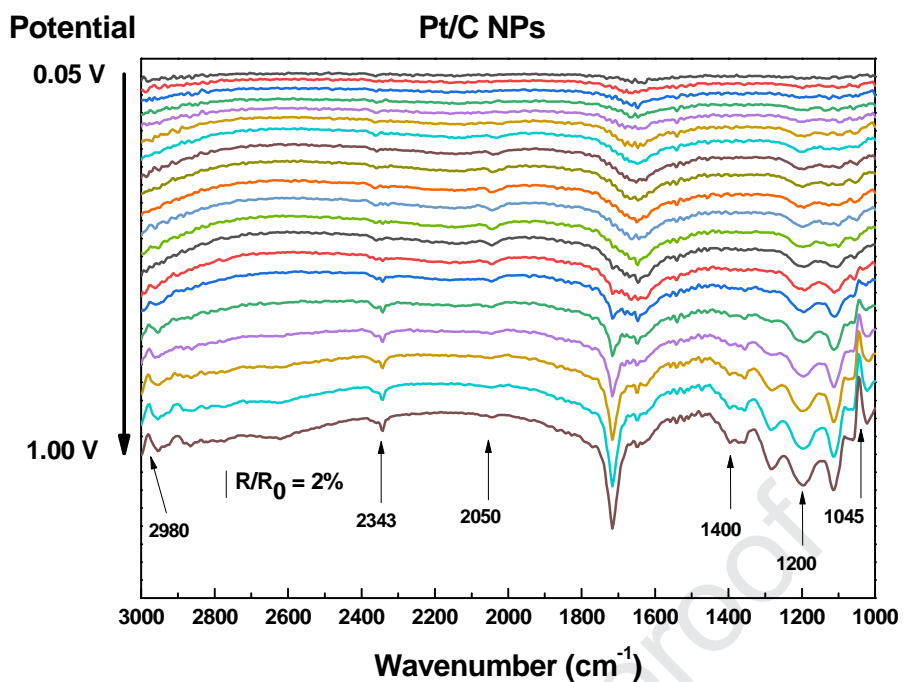
767 **Figure 7.** First derivative voltammograms of the anodic scans for ethanol oxidation on Pt/C768 NPs Johnson Matthey and Pt/C NWs catalysts obtained in $0.5 \text{ mol L}^{-1} \text{ H}_2\text{SO}_4 + 0.5 \text{ mol L}^{-1}$ 769 ethanol at 20 mV s^{-1} .

