

Er:Ta₂O₅ waveguide optimization & spectroscopy

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Abstract. The optimization of erbium-doped Ta₂O₅ thin film waveguides deposited by magnetron sputtering is described. Background losses below 0.4dB/cm have been obtained before post-annealing. A broad photoluminescence spectrum centered at 1534nm is obtained, and the photoluminescence power and fluorescence lifetime increase with post-annealing, yielding promising results for compact amplifiers.

1. Introduction

Optical amplifiers are key components in optical telecommunications and in fully-integrated optical systems. Erbium doped materials are of particular importance in optical communications technology, due to their excellent performance as gain media for amplifiers and lasers at the telecommunications wavelength of 1.5 microns. Low-cost, compact erbium doped waveguide amplifiers (EDWAs) are essential for local-loop optical systems, and fully-functional densely integrated planar lightwave circuits (PLCs) will rely upon gain as electronic integrated circuits do at present. Erbium-doped high index contrast materials have generated great interest [1-4], and will allow strong confinement of light, ultra compact photonic devices, and non-linear processes at moderate power levels. Tantala (Ta₂O₅) has already been used as a host for rare earth ions [5-7], with lasing being achieved only in Nd:Ta₂O₅ to date [5]. This, combined with high refractive index (>2.0), moderate phonon energy for high radiative efficiency [8], a large third order non-linearity [9], and high photosensitivity [10], makes it an ideal material for realising multifunctional PLCs. In this paper, the deposition and optimisation of erbium-doped Ta₂O₅ (Er:Ta₂O₅) thin films using magnetron sputtering is presented. The photoluminescence and fluorescence lifetime characterisation of these Er:Ta₂O₅ films annealed at different temperatures are also presented to evaluate this material's potential as a high index contrast host for erbium and as an EDWA.

2. Waveguide fabrication & characterization

Slab waveguides were fabricated by magnetron sputter deposition of a powder pressed, Er:Ta₂O₅ target onto an oxidised silicon substrate (oxide thickness ~ 2.1 μ m). The target was doped with 1 wt. % of Er₂O₃ (~2.5 x 10⁻²⁰ ions/cm³). The deposition was carried out in a vacuum chamber pumped to a base pressure of 10⁻⁸ Torr and backfilled with an Ar:O₂ ambient. In order to obtain high-quality as-deposited films, substrate temperature, magnetron power and O₂ gas flow rate were optimised to yield low-loss films. The deposition rate was determined by measuring the film thickness for various sputtering times, using a stylus profilometer. Figure 1a shows the thickness plotted against time, for the deposition conditions optimised below, with the average deposition rate found to be ~ 2 nm.min⁻¹.

The slab waveguide losses were measured at 633nm (He-Ne laser) by prism coupling and measuring the propagation decay length. The losses were estimated at several different places on a sample to check the homogeneity of the sample, and averaged. The optical loss variation with substrate temperature is shown in Figure 1b. The sample sputtered at 200°C gave the lowest losses (~0.4 dB/cm), so this was chosen as the

optimum substrate temperature. Losses are expected to be substantially lower at wavelengths near 1550nm.

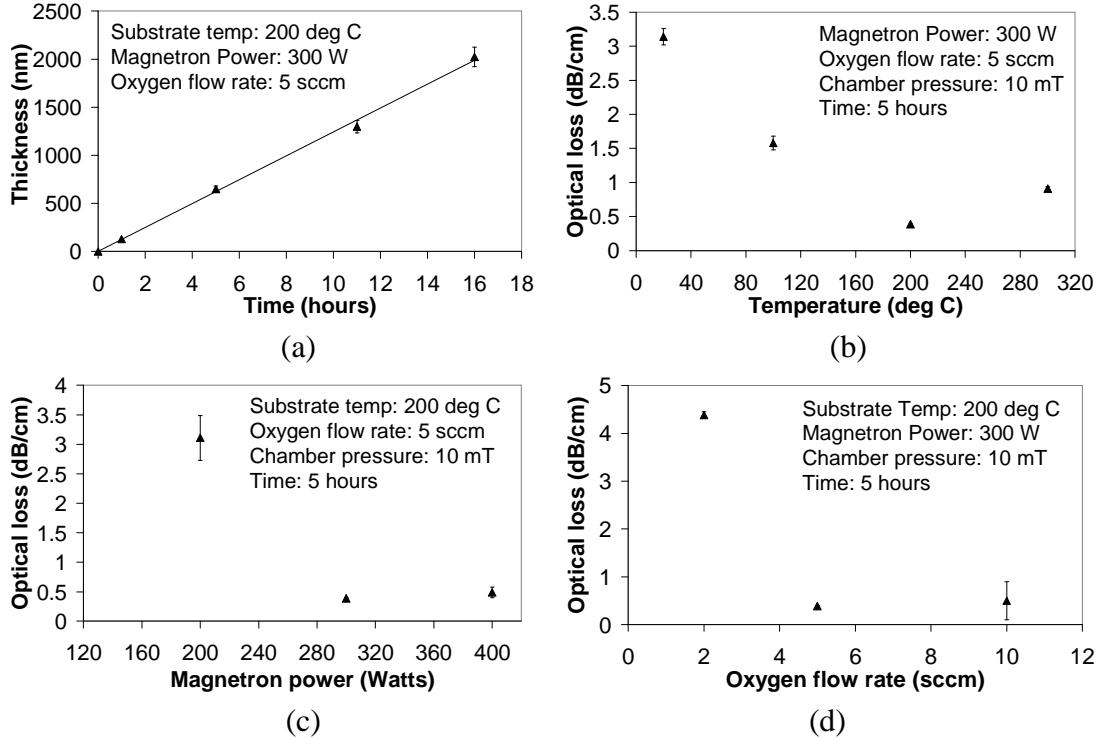


Fig 1 Er:Ta₂O₅ waveguide optimization a) Er:Ta₂O₅ deposition rate. b) Loss vs. substrate temperature, c) Loss vs. magnetron power, and d) Loss vs. oxygen flow rate.

Figure 1c shows the variation of optical loss with magnetron power, with low loss being achieved at 300W. Argon is used to start and maintain the plasma discharge and thus its flow rate is not critical, but the oxygen flow rate plays an important role in achieving low loss films as shown in figure 1d. With the increase in the O₂ flow rate, the film approaches its stoichiometric composition and hence lowest possible loss, but a further increase will lead to increased oxidisation of the target surface and an unacceptably low deposition rate. For our samples, a flow rate of 5sccm achieved the lowest loss value and a reasonable deposition rate. Ellipsometry measurements were performed on the sputtered Er:Ta₂O₅ samples to determine the thin film refractive index at various wavelengths in the visible region. Figure 2 shows the results for the fully optimized Er:Ta₂O₅ film.

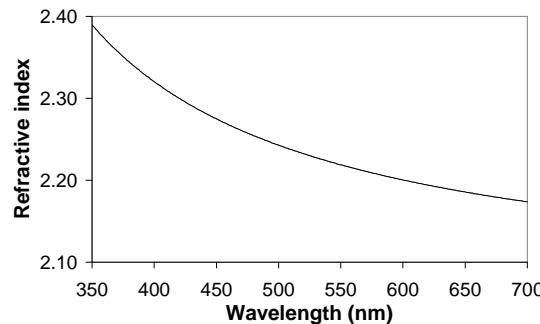


Fig 2. Refractive index of magnetron sputtered Er:Ta₂O₅ in the visible wavelength region.

2. Photoluminescence characterization

Photoluminescence measurements were performed at room temperature by pumping erbium ions into their $^4I_{11/2}$ level using a Ti:Sapphire laser emitting at 980nm. The thickness of the sample was approximately 2 microns. The power density was of the order of 1 kW/cm^2 (180mW total power), and was chopped at 25Hz. The luminescence was collected perpendicular to the sample, sent to a 30cm focal monochromator, and detected with an InGaAs photodiode through a lock-in amplifier. The resolution of the spectra was 10nm. Lifetime measurements were also performed with 0.2ms resolution. The photoluminescence spectra of annealed and non-annealed magnetron sputtered Er:Ta₂O₅ samples are shown in Figure 3a. Four samples were annealed at 450, 500, 550 and 600°C, respectively, in a tube furnace for one hour in oxygen. Higher temperatures were not employed as annealing above 600°C is expected to result in a lossy polycrystalline film [11].