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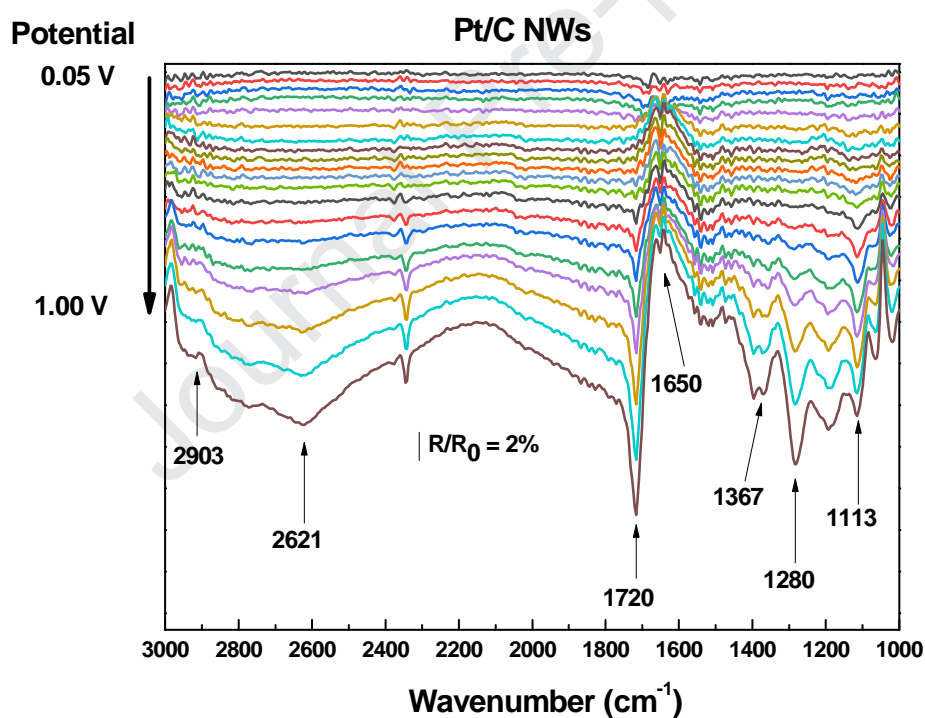
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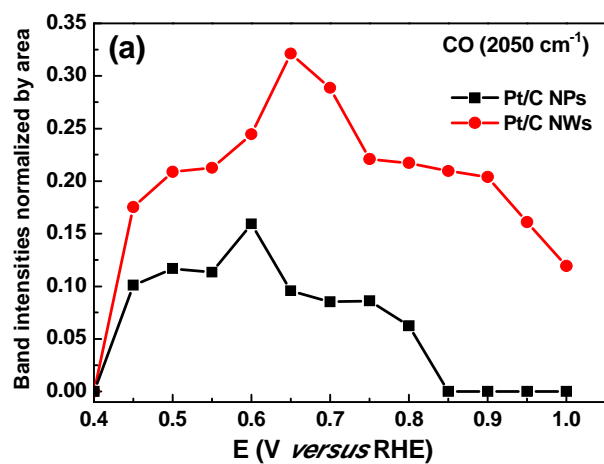
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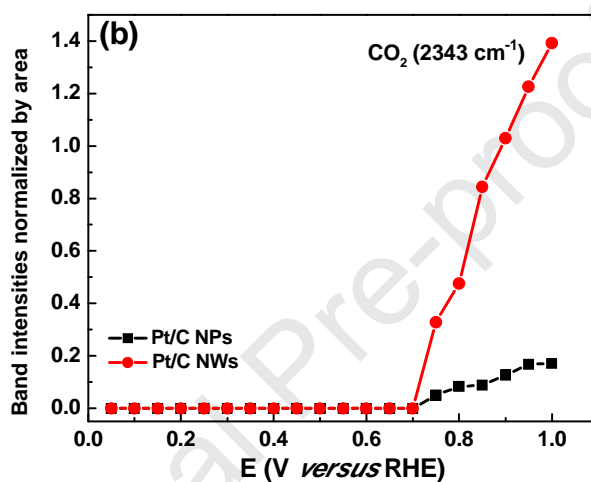
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776 **Figure 8:** In situ FTIR spectra obtained during ethanol electrochemical oxidation on Pt/C
 777 NPs and Pt/C NWs catalysts, after 128 scans, H_2SO_4 0.5 mol L^{-1} + ethanol 0.5 mol L^{-1}
 778 solution. Reference spectra collected at 0.05 V during chronoamperometry steps and 4 cm^{-1}
 779 resolution. Potentials and compositions are indicated inside the figure.

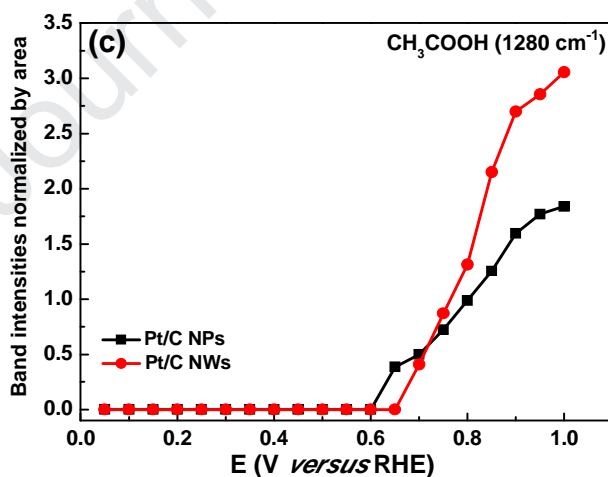
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784 **Figure 9:** Intensity integration of (a) CO (2050 cm^{-1}), (b) CO₂ (2343 cm^{-1}), and (c)
 785 CH₃COOH (1280 cm^{-1}) bands obtained from the FTIR spectra of the Pt/C NPs and Pt/C NWs
 786 catalysts.

787

Highlights

- Pt/C nanowires were made without using surfactants, templates or stabilising agents
- The existence of extended terraces in NW catalysts improves ethanol electro-oxidation
- The current density for Pt/C NWs was 5-fold higher than the commercial NPs catalyst
- The nanowires morphology produces more significant quantities of CO₂ and acetic acid

Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:



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