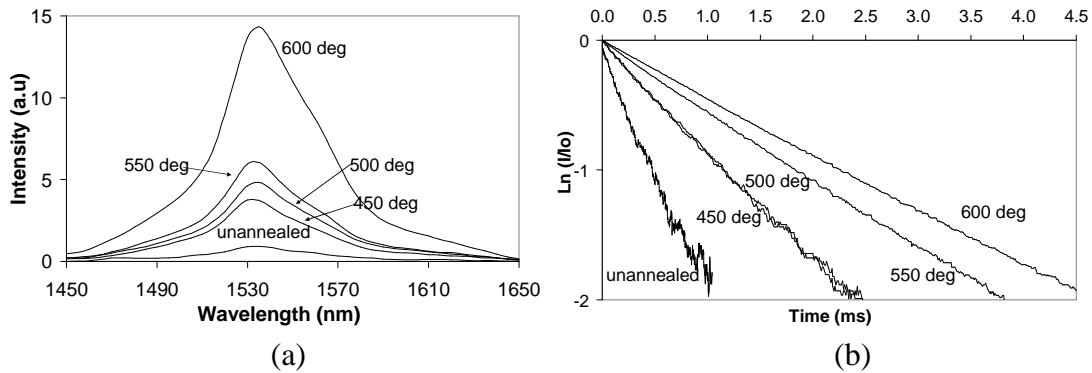


Fig 3 a) Photoluminescence spectra (bottom: unannealed, topmost: 600 °C) and b) Luminescence decay (leftmost: unannealed, rightmost: 600 °C) for annealed and unannealed samples.

The emission spectra correspond to the transition between the $^4I_{13/2}$ - $^4I_{15/2}$ levels of the Er³⁺ ion and peak at 1534nm. The bandwidth of the spectrum (FWHM) was measured to be 50nm which is substantially broader than those obtained from non-telluride glasses (~30nm) [12] and comparable to high index contrast hosts such as telluride glasses ($n\sim 2.1$, 65nm) [13] and alumina ($n\sim 1.69$, 55nm) [14] and shows potential for broadband applications. The photoluminescence intensity increases with annealing temperature to about 14 times that of the unannealed sample, at 600°C. The luminescence lifetime of the erbium ions is shown in Figure 3b and was found to increase from 0.53ms for the as-deposited sample to 2.4ms for the sample annealed at 600°C, where I/I_0 is the intensity normalized with respect to the maximum intensity. The relationship between the intensity and lifetime can be explained in terms of the decrease in the non radiative decay rate when annealed at higher temperatures. The 2.4ms lifetime decay is smaller than those obtained from non-telluride glasses (10-15ms) [12] and alumina (6ms) [14] but comparable to high refractive index hosts such as telluride glasses (3.5ms) [15] and zirconia ($n\sim 2.04$, 1.8ms) [2].

3. Conclusions

The deposition of Er:Ta₂O₅ by magnetron sputtering has been optimized to yield low loss slab waveguides (<0.4 dB/cm at 633nm) without annealing and losses can be expected to reduce further upon annealing at high temperatures, and with use at longer wavelengths. The refractive index of the thin film was determined over the wavelength range from 350nm to 700nm. A relatively broad photoluminescence spectrum

(FWHM~50nm) peaking at 1534nm was obtained, and a luminescence lifetime of 2.4ms was measured for the erbium ions in the Er:Ta₂O₅ film for optimised sputtering and annealing conditions. The results obtained for the losses and radiative lifetime are promising, for realizing erbium-doped integrated amplifier/laser and multifunctional photonic circuits based on Er:Ta₂O₅.

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References

- [1] G.N. van den Hoven, R.J.I.M. Koper, A. Polman, C. van Dam, J.W.M. van Uffelen and M.K. Smit, "Net optical gain at 1.53 microns in Er-doped Al₂O₃ waveguides on silicon," *Appl. Phys. Lett.*, vol. 68, pp. 1886-1888, 1996.
- [2] R. Schermer, W. Berglund, C. Ford, R. Ramberg and A. Gopinath, "Optical amplification at 1534 nm in erbium doped Zirconia waveguides," *IEEE J. Quantum Electron.*, vol. 39, pp. 154-159, 2003.
- [3] C.C. Baker, J. Heikenfeld, Z. Yu and J. Steckl, "Optical amplification and electroluminescence at 1.54 micron in Er-doped zinc silicate germanate on silicon," *Appl. Phys. Lett.*, vol. 84, pp. 1462-1464, 2004.
- [4] P. Nandi and G. Jose, "Erbium doped phospho-tellurite glasses for 1.5 micron optical amplifiers," *Opt. Commun.*, vol. 265, pp. 588-593, 2006.
- [5] B. Unal, M.C. Netti, M.A. Hassan, P.J. Ayliffe, M.D.B. Charlton, F. Lahoz, N.M.B. Perney, D.P. Shepherd, C.Y. Tai, J.S. Wilkinson and G.J. Parker, "Neodymium doped tantalum pentoxide waveguide lasers," *IEEE J. Quantum. Electron.*, vol. 41, pp. 1565-1573, 2005.
- [6] H. Rigneault, F. Flory, S. Monneret, S. Robert and L. Roux, "Fluorescence of Ta₂O₅ thin films doped by kilo-electron-volt Er implantation: application to microcavities," *Appl. Opt.*, vol. 35, pp. 5005-5012, 1996.
- [7] N. Maeda, N. Wada, H. Onoda, A. Maegawa, K. Kojima, "Preparation and optical properties of sol-gel derived Er³⁺-doped Al₂O₃-Ta₂O₅ films," *Opt. Mater.*, vol. 27, pp. 1851-1858, 2005.
- [8] P.S. Dobal, R.S. Katiyar, Y. Jiang, R. Guo and A.S. Bhalla, "Raman scattering study of a phase transition in tantalum pentoxide," *J. Raman Spectrosc.*, vol. 31, pp. 1061-1065, 2000.
- [9] C.Y. Tai, J.S. Wilkinson, N.M.B. Perney, M.C. Netti, F. Cattaneo, C.E. Finlayson and J.J. Baumberg, "Determination of nonlinear refractive index in a Ta₂O₅ rib waveguide using self-phase modulation," *Opt. Exp.*, vol. 12, pp. 5110-5116, 2004.
- [10] C.Y. Tai, C. Grivas and J.S. Wilkinson, "UV photosensitivity in a Ta₂O₅ rib waveguide Mach-Zender interferometer," *IEEE Photon. Tech. Lett.*, vol. 16, pp. 1522-1524, 2004.
- [11] P.C. Joshi and M.W. Cole, "Influence of postdeposition annealing on the enhanced structural and electrical properties of amorphous and crystalline Ta₂O₅ thin films for dynamic random access memory applications," *J. Appl. Phys.*, vol. 86, pp. 871-880, 1999.
- [12] P.M. Peters, D.S. Funk, A. P. Peskin, D.L. Veasey, N.A. Sanford, S.N. Houde-Walter, and J.S. Hayden, "Ion-exchanged waveguide lasers in Er³⁺/Yb³⁺ codoped silicate glass," *Appl. Opt.*, vol. 38, pp. 6879-6886, 1999.
- [13] R. Rolli, M. Montagna, S. Chaussédent, A. Monteil, V.K. Tikhomirov and M. Ferrari, "Erbium-doped tellurite glasses with high quantum efficiency and broadband stimulated emission cross section at 1.5 μ m," *Opt. Mater.*, vol. 21, pp. 743-748, 2003.
- [14] G.N. van den Hoven, E. Snoeks, A. Polman, J.W.M. van Uffelen, Y.S. Oei, and M.K. Smit, "photoluminescence characterization of Er-implanted Al₂O₃ thin films," *Appl. Phys. Lett.*, vol. 62, pp. 3065-3067, 1993.
- [15] H. Yamauchi, G.S. Murugan, and Y. Ohishi, "Optical properties of Er³⁺ and Tm³⁺ ions in a tellurite glass," *J. Appl. Phys.*, vol. 97, pp. 043505-1 -043505-8, 2005